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Dann, E., Gibson, E., Blackmore, R., Catlow, R., Collier, P., Chutia, A., Eralp Erden, T., Hardacre, C., Kroner, A., Nachtegaal, M., Raj, A., Rogers, S., Taylor, R., Thompson, P., Tierney, G., Zeinalipour-Yazdi, C., Goguet, A., & Wells, P. (2019). Structure Selectivity of Supported Pd nanoparticles for Catalytic NH3 Oxidation resolved using combined Operando Spectroscopy. *Nature Catalysis*, *2*, 157–163. https://doi.org/10.1038/s41929-018-0213-3

### Published in:

Nature Catalysis

### **Document Version:** Peer reviewed version

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# Structure Selectivity of Supported Pd nanoparticles for Catalytic NH<sub>3</sub> Oxidation resolved using combined *Operando* Spectroscopy

Ellie K. Dann<sup>a, b</sup>, Emma K. Gibson<sup>b, c</sup>, Rachel H. Blackmore<sup>b, d</sup>, C. Richard A. Catlow<sup>a, b, e</sup>, Paul Collier<sup>f</sup>, Arunabhiram Chutia<sup>b, g</sup>, Tugce Eralp Erden<sup>f</sup>, Christopher Hardacre<sup>b, h</sup>, Anna Kroner<sup>i</sup>, Maarten Nachtegaal<sup>j</sup>, Agnes Raj<sup>f</sup>, Scott M. Rogers<sup>a, b</sup>, S. F. Rebecca Taylor<sup>h</sup>, Paul Thompson<sup>k</sup>, George F. Tierney<sup>b, d</sup>, Constantinos D. Zeinalipour-Yazdi <sup>b, l</sup>, Alexandre Goguet<sup>b, m</sup> \* and Peter P. Wells<sup>b, d, i</sup> \*

<sup>a</sup> Department of Chemistry, University College London, 20 Gordon Street, London WC1H 0AJ

<sup>b</sup> UK Catalysis Hub, Research Complex at Harwell, Rutherford Appleton Laboratory, Oxfordshire, OX11 0FA, United Kingdom

<sup>c</sup> School of Chemistry, University of Glasgow, Joseph Black Building, University Avenue, Glasgow, G12 8QQ, United Kingdom

<sup>d</sup> University of Southampton School of Chemistry, University Road, Southampton, SO17 1BJ, United Kingdom

- e Cardiff Catalysis Institute, School of Chemistry, Cardiff University, Main Building, Park Place, Cardiff, CF10 3AT, United Kingdom
- <sup>f</sup> Johnson Matthey Technology Centre, Blounts Court Road, Sonning Common, Reading, RG4 9NH, United Kingdom
- <sup>g</sup> College of Science, University of Lincoln, Brayford Pool Campus, Lincoln, LN6 7TS, United Kingdom
- <sup>h</sup> School of Chemistry, The University of Manchester, Oxford Road, Manchester, M13 9PL, United Kingdom
- <sup>1</sup>Diamond Light Source Ltd., Harwell Science and Innovation Campus, Chilton, Didcot OX11 0DE, United Kingdom
- <sup>j</sup> Paul Scherrer Institute, WLGA/ 219, 5232 Villigen PSI, Switzerland
- <sup>k</sup> XMaS UK CRG Beamline, ESRF The European Synchrotron, 71, avenue des Martyrs, 38000 Grenoble, France.
- <sup>1</sup>School of Science, University of Greenwich, Central Avenue, Chatham Maritime, Kent ME4 4TB, UK

<sup>m</sup> Queen's University Belfast School of Chemistry, David Keir Building, Stranmillis Rd, Belfast BT9 5AG, United Kingdom.

The selective catalytic oxidation of NH<sub>3</sub> to N<sub>2</sub> (NH<sub>3</sub>-SCO) presents a promising solution for abatement of unused NH<sub>3</sub>-based reductants from diesel exhaust after treatment systems. Supported Pd nanoparticle catalysts show selectivity to N<sub>2</sub>, rather than NO<sub>x</sub>, which is investigated in this work. The link between Pd nanoparticle structure and surface reactivity is found by *operando* X-ray absorption Fine Structure (XAFS) spectroscopy, diffuse reflectance infrared Fourier transformed spectroscopy (DRIFTS) and on-line mass spectrometry. Nitrogen insertion into the metallic Pd nanoparticle structure at low temperatures (< 200°C) is found to be responsible for high N<sub>2</sub> selectivity; whereas the unfavourable formation of NO is linked to adsorbed nitrates which form at the surface of bulk PdO nanoparticles at high temperatures (> 280°C). Our work demonstrates the ability for combined *operando* spectroscopy and DFT calculations to characterise a previously unidentified PdN<sub>x</sub> catalyst, and clarify the selectivity directing structure of supported Pd catalysts for NH<sub>3</sub>-SCO.

The selective catalytic oxidation of ammonia (NH<sub>3</sub>-SCO) is needed to eliminate the slip of unreacted NH<sub>3</sub> from automotive exhaust. NH<sub>3</sub> is used as the reductant for the selective catalytic reduction (NH<sub>3</sub>-SCR) of harmful nitrogen oxides (NO<sub>4</sub>) that are released from diesel engines. NH<sub>3</sub> slip from these exhaust systems is increasing as legislation pushes for greater efficiency of de-NO<sub>4</sub> technology, which is achieved using greater amounts of the NH<sub>3</sub> reductant. High performance catalysts for the NH<sub>3</sub>-SCO reaction downstream of the NH<sub>3</sub>-SCR must be able to achieve complete conversion of NH<sub>3</sub> to N<sub>2</sub> and H<sub>2</sub>O, at low temperatures ( $T < 400^{\circ}$ C) and without the reformation of harmful NO<sub>4</sub>.

Noble metals supported on metal oxides have repeatedly been found to possess high catalytic activity for NH<sub>3</sub> oxidation at low temperatures, but over oxidation to nitrogen oxides is problematic.1-7 Various attempts to improve selectivity and avoid the use of noble metals have investigated the use of base metal oxides and transition metal ion exchanged zeolites.<sup>8-10</sup> However, even the more selective of these catalysts,  $CuO_{,11,12}^{,11,12}$  and  $V_{2}O_{5,13}^{,13}$ have lacked sufficient activity to be of practical use.<sup>14</sup> Of all the noble metals, the most promising  $N_{\rm 2}$  selectivity is found for  $\mathsf{Pd}$ based catalysts.<sup>15</sup> Various studies with Pd catalysts have examined incorporation of Pd into multicomponent metal systems, 3,16 mixed metal oxide perovskite materials17, the use of various supports<sup>18</sup> and varying metal loading<sup>19</sup>, concluding that Pd nanoparticles (NPs) are the active species irrespective of the support.<sup>19,20</sup> Despite these studies, there remains little understanding of the fundamental chemistry that governs the catalytic activity of Pd NPs for NH<sub>3</sub>-SCO.

The use of Pd NPs in catalysis is well established and it is accepted that Pd NP structure can change under different reaction environments. For example, the bulk oxidation and

reduction of Pd NPs is demonstrated by switching the oxygen/fuel ratio during lean/rich methane oxidation.<sup>21</sup> The formation of palladium hydrides and carbides is also reported during operation of hydrogenation reactions, which are found to affect the catalytic performance.<sup>22-24</sup> However there is a lack of evidence regarding the noble metal structure and speciation for the NH<sub>3</sub>-SCO reaction. Zhang *et al.* have suggested that the oxidation state of the precious metal catalyst influences the NH<sub>3</sub>-SCO activity, but without *in situ* measurements it is difficult to draw definite conclusions.<sup>25</sup> Furthermore, there is no explanation for the increasing formation of nitrogen oxides at higher temperatures.<sup>20</sup>

These problems highlight the importance of in situ methods in studying the catalyst under real working environments. The high energy of the Pd K-edge (24357 eV), is advantageous for in situ and operando X-ray absorption fine structure (XAFS) spectroscopy, which is able to penetrate the reaction medium and probe the 1s to 4p electronic transitions of Pd atoms in the supported NP catalysts. X-rays at the Pd L<sub>3</sub>-edge (3173 eV) have lower penetrating power, but can also be used under carefully controlled in situ conditions to probe the 2p to d electronic transitions, giving information regarding the empty d density of states which is crucial for distinguishing between hydride, carbide and other potential phases.<sup>26</sup> Diffuse reflectance infrared Fourier transform spectroscopy (DRIFTS) may also be applied as a complementary in situ method, to provide information for the reaction mechanism by probing molecular vibrations of adsorbed intermediates at the catalyst surface.

In this study two high performance supported Pd NP catalysts  $(Pd/\gamma-Al_2O_3 \text{ and }Pd/zeolite-Y)$  are investigated for  $NH_3$ -SCO using a combination of XAFS with DRIFTS and on-line mass spectrometry (MS). By performing simultaneous XAFS and

DRIFTS spectroscopy of the catalyst under operating conditions, links between bulk nanoparticle structure, reaction mechanisms and catalytic activity are revealed, which are also supported by our theoretical calculations based on density functional theory and dispersion corrections (DFT+D3). In this way, not only have we identified the Pd structural changes and formation of a previously unidentified Pd nitride that occur within the temperature range for NH<sub>3</sub>-SCO; but we can also correlate these with the change in surface speciation, catalytic reactivity and product distribution.

### Results

*Ex situ* characterisation Two supported 1.5wt% Pd catalysts were prepared by incipient wetness impregnation. Pd K-edge XANES of the supported Pd catalysts after impregnation and calcination were consistent with that of bulk PdO (Supplementary Figure 1). Transmission electron microscopy (TEM) images of the as prepared catalysts show PdO nanoparticles supported on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (Supplementary Figure 2) and zeolite-Y (Supplementary Figure 3). PdO nanoparticles can be observed as dark spots contrasted against the support, which appear to have smaller average particle size when supported on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> compared with zeolite-Y.

After a reduction pre-treatment in 10% H<sub>2</sub>/He, the Pd K-edge XANES of the supported Pd catalysts can be matched to that of Pd foil (Supplementary Figure 4), with small differences attributed to particle size and temperature effects.<sup>27</sup> By fitting the Fourier transformed Pd K-edge EXAFS spectra of the reduced catalysts (Supplementary Figure 5 and Supplementary Table 1) to a model using a single Pd-Pd scattering path, the Pd coordination number of Pd/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> and Pd/zeolite-Y were found as 9.7 and 11.3, respectively. These values correspond to average particle diameters of 2. 1 nm and 5.2 nm, respectively, calculated using a method reported previously,<sup>28</sup> and are consistent with the TEM images.

**Catalytic Activity.** NH<sub>3</sub> conversion and product selectivity of the reduced Pd catalysts during the *operando* XAFS/DRIFTS/MS experiment were calculated from *m*/z signals of NH<sub>3</sub>, N<sub>2</sub>, NO, N<sub>2</sub>O and NO<sub>2</sub> (*m*/z = 17, 28, 30, 44 and 46) relative to the inlet NH<sub>3</sub> signal, and are plotted in Figure 1 for Pd/γ-Al<sub>2</sub>O<sub>3</sub> and Pd/zeolite-Y. Both catalysts exhibit similar catalytic activity, achieving 50% NH<sub>3</sub> conversion at 213°C (1.5wt% Pd/γ-Al<sub>2</sub>O<sub>3</sub>) and 190°C (1.5wt% Pd/γ-

During the mid-temperature region,  $T_2$ ,  $N_2$  formation reached a plateau and the formation of  $N_2O$  increased steadily. The start of the high temperature region,  $T_3$ , was defined by onset of NO formation and decrease in  $N_2O$  formation. Due to similar properties of both Pd NP catalysts in this study, only results from the Pd/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> catalyst will be discussed hereafter (results from Pd/zeolite-Y are shown in Supplementary Information).

Pd L<sub>3</sub>-edge XANES. In situ Pd L<sub>3</sub>-edge XANES were performed to determine the Pd electronic structure during NH<sub>3</sub>/O<sub>2</sub> reaction conditions at low temperature. The Pd L<sub>3</sub>-edge XANES of Pd/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> are shown in Figure 2 before and after switching to NH<sub>3</sub>/O<sub>2</sub> atmosphere, together with a reference spectrum of Pd hydride collected by cooling the sample to room temperature in 5% H<sub>2</sub>/He. Palladium hydride gives rise to a distinct absorption

feature at 3182 eV due to excitation of 2p electrons to a Pd-H antibonding state (Figure 2).<sup>29-31</sup> The XANES spectrum of Pd/ $\gamma$ -



Figure 1. Catalytic activity of supported Pd catalysts for  $NH_3$ oxidation at increasing temperatures. a, 1.5wt% Pd/zeolite-Y. b, 1.5wt% Pd/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub>. Reactant gas feed: 0.5%  $NH_3$ , 2.5% O<sub>2</sub>, 97% He.

 $AI_2O_3$  collected under He at 100°C, after a reduction pretreatment (Figure 2), is consistent with that of Pd foil for metallic Pd<sup>(0)</sup> oxidation state. After switching to the reactant (NH<sub>3</sub>/ O<sub>2</sub>) gas feed at 100°C, the Pd L<sub>3</sub>-edge XANES show increased intensity and broadening of the white line (Figure 2) but without the characteristic absorption feature at 3182 eV for palladium hydride.

The Pd L<sub>3</sub>-edge XANES spectra confirm there is a change to the Pd structure upon introducing the reactant (NH<sub>3</sub>, O<sub>2</sub>) gas feed that cannot be attributed to formation of palladium hydride. The increase in intensity and broadening of the white line intensity of Pd L<sub>3</sub>-edge XANES has previously been reported for the *in situ* formation of a carbide like phase.<sup>22</sup> However, the absence of carbon in this system leads us to infer the formation of another type of interstitial structure, palladium nitride.<sup>32</sup> Upon repeating this procedure with a gas feed of 0.5% NH<sub>3</sub>/He, the same result is obtained (Supplementary Figure 6), showing that the same structural phase is also formed without the presence of O2. Further evidence of the Pd nitride species was inferred from XPS spectra of the sample after treating with NH<sub>3</sub> gas. Deconvoluting the peak in the N1s region identifies the presence of two nitrogen species at 402 eV and 398 eV (Supplementary Figure 7). The N1s species located at 400 eV is consistent with NH<sub>3</sub> molecular species, whereas the species located at 395 eV is consistent with the formation of a reduced nitrogen as in metal nitrides.<sup>32,33</sup>



Figure 2. In situ Pd L<sub>3</sub>-edge XANES of Pd/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> in different gas environments. XANES collected after in situ reduction in H<sub>2</sub> at 200°C, then exposure to a) inert He at 100°C (black), b) 10 % H<sub>2</sub> at room temperature (magenta) and c) 0.5% NH<sub>3</sub>/2.5%O<sub>2</sub>/He at 100°C (green).

Operando Pd K-edge XAFS. The change in electronic structure of the Pd species upon exposure to the reactant NH<sub>3</sub>/O<sub>2</sub>/He gas feed is confirmed again by comparison of the Pd K-edge XANES recorded before and after gas switching from inert to NH<sub>3</sub>/O<sub>2</sub>/He, as seen in Figure 3a. On addition of the reactant  $NH_3/O_2/He$  gas feed, the near edge features at 24393 eV and 24432 eV shifted by ca. 4 eV to lower energies. These observations could be reproduced by exposing the catalyst to a 0.5%NH<sub>3</sub>/He gas feed (Supplementary Figure 9). As oxygen is not required to impart the Pd structural change observed by XANES, we can rule out that these observations are a consequence of oxidation. Moreover, a surface oxidation would result in a different XANES spectrum from that observed, as shown later in Figure 4. The observed change in Pd K-edge XANES features has also been reported for heteroatom insertion inside the metal Pd fcc framework of supported Pd NPs during in situ formation of a palladium carbide phase, suggesting similar structural and electronic changes to the Pd nanoparticles in this case.  $^{\rm 24,26}$  These observations again support the proposed formation of an interstitial compound.

The Pd K-edge extended X-ray absorption fine structure (EXAFS) gives information on the average local atomic environment of the absorbing palladium atoms in the sample. Major disruption to the local atomic arrangement of the metallic Pd NPs upon introducing the NH<sub>3</sub>/O<sub>2</sub> reactant gas mixture (and NH<sub>3</sub>/He only) can be observed by decreased amplitude and phase shift of the  $k^2$  weighted Pd K-edge EXAFS oscillations, seen in Figure 3b, which shows that there must be substantial heteroatom insertion not only to cause expansion of the fcc lattice but also to cause significant disruption of the local atomic ordering. The Fourier transformed EXAFS spectrum (Figure 3c) can be fitted with a model using first shell coordination to nitrogen ( $R_{Pd-N} = 2.00$  Å) and a second shell coordination to palladium ( $R_{Pd-Pd} = 2.81$  Å), with coordination numbers of 1 and 9.9, respectively (Table 1).

The Pd K-edge XANES spectra collected during the high temperature NH<sub>3</sub> oxidation regime are plotted in Figure 4. The spectrum collected at 240°C (plotted in green, Figure 4) shows that the XANES features return to energies that resemble the initial Pd<sup>(0)</sup> spectrum collected under inert atmosphere (plotted in black, Figure 4), but with a slight increase in white line intensity and dampening of the metallic feature at 24393 eV.

**Table 1.** Pd K-edge EXAFS fitting parameters of 1.5wt% Pd/ $\gamma$ -Al2O3 at 100°C in 0.5% NH3/2.5%O2/He.

| Abs -   | N      | R/Å      | σ² / Ų    | <i>E₀</i> / eV | R      |
|---|--------|----------|-----------|----------------|--------|
| Scat  |        |          |           |                | factor |
|   |        |          |           |                |        |
| Pd-N  | 1.0(2) | 2.00(1)  | 0.003(2)  | -5.7(4)        | 0.02   |
| Pd-Pd   | 9.9(1) | 2.810(5) | 0.0161(8) | 1              |        |
| Note: Fitting parameters $S_0^2 = 0.8$ ; Fit range $3 < k / Å^2 < 10.6$ , $1 < 0.6$ |        |          |           |                |        |

R / Å < 3; Number of independent points = 9.

This flattening of the metallic feature and increase in white line intensity can result from temperature differences, as well as partial oxidation of  $Pd^{(0)}$  to  $Pd^{2+}$  due to PdO formation. Linear combination fitting (detailed in Supplementary Table 3 and Supplementary Figure 11) describes the Pd K-edge XANES collected at 240°C as an approximate summation of ~40%  $Pd^{(0)}$ and ~60%  $Pd^{2+}$  species. This fitting used the Pd K-edge XANES spectra collected at 100°C in He and at 300°C in NH<sub>3</sub>/O<sub>2</sub> to represent Pd<sup>(0)</sup> and Pd<sup>2+</sup> species of this particle size, respectively. The ratio of 40%  $Pd^{(0)}$  and 60%  $Pd^{2+}$  is expected for a thin oxidic surface layer of supported Pd NPs of approximately 2 nm diameter, as reported previously.<sup>34</sup> The greater ratio of Pd located at the surface of small NPs has significant contribution to the averaged Pd signal of the measured XANES spectrum.

The Fourier transformed EXAFS of the sample at 240°C, shown in Supplementary Figure 12, shows scattering features at 1.8 and 2.6 Å for first shell Pd-O and Pd-Pd neighbours, respectively, but absence of a second shell Pd-Pd<sub>(2)</sub> scattering feature related to a bulk Pd oxide, expected at 3.01 Å. These observations confirm that any Pd-O formation at 240°C is short range and limited only to the Pd NP surface.

At temperatures beyond 240°C, the white line intensity of Pd Kedge XANES spectra increase further and metallic features diminish such that the XANES spectrum collected at 300°C (plotted in orange, Figure 4) resembles that of a bulk PdO phase. The Fourier transformed Pd K-edge EXAFS spectra (Supplementary Figure 12) show major transition from metallic NP structure at 240°C to oxide NP structure at 300°C, and growth in amplitude of the oxide scattering features from 300°C to 340°C. The Fourier transformed EXAFS of the sample at 340°C can be fitted to a model using scattering paths of Pd-O ( $R_{Pd-O}$  = 2.01 Å), Pd-Pd<sub>(1)</sub> ( $R_{Pd-Pd1}$  = 3.01 Å) and Pd-Pd<sub>(2)</sub> ( $R_{Pd-Pd2}$  = 3.455 Å), with coordination numbers 3.7(3), 1.0(9) and 2(2), respectively (Supplementary Figure 13 and Supplementary Table 4).



Figure 3. Operando Pd K-edge XANES and EXAFS of Pd/y-Al<sub>2</sub>O<sub>3</sub> in different reaction conditions. *a*, Pd K-edge XANES of 1.5wt% Pd/y-Al<sub>2</sub>O<sub>3</sub> after reduction pre-treatment in H<sub>2</sub>, then exposure to a) He at 100°C (black), b) 0.5% NH<sub>3</sub>/2.5% O<sub>2</sub>/He at 100°C (red) and c) 0.5% NH<sub>3</sub>/2.5% O<sub>2</sub>/He at 200°C (blue). *b*, Pd K-edge k<sup>2</sup> weighted EXAFS of 1.5wt% Pd/y-Al<sub>2</sub>O<sub>3</sub> after reduction pre-treatment, then exposure to a) He (black) and b) 0.5% NH<sub>3</sub>/2.5% O<sub>2</sub>/He (red) and c) 0.5% NH<sub>3</sub>/He (blue) at 100°C. *c*, Stacked plot of magnitude (top) and imaginary (bottoms) non-phase corrected Fourier transformed Pd K-edge EXAFS of 1.5wt% Pd/y-Al<sub>2</sub>O<sub>3</sub> at 100°C in 0.5% NH<sub>3</sub>/2.5% O<sub>2</sub>/He atmosphere, plotted with fit constructed from Pd-N and Pd-Pd scattering paths.

*Operando* Pd K-edge XAFS have therefore identified three structural phases of Pd NPs during NH<sub>3</sub> oxidation. The first is a PdN<sub>x</sub> species that forms during the low temperature regime (100°C  $\leq T_1 \leq 200$ °C), where the main product of NH<sub>3</sub> oxidation is N<sub>2</sub>. The second is a Pd NP with bulk metallic structure and a surface oxide that forms in the mid-temperature regime (240°C  $< T_2 < 300$ °C), where an increasing amount of NH<sub>3</sub> is oxidised to N<sub>2</sub>O. The third structural phase is a bulk PdO NP formed at high temperature ( $T_3 > 300$ °C), which is linked to the production of NO.



Figure 4. Operando Pd K-edge XANES of Pd/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> in different reaction conditions. Pre-treatment in H<sub>2</sub>, then exposure to He at 100°C (black), 0.5% NH<sub>3</sub>/2.5% O<sub>2</sub>/He at 240°C (green) and 300°C (orange). Pd K-edge XANES of bulk PdO at ambient conditions shown as reference (blue, dashed).

Operando DRIFTS. Figure 5 shows the difference DRIFTS spectrum of catalyst Pd/y-Al<sub>2</sub>O<sub>3</sub>, collected simultaneously with the XAFS, after switching from inert He to the reactant  $NH_3/O_2$ gas feed at 100°C. The sharp bands between 1250 and 1600 cm<sup>-</sup> <sup>1</sup> are typical of NH<sub>3</sub> adsorbed to surface acid sites. The bands at 1596 cm<sup>-1</sup> and 1256 cm<sup>-1</sup> are assigned to asymmetric and symmetric deformation of coordinated ammonia, respectively.<sup>14-17</sup> The band at 1450 cm<sup>-1</sup> can be assigned to asymmetric deformation of ammonium ions adsorbed on Brønsted sites of the support, <sup>14,18-20</sup> and the shoulder at 1490 cm<sup>-</sup> <sup>1</sup> to NH<sub>3</sub> coordinated to Pd.<sup>11</sup> Low intensity combination bands of ammonia and ammonia dimers may give rise to the broad band

around 2470 cm<sup>-1</sup>.<sup>35</sup> Absorption bands between 3080 and 3480 cm<sup>-1</sup> are typical of symmetric and asymmetric N-H stretching of coordinated ammonia.  $^{\rm 36,37}$  Negative bands above 3500  $\rm cm^{-1}$ show the absence of O-H stretches due to NH4<sup>+</sup> occupying surface hydroxyl Brønsted acid sites. The dissociative adsorption of ammonia on the Pd nanoparticle surface is shown by a weak band at 1377 cm<sup>-1</sup> which is attributed to the -NH<sub>2</sub> wagging frequency<sup>12,38</sup>, usually coupled with the band at 1596 cm<sup>-1</sup>. The broad, low intensity band at 1850 cm<sup>-1</sup> is attributed to small amounts of neutral and positively charged nitrosyl (NO<sup>+</sup>) species adsorbed on  ${\rm Pd}^{(0)}{}^{.39}$  The broad band of small intensity at 2170 cm<sup>-1</sup> is present in the DRIFTS spectrum of both catalysts after switching to the NH<sub>3</sub>/O<sub>2</sub> reactant gas feed at 100°C, and when switching to an NH<sub>3</sub>/He atmosphere (Supplementary Figure 14). Absorption bands at this wavenumber have previously been associated with the vibrational frequency of an asymmetric N-N bond of bridging N<sub>2</sub> in multinuclear metal-dinitrogen complexes.<sup>22</sup> The observation of vibrational bands near 2200 cm<sup>-</sup> <sup>1</sup> during NH<sub>3</sub> oxidation conditions are sometimes attributed to the N-N stretch of adsorbed  $N_2O$  species.<sup>40,41</sup> However, the absence of a corresponding N-O stretch for N<sub>2</sub>O (expected at 1262 – 1220 cm<sup>-1</sup>) and absence of N<sub>2</sub>O detected in mass spectrometry signals at 100°C means that the band at 2170 cm<sup>-1</sup> cannot be assigned to N2O species at this temperature. Similar vibrational bands at 2170 cm<sup>-1</sup> are observed in the DRIFTS spectrum of 1.5wt% Pd/Zeolite under analogous conditions (Supplementary Figure 19 and Supplementary Figure 20) and are reported for  $\mathsf{NH}_3$  adsorption over an  $\mathsf{Ag}/\mathsf{Al}_2\mathsf{O}_3$  catalyst between 160 - 220°C, which also cannot be attributed to N-N stretching of  $N_2O.^{37}$  We propose that this low intensity absorption band results from an asymmetric stretch of dinitrogen species at the catalyst surface, following the dissociated adsorption of ammonia molecules. An absorption band at 1455 cm<sup>-1</sup> due to -NH wagging at the sample surface decreases in intensity from 200°C to 220°C followed by formation of small intensity bands at 1418, 1302 and 1548 cm<sup>-1</sup> (Supplementary Figure 15) due to the formation of adsorbed nitrate species.37,42-44 The intensity of these nitrate bands increase sharply at 280°C, suggesting the onset of NO formation at the catalyst surface which are then desorbed and detected in the mass spectrometer from 300°C.



Figure 5. Operando DRIFTS spectrum of  $Pd/\gamma$ -Al<sub>2</sub>O<sub>3</sub> during NH<sub>3</sub> oxidation. Difference DRIFTS spectrum between spectrum collected after reduction pre-treatment in He and in 0.5% NH<sub>3</sub>/2.5% O<sub>2</sub>/He reactant gas at 100°C.

DFT + D3 Calculations. To clarify how the presence of interstitial N-atoms influence the Pd-Pd bond distances, DFT+D3 calculations were performed. These calculations were done in the Pd(111) system with 0.12 wt%, 0.23 wt%, 0.47 wt%, 0.70 wt%, and 1.39 wt% of interstitial N-atoms (optimised structures shown in Supplementary Figure 21), which showed that the average Pd–Pd distances in the vicinity of interstitial N-atoms are ~2.83 Å. This is consistent with the lattice expansion observed by Pd K-edge EXAFS during N-atom insertion under NH<sub>3</sub>/O<sub>2</sub> reaction conditions. The lateral and vertical Pd-Pd distances were found to be in the range of 2.83 - 2.85 Å and 2.80 - 2.83 Å, respectively, irrespective of the concentration of N-atom insertion. The Pd-N distances were in the range 1.91 – 1.99 Å, which is in agreement with the distances found from the Pd K-edge EXAFS data. Before relaxation, distances between the interstitial N-atoms in all the above models were in the range of 2.39 - 2.47 Å and after relaxation these moved by a distance approximately in the range of 3.03 – 3.34 Å, with some of the N-atoms moving closer to the surface. The diffusion barrier for an interstitial N-atom to move from an octahedral interstitial site to a surface 3-fold hollow was low (0.25 eV), suggesting that interstitial N-atoms are able to diffuse to the surface to participate in the catalytic mechanism.

The adsorption properties of  $NH_3$  on Pd(111) surface were considered with and without the presence of interstitial Natoms. The adsorption energies of NH3 molecule close to the interstitial N-atom is more favourable as compared to away from it in the same system, and also in comparison to the pristine Pd(111) surface by ~0.156 eV (all calculated adsorption energies of NH<sub>3</sub> on Pd(111) surface with and without interstitial N-atoms are given in Supplementary Tables 6, 7 and 8). This is due to the interaction of N p and Pd d molecular orbitals in the bonding region. For pristine Pd(111) surface the Pd  $d_{z^2}$  orbital interacts with the N p orbitals whereas in the case of a Pd(111) surface with an interstitial N-atom the Pd  $d_{z^2}$  and  $d_{yz}$  interact with the N  $\ensuremath{\textit{p}}$  orbital leading to a stronger bonding interaction between Pd atoms with NH<sub>3</sub> molecules. In the latter case there is also trace contribution from other d orbitals. From this analysis we can conclude that the Pd(111) surface with interstitial N-atoms would have greater coverage of adsorbed NH<sub>3</sub> as compared to the pristine Pd(111) surface.

### Discussion

Three distinct regions of catalytic activity are identified for  $NH_3$  oxidation over supported Pd NP catalysts. During the low temperature region ( $100^{\circ}C \le T_2 \le 200^{\circ}C$ ), the Pd K-edge and Pd

L<sub>3</sub>-edge XANES show that the palladium structure experiences lattice expansion with insertion of nitrogen into the metallic Pd NP structure, which is confirmed by our DFT+D3 calculations. The dissociation of NH<sub>3</sub> during this temperature region is shown by  $-NH_2$  scissoring (at 1377 cm<sup>-1</sup>), -NH wagging (1455 cm<sup>-1</sup>) and a low intensity vibrational band at 2170 cm<sup>-1</sup> for N-N stretching at the catalyst surface. The greater stability of NH<sub>3</sub> adsorption on the Pd nitride NP surface is calculated from molecular models and infers a high NH<sub>3</sub> surface coverage. Additionally, the low energy barrier for diffusion of interstitial N to the surface indicates the ability for interstitial N to participate in the reaction mechanism by coupling with adsorbed NH<sub>3</sub> at the NP surface in a Mars Van Krevelen type mechanism. Indeed, a small contribution of m/z 28 for N<sub>2</sub> gas is detected by on-line MS at 100°C, which increases sharply at 150°C in line with NH<sub>3</sub> conversion. By correlating the N-N surface species, Pd NP structure and  $N_{\rm 2}$  formation (Supplementary Figure 22) the N-N surface species are only observed when the Pd nanoparticles exist as PdN<sub>x</sub>, and the desorption of such species coincides with the onset of N<sub>2</sub> formation. Following this, the contraction of the Pd distance - indicating the removal of interstitial N from the Pd lattice - is found to occur when N2 formation exceeds 50%, relative to the inlet NH<sub>3</sub>. This suggests the involvement of interstitial N in the reaction mechanism for the selective oxidation of  $NH_3$  to  $N_2$  (Equation 1), which are depleted when the rate of consumption is greater than the rate of replacement.

 $4NH_3 + 3O_2 \rightarrow 2N_2 + 6H_2O$  (1)

The active PdN<sub>x</sub> NP structure is observed by *operando* Pd K-edge and Pd L<sub>3</sub>-edge XANES up to 200°C. At temperatures beyond 200°C, the interstitial N is removed from the Pd lattice and the Pd K-edge XANES show that the structure of the Pd NPs return to predominantly a metallic oxidation state, with a small oxidic surface layer. During this period DRIFTS spectra show the presence of 2240 and 2210 cm<sup>-1</sup> absorption bands for N<sub>2</sub>O molecular vibrations beyond 180°C and there is an increase in N<sub>2</sub>O gas reaching the mass spectrometer. Therefore an increasing number of NH<sub>3</sub> molecules are oxidised to N<sub>2</sub>O at the Pd surface, as per Equation 2. The formation of N<sub>2</sub>, as detected by MS can result from the NH<sub>3</sub>-SCR reaction of adsorbed N<sub>2</sub>O with incoming gaseous NH<sub>3</sub>, as in Equation 3.<sup>45</sup>

 $2NH_3 + 2O_2 \rightarrow N_2O + 3H_2O$  (2)  $2NH_3 + 3N_2O \rightarrow 4N_2 + 3H_2O$  (3)

The presence of surface nitrate absorption bands are first observed in the DRIFTS spectra (as absorption bands at 1418, 1302 and 1548 cm<sup>-1</sup>) at 280°C and increase significantly in intensity at higher temperatures. Nitrate formation at the catalyst surface correlates with the temperature for bulk oxidation of the Pd NPs, observed by change in Pd K-edge XANES, and is followed by onset of NO gas, observed by MS, at temperatures  $\geq$  300°C. The complete oxidation of the Pd NPs is confirmed by fitting the Pd K-edge EXAFS data collected at 340°C to Pd-O and Pd-Pd paths of PdO. The over oxidation of NH<sub>3</sub> to NO, by Equation 4, has previously been attributed to high oxygen coverage at the precious metal surface which prevents N coupling,<sup>23</sup> but this study indicates that bulk oxide precious metal NPs are the active catalytic species for NH<sub>3</sub> oxidation to NO above 280°C.

$$4NH_3 + 5O_2 \to 4NO + 6H_2O \quad (4)$$

The powerful combination of XAFS and DRIFTS spectroscopy, together with online mass spectrometry and DFT calculations

have allowed thorough characterisation of the structure and surface reactivity of supported Pd NPs during reaction for NH<sub>3</sub>-SCO. The use of in situ Pd L<sub>3</sub>-edge and operando Pd K-edge XAFS have identified a PdN<sub>x</sub> species in the presence of NH<sub>3</sub>/O<sub>2</sub> reactant gas feed which exists between 100 - 200°C and is responsible for nitrogen coupling and dinitrogen formation in the direct oxidation pathway. During the mid-temperature range 240°C <  $T_2$  < 300°C, the catalyst comprises metallic Pd NPs with a surface oxidic layer which permits the formation of  $N_2$  and  $N_2O$ . The formation of bulk PdO NPs is achieved at higher temperatures,  $T_3 > 300^{\circ}$ C, and coincides with the formation of nitrates at the catalyst surface and onset of NO gas as a reaction product. Therefore the selectivity of the catalytic NH<sub>3</sub> oxidation can be attributed to the Pd NP structure, with preferred selectivity to  $N_2$ achieved from a previously unidentified PdN<sub>x</sub> species. Whereas the unfavourable over-oxidation to NO occurred at the surface of bulk PdO NPs.

### Methods

**Sample Preparation.** Supported 1.5wt% Pd catalysts were prepared by incipient wetness impregnation of an acidified aqueous solution of palladium nitrate (15.11 wt% Pd, Johnson Matthey) onto the commercial support;  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> (SASOL), zeolite–Y (Zeolyst, CBV600, H-form, SiO<sub>2</sub>/Al<sub>2</sub>O<sub>3</sub> molar ratio of 5.2). Incipient wetness impregnation was carried out at room temperature. The impregnated supports were subsequently dried at 100°C, then calcined in air at 500°C for 2 hours.

**Ex situ** Characterisation. TEM images of the samples were obtained using a JEOL JEM 2100 transmission electron microscope. Samples were prepared for TEM analysis by dispersing in high-purity ethanol using ultrasonication. 10  $\mu$ L of the sonicated suspension were pipetted onto a holey carbon supporting Cu grid, and the solvent evaporated. XPS analysis was performed with a Kratos AXIS Ultra DLD apparatus, equipped with monochromated Al K $\alpha$  radiation X-ray source, a charge neutralizer and a hemispherical electron energy analyser. During data acquisition, the chamber pressure was kept below 10-9 mbar. The spectra were analysed using the CasaXPS software pack and corrected for charging using C 1s binding energy (BE) as the reference at 284.8 eV.

Operando Pd L3-edge XANES. Operando Pd L3-edge XANES (3175 eV) were collected in fluorescence mode at the XMaS beamline of the ESRF. The catalyst sample (125-250 µm pellet fraction) was contained between two Kapton windows as a packed bed within a 100  $\mu m$  trench of a flow reactor, reported elsewhere.  $^{46}$ The reactor was mounted within a He chamber and connected to gas lines for flow of reactant gases to the sample. The effluent flow from the reactor was sent to a Hiden QGA mass spectrometer. The temperature was controlled by a thermocouple inserted close to the catalyst bed. The energy range (3150 - 3220 eV) was selected using a double crystal Si (III) monochromator and spectra were collected using a fluorescence detector at 90°C to the X-ray beam and with the sample orientated 45° to the incoming beam. Each scan measured 265 points with a counting time of 5 seconds per point, taking a total time of 22 minutes per spectrum. The experimental procedure consisted of online pre-treatment at 200°C in a reducing (5% H<sub>2</sub>/He) atmosphere. The reactant gas mixture was introduced (0.5% NH<sub>3</sub>, 2.5% O<sub>2</sub>, 97% He) at 100°C with a total flow of 40 ml min<sup>-1</sup>, before ramping 2.5°C min<sup>-1</sup> to 200°C. The Pd hydride was formed in situ by cooling the sample to room temperature in 10%  $H_2/He$ . The Pd nitride was formed by introducing 0.5% NH<sub>3</sub>/He to the reduced catalyst at 100°C.

Operando Pd K-edge XAFS/DRIFTS/MS. Operando XAFS and DRIFTS measurements were conducted at SuperXAS X10DA, Swiss Light Source, Paul Scherrer Institute, using a previously reported Harrick DRIFTS cell<sup>47-49</sup> integrated with an Agilent Carey 680 FTIR spectrometer. A DaVinci arm accessory fitted with praying mantis optics was used to refocus the IR beam outside the FTIR spectrometer for positioning of the Harrick DRIFTS cell in the X-ray path. The Harrick DRIFTS cell has an X-ray path length of 3.17 mm through the sample positioned 1.04 mm below the surface of the catalyst bed. The experiment used a controlled flow of mixed gases into the reaction chamber under atmospheric pressure using Brooks mass flow controllers, and the temperature of the catalyst controlled and monitored by a thermocouple positioned inside the catalyst bed. Each run used ~10 mg of pelletized (125-250 μm) catalyst. XAFS measurements were performed at the Pd K-edge (24358 eV) in transmission mode using QEXAFS setup with an oscillating Si(311) double crystal monochromator operating at 1 Hz. All XAFS spectra were acquired with a Pd foil placed between  $I_t$  and  $I_r$ , and the time resolution was 0.5 seconds per spectrum ( $k_{max}$ = 16.2). DRIFTS spectra were collected taking 64 scans with a resolution of 4 cm<sup>-</sup> <sup>1</sup> using the liquid nitrogen cooled MCT detector. The time resolution of DRIFTS were 30 seconds per spectrum (400 – 4000 cm<sup>-1</sup>). At the same time, the composition of effluent gas was measured using a mass spectrometer (Hiden QGA) for  $H_2$  (m/z =2), He (m/z = 4), NH<sub>3</sub> (m/z = 17), H<sub>2</sub>O (m/z = 18), N<sub>2</sub> (m/z = 28), NO (m/z = 30), O<sub>2</sub> (m/z = 32), N<sub>2</sub>O (m/z = 44) and NO<sub>2</sub> (m/z = 46). The experimental procedure consisted of online pre-treatment at 400°C in a reducing (5%  $H_2/He$ ) atmosphere, then cooling to 100°C in He. The reactant gas mixture was introduced (0.5% NH<sub>3</sub>. 2.5% O<sub>2</sub>, 97% He) at 100°C with a total flow of 40 ml min<sup>-1</sup>, before ramping 2.5°C min<sup>-1</sup> to 400°C. Background DRIFTS spectra were recorded in flowing He at 100°C and subtracted from the sample spectrum for each measurement.

DFT + D3 Calculations. All the spin polarized periodic density functional theory based calculations were performed using Vienna Ab Initio Simulation Package (VASP).50-52 The projector augmented wave (PAW) method was used and the cut-off energy for the expansion of the plane-wave basis sets was set to 550 eV, which gave bulk energies converged to within 10-5 eV.<sup>53</sup> A convergence criterion of 0.01 eV Å<sup>-1</sup> was chosen for structural optimizations and a k-point grid of 3×3×1 was employed for all slab calculations. Benchmark calculations were used to check the results obtained with Monkhorst-Pack k-point grids of different grid densities, detailed in Supplementary Note 1, which gave negligible differences in calculated adsorption energies (Supplementary Table 9). A smearing value of 1 was used, which is recommended in the VASP manual to be appropriate for metals,<sup>54</sup> and a partial occupancy value of 0.2 eV. Benchmark calculations were used (detailed in Supplementary Note 2) to check the influence of the partial occupancy value on the calculated adsorption energies (Supplementary Table 10). The Perdew-Burke-Ernzerhof (PBE) version of generalized gradient approximation (GGA) was used to carry out geometry optimizations and the total energy calculations.<sup>55</sup> The ideal Pd(111) surfaces were modelled by a 2×2 cell with seven atomic layers. For calculations involving the adsorption of NH<sub>3</sub>, of the seven atomic layers the bottom four layers were fixed to mimic the bulk of the material. The slabs were cut from bulk Pd with a calculated energy minimized lattice constant of 3.904 Å (which compares well with the experimental value of 3.891 Å) while in the direction perpendicular to the surface, a vacuum gap of ~15 Å was used. To check the increase in the Pd–Pd distances due to the presence of interstitial N-atoms series of calculations were performed by systematically increasing the number of N-atoms at the interstitial positions. For all these calculations, we placed the interstitial N-atoms at the centre of the octahedral sites that are close to both the exposed surfaces and relaxed the atomic coordinates within a fixed cell. This study was extended to clarify the influence of the interstitial N-atoms on the adsorption properties of NH<sub>3</sub> molecule on the Pd(111) surface. The adsorption of the NH<sub>3</sub> molecule was allowed on only one of the two exposed surfaces. The dipole moment, due to the adsorbed species, was taken into account by using the methods implemented in VASP according to the procedures of Markov et al. and Neugebauer et al.56, 57 In this study we have also employed Grimme's dispersion correction (DFT+D3) as dispersive effects might be significant for such systems.<sup>58</sup> The climbing-image nudged elastic band (CI-NEB) method was employed to determine the minimum energy path for evaluating the activation energy barrier for the diffusion of interstitial Natom. 59,60

**Data Availability.** The data that support the findings of this study are available from the University of Southampton repository with the identifier 10.5258/SOTON/D0709

**Competing Interests.** The authors declare no competing interests.

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### Acknowledgements

UCL and EPSRC are acknowledged for the iCASE studentship of E.K.D. The beamline scientists at Swiss Light Source and staff at Paul Scherrer Institut are acknowledged for provision of beamtime (proposal 20160396). The beamline scientists at XMaS, European Synchrotron Radiation Facility are acknowledged for provision of beamtime (experiment 28-01-1213). XMaS is a UK national facility supported by EPSRC, and staff at University of Warwick and University of Liverpool are recognised for their support in facilitating the beamtime. The RCaH are acknowledged for use of facilities and Johnson Matthey for providing catalyst precursor materials and testing facilities. UK Catalysis Hub is thanked for resources and support provided via our membership of the UK Catalysis Hub Consortium (portfolio grants EP/K014706/1, EP/K014668/1, EP/K014854/1, EP/K014714/1 and EP/I019693/1). Via our membership of the UK's HEC Materials Chemistry Consortium, which is funded by EPSRC (EP/L000202), this work used the UК National ARCHER Supercomputing service (http:www.archer.ac.uk). We acknowledge the use of Athena at HPC Midlands+, which was funded by the EPSRC on grant EP/P020232/1, in this research, via the EPSRC RAP call of spring 2018. We also thank HPC Wales for the computing time.

### Author Contributions

The experiments on SuperXAS were performed by PPW, EKD, EKG, RHB, MN and AG. The experiments on XMAS at the ESRF were performed by PPW, EKD, GFT, PT, and SMR. The catalysts were prepared by EKD and AR, with routine characterisation performed by EKD. The XAS data was interpreted by PPW, EKD, AK and EKG. XPS measurements were performed and analysed by EKD, SFRT, and TEE. The computational work was performed by AC and CDZY. The work was conceived and designed by PPW, CRAC, AG, PC, and CH. The manuscript was written by PPW, EKD, AC, and AG.