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1 Extrusion-based additive manufacturing with cement-based materials – Pro- 2 duction steps, processes, and their underlying physics: A review

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16 Abstract

17 This article offers a comprehensive overview of the underlying physics relevant to an under-
18 standing of materials processing during the various production steps in extrusion-based 3D
19 Concrete Printing (3DCP). Understanding the physics governing the processes is an important
20 step toward the purposeful design and optimization of 3DCP systems as well as their efficient
21 and robust process control. For some processes, analytical formulas based on the relevant phys-
22 ics have already enabled reasonable predictions with respect to material flow behavior and
23 buildability, especially in the case of relatively simple geometries.

24 The existing research in the field was systematically compiled by the authors in the framework
25 of the activities of the RILEM Technical Committee 276 “Digital fabrication with cement-
26 based materials”. However, further research is needed to develop reliable tools for the quanti-
27 tative analysis of the entire process chain. To achieve this, experimental efforts for the charac-
28 terization of material properties need to go hand in hand with comprehensive numerical simu-
29 lation.

30
31 Keywords: Concrete technology; additive manufacturing; 3D-printing; extrusion; production;
32 underlying physics

33 1. Introduction

34 Digitalization and automation in construction bring great potential in respect of increases in
35 productivity, in creating more attractive jobs, and in compensating for shortages of skilled labor
36 [1, 2]. Large-scale Additive Manufacturing (AM) with cement-based materials, commonly re-
37 ferred to as 3D Concrete Printing (3DCP), belongs to the most promising new concrete tech-
38 nologies for implementing digital data from the planning phase and ultimately to actual auto-
39 mated production in factories and on construction sites [3, 4]. Not only can 3DCP facilitate and
40 quicken production processes considerably, it can also make it technically and economically
41 feasible to realize topologically optimized, geometrically complex structural elements designed
42 according to the principle *form follows force* and also enable the integration of various func-
43 tionalities. Such topological optimization allows for elegant, material-minimized, and resource-
44 saving structures [2, 5].

45 Over the last decade various 3DCP approaches have been developed, and a number of real-
46 scale pilot projects have been successfully completed; see e.g. [6, 7]. According to the RILEM
47 process classification framework for Digital Fabrication with Concrete (DFC) technologies [8],
48 the existing approaches of AM with cement-based materials can be subdivided into different

49 groups, of which material extrusion, particle-bed binding, and material jetting are three that
50 have been demonstrated at scale. While these groups and individual approaches within the
51 groups differ considerably with respect to their respective material concepts, equipment, and
52 production steps, all of them base themselves on sound interaction between material and ma-
53 chine along the processing chain. Hence, mastering material-machine interactions is a prereq-
54 uisite for efficient and robust processes, and for their optimization and control. Such mastering
55 is only possible if the underlying physics of individual 3DCP processes are well understood and
56 purposefully applied.

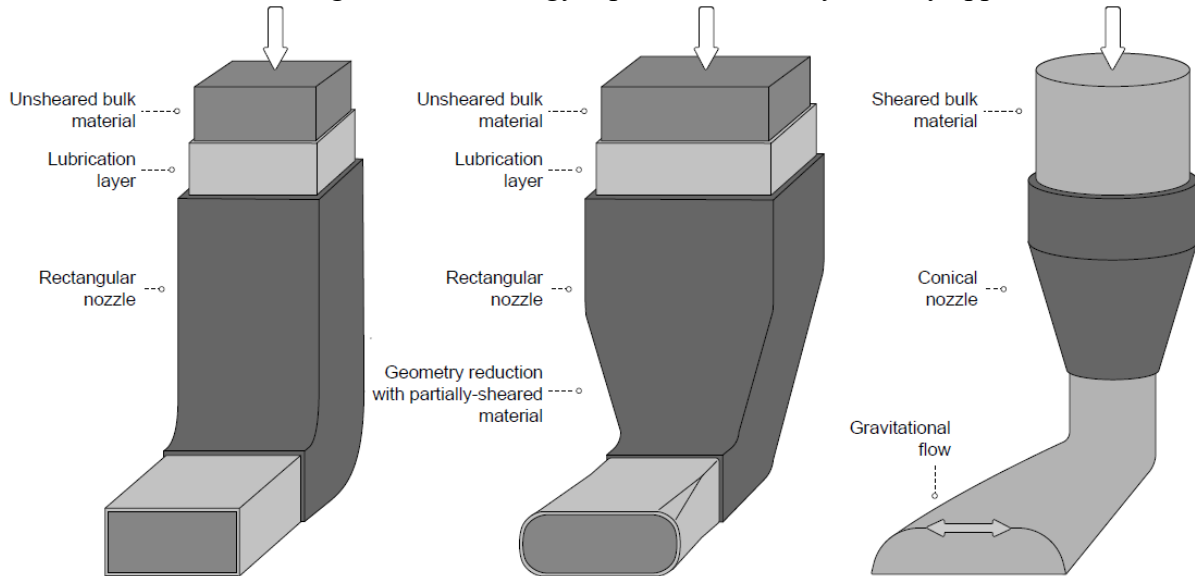
57 In this light and in understanding the importance of the topic with respect to successful imple-
58 mentation of DFC technologies into the practice of construction, the authors dedicated consid-
59 erable time and effort to the systematic analysis of the physical background of various DFC
60 technologies and their individual processing steps. Initiated as an activity of the RILEM Tech-
61 nical Committee 276 “Digital fabrication with cement-based materials”, an important outcome
62 of this work is the article at hand, which focuses on the first group of AM approaches – namely,
63 DFC technologies whose basis is material extrusion. The choice of this DFC group was straight-
64 forward since extrusion-based 3DCP approaches dominate the field both in research and first
65 applications worldwide. Thus, at present material extrusion seems a clear favorite with respect
66 to both the level of overall technological readiness and economic viability [4, 9].

67 The goal of this contribution is to bring together in orderly fashion the relevant knowledge in
68 physics needed to understand and to shape purposefully the relevant processes belonging to
69 extrusion-based 3DCP. In doing so, first, the existing approaches of this DFC group are subdiv-
70 ided into three categories according to the respective major concepts of material handling.
71 Second, the main production steps common to all extrusion-based approaches are defined. Next,
72 the processes and underlying physics relevant to individual production steps are specified for
73 each approach category. This is all laid out in Section 2 while the subsequent sections focus on
74 the main processes, i.e., their physics, their corresponding key physical properties, and their
75 evolution over time. Given the wide spectrum of the relevant processes and their complexity,
76 the authors have aimed at providing a comprehensive overview of crucial aspects as a basis for
77 a general understanding of the technology in the first place rather than trying to deliver the
78 details on every process and its underlying physics. However, numerous references to important
79 sources in the literature should help in finding additional information easily. To make the pro-
80 cess and physical parameters “tangible” while providing some rough guidance with respect to
81 relevant quantitative information, the ranges of absolute values are given as first estimates for
82 the reader’s consideration whenever it is possible and meaningful. Finally, attention is paid to
83 research needs, of which a great many remain.

84 **2. Processing steps and their underlying physics**

85 The additive manufacturing approaches of material deposition by extrusion can be subdivided
86 into three categories: i) extrusion of stiff material, similar to conventional extrusion, ii) extru-
87 sion of flowable material with or without adding admixture(s) in the printhead, and iii) extrusion
88 of material using additional energy input, e.g. vibration, which facilitates the delivery and dep-
89 osition of stiff mixtures. The ideal case of the first category is the “infinite brick” extrusion
90 regime, where the filament and nozzle cross-sections are equal; see Figure 1a. In the second
91 category the ideal case is free flow deposition, where the material flows freely until the stress
92 induced by gravity equals the yield stress of the printable material; see [10] and Figure 1c. In
93 the context of additive manufacturing with concrete, most extrusion flows are located some-
94 where between these two asymptotic deposition approaches; see Figure 1b. Note that while two
95 asymptotic cases can be mathematically modeled as described in the next sections, the predic-
96 tion of more realistic final shapes for real printable, cement-based materials is not straightfor-

ward due to several mechanisms and boundaries involved. These mechanisms/boundaries include solicitation gradient, shear-induced particle migration, non-uniform structural build-up, complex nozzle geometry, and so on. In such cases adequate multi-physical, numerical tools would be required to describe the extrusion/deposition process fully. Analytical formulas can, however, still provide reasonable estimations. Finally, the third manufacturing approach, the extrusion of material using additional energy input, has been as yet rarely applied.



103
104
105 Figure 1. Two asymptotic regimes of material deposition by extrusion: a) extrusion of a very
106 stiff material (so-called infinite brick strategy) and c) extrusion of very flowable material that
107 spreads out after deposition (adapted from [10]); regime b) shows a more realistic case of ex-
108 trusion of sufficiently stiff material by means of a nozzle with a geometric reduction.
109

110 In general the major processing steps in these manufacturing approaches are similar since they
111 all include: 1) transportation of build material to the printhead, 2) printhead process/extrusion
112 by the printhead, 3) deposition of build material, accompanied by its deformation, and 4) de-
113 positions of further layers, accompanied by loading earlier deposited upper layer(s) by self-
114 weight and process-induced forces, related deformation of build material after deposition fol-
115 lowed by further deformation due to early-age shrinkage, early-age creep, and thermal dilation.

116 Note that a technical discussion of the initial processing step, namely “concrete mixing” is be-
117 yond the scope of this publication. Thus, the first step in each presented approach is seen as
118 “transportation”. However, mixing in the printhead to disperse chemical admixtures such as
119 hydration-accelerating agents is addressed here as a part of the printhead process. Table 1 lists
120 three manufacturing approaches of selective material deposition by extrusion and four pro-
121 cessing steps. The underlying physical processes for each processing step for a given fabrication
122 approach are listed in Table 1 as well. These underlying physical processes, such as gravita-
123 tional flow, pumping, i.e., pressure-induced flow in a pipe, or extrusion or pressure-induced
124 flow through a section contraction, are presented and discussed in the following sections. How-
125 ever, early-age shrinkage, early-age creep and thermal dilation of the material after deposition
126 are not covered in this article since they 1) do not represent a production step as such, b) are
127 not considered to be different to conventional concrete construction from the physical point of
128 view.

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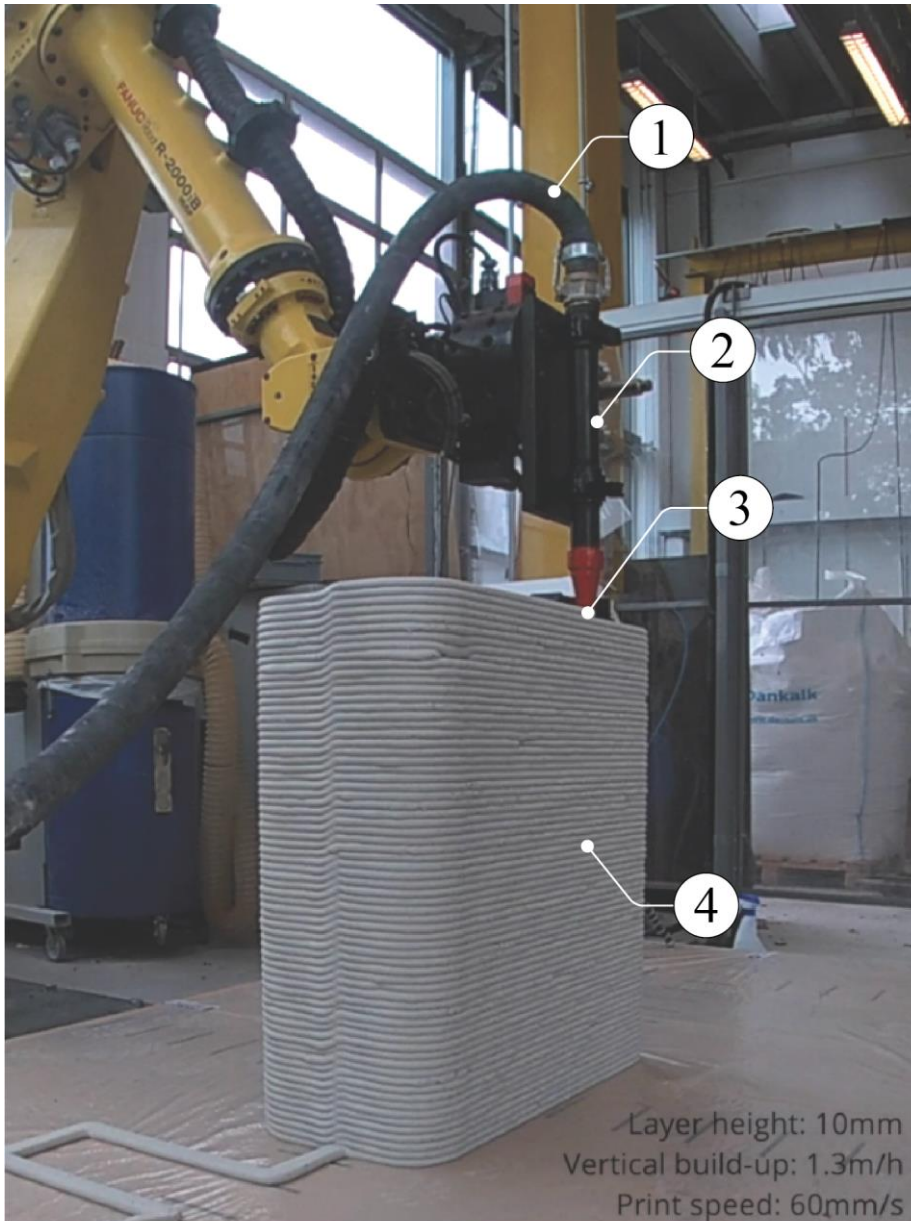
131 Table 1. Production steps, related processes and underlying physics relevant to three different
 132 approaches within the extrusion-based additive manufacturing with cement-based materials.
 133

Production step	Extrusion-based manufacturing approach		
	<i>Extrusion of stiff material</i>	<i>Extrusion of flowable material with or without dispersion of admixture(s) in the printhead</i>	<i>Extrusion of material using additional energy input</i>
<i>Transportation of build material (also primary motivation)</i>	Pumping (over short distances) or gravitational flow with or without energy input	Pumping	Pumping or gravitational flow with or without energy input
<i>Printhead process</i>	Extrusion using primary motivation (pumping), ram extrusion, or screw extrusion	Extrusion using primary motivation, ram extrusion, or screw extrusion; optional dispersion of admixture by high-energy mixing	Screw extrusion or gravitational flow supported by vibration
<i>Deformation of build material during deposition</i>	Gravitational flow, viscoelastic-plastic deformations		Gravitational flow and compaction supported by vibration, elastic deformations
<i>Behavior of build material after deposition</i>	Deformations due to self-weight and kinetic energy of deposition; additionally: early-age shrinkage, early-age creep, thermal dilation		

134
 135 For the sake of clarity one of the approaches is depicted in Figure 2, which illustrates the pro-
 136 cessing steps belonging to the extrusion of flowable materials without addition of admixture in
 137 the printhead.
 138

139 Generally, the behavior of fresh cementitious materials can be considered as visco-plastic. This
 140 means that, similar to solids, they do not flow until a given critical shear stress, i.e. yield stress
 141 τ_c , is exceeded and, similar to a liquid, they flow when subjected to shear stresses greater than
 142 τ_c [11, 12]. Depending on the processing step during additive fabrication, either solid or fluid
 143 behavior is involved, e.g. during the conveying and feeding processes the material flows inside
 144 the transportation system and printhead, i.e. fluid behavior, while after deposition the material
 145 has to remain static (i.e. solid behavior, viscoelastic-plastic) [10].

146 Note that cement-based materials are heterogeneous, and this can have a pronounced effect on
 147 their flow behavior in specific cases. For example, in the case of pumping, shear-induced par-
 148 ticle migration creates a lubrication layer that greatly influences the flow of the material [13].
 149 As a result, neither the velocity profile nor the flow rate can be accurately predicted based on
 150 viscosity and yield stress of the bulk material. The formation of a lubrication layer may occur
 151 also during the extrusion process; see Figure 1a.
 152



1. Transportation of build material – pumping
2. Printhead process – extrusion using primary motivation
3. Deformation of the build material during deposition – gravitational flow, viscoelastic-plastic deformations
4. Behavior of build material after deposition – deformations due to self-weight and kinetic energy from deposition; early-age shrinkage, early-age creep, thermal dilation

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154
155

Figure 2. Processing steps in the extrusion of flowable materials.

156 3. Gravitational flow

157 During the 3D concrete printing process gravity plays a role in the transport of the material,
158 particularly between mixing and placement in a pumping or extrusion system. The gravity-
159 induced flow of material from a hopper to the pump is one of the printing steps that can be
160 studied by analogy to the discharge flows of the March cone, for example.

161 After deposition, there is “competition” between, the force of gravity and the material strength,
162 in rheological terms yield stress. This competition can be described by a dimensionless number,
163 i.e. the ratio of the gravity-induced stress ρgh over the material yield stress τ_c , thus $\rho gh / \tau_c$,

164 where ρ is the material density, g is the acceleration of gravity, and h is the height of the depos-
165 ited layer. Note that the balance between gravitationally induced stresses and the yield stress
166 can be studied using the slump/spread flow theory developed by Roussel and Coussot [14]; see
167 also Perrot *et al.* [15].

168 In the literature related to extrusion-based 3D printing process using cementitious materials, a
169 wide range of material yield stress has been reported. Depending on the dimensionless number
170 $\rho gh / \tau_c$ and the geometry of the nozzle's cross section, the final section of the material layer
171 can vary from rectangular to semi-ellipsoidal, cf. [10]. Figure 1a shows a high yield stress of
172 over 500 Pa for a cement-based material that keeps its shape after being deposited pursuant to
173 the infinite brick strategy, while Figure 1c shows a filament of fluid cement-based material that
174 spreads out after deposition with a yield stress of less than 100 Pa. It is important to note here
175 that for the case of printable cementitious materials with yield stress in the order of magnitude
176 between hundred and thousands of Pa at the nozzle exit, the effect of the surface tension on the
177 final shape of the layer can be neglected [10].

178 In the case where the yield stress of the material is greater than the stress induced by gravity,
179 the layer of cementitious material will not be deformed after deposition. This scenario applies
180 if the material is placed gently and relatively slowly; otherwise additional effects need to be
181 considered. As proposed by Roussel and Coussot [14], this ideal case can be described using a
182 purely elongational plastic model wherein the critical height of a stable layer reads $h_{stable} =$
183 $\sqrt{3}\tau_c / \rho g$ where $\sqrt{3}\tau_c$ is the elongational yield stress of the cementitious materials conditioned
184 on their following von Mises plasticity criterion [10, 14, 15, 16]. In other words, to enable
185 cementitious materials to keep their shape after deposition, the target yield stress value should
186 be at least around 13 Pa per millimeter of layer thickness (for ρ of 2200 kg/m³). Note that the
187 plasticity criterion and its time dependency are still an open research question, especially for
188 stiff mixtures, which can exhibit pressure-dependent behavior [17, 18].

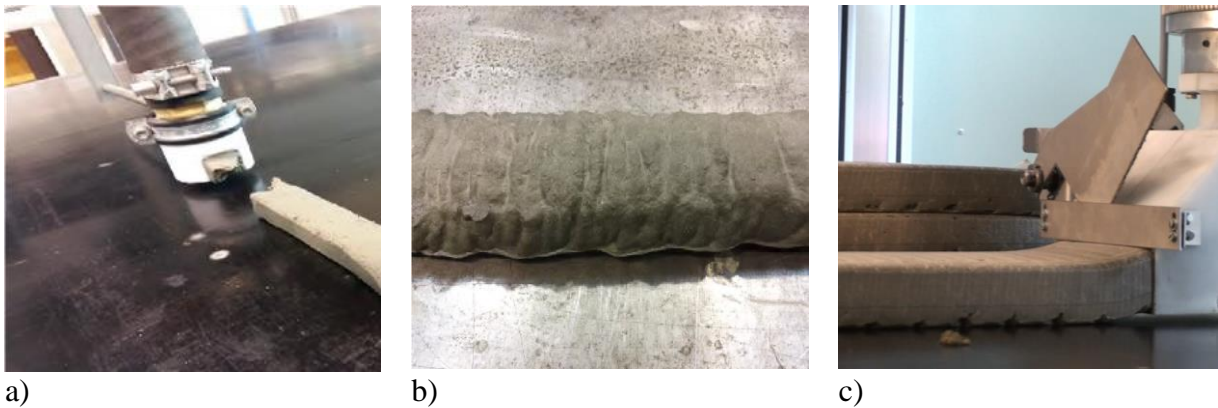
189 If gravitational force overcomes the material elongational strength at the layer bottom, the ma-
190 terial will deform until an equilibrium between yield stress and gravity-induced stress is
191 reached. As a result, the layer being deposited will deform and spread on the previously printed
192 layer. The limiting case of this type of behavior for a low yield-stress value tends to the case of
193 shear flow on a horizontal plate. In this shear flow regime the final height of the layer should
194 be equal to h in ρgh [14]. In many 3D printing applications the actual filament deformation
195 behavior ranges somewhere between the pure shear flow regime and the infinite brick, no-flow
196 regime; thus, the final shape of the layer will depend on its material yield stress, on the cross-
197 section of the nozzle opening, and on the gap between nozzle and printed layers, and this should
198 be studied on a case-by-case basis. In this context, numerical analysis using e.g. Computation
199 Fluid Dynamics (CFD) approaches can be instrumental in understanding material behavior bet-
200 ter under specific boundary conditions and, prospectively, in predicting this behavior.

201 The effect of the gap between nozzle and deposition area on the shape of the deposited layer
202 can also be important because tensile forces, i.e. in under-extrusion, where the material flow
203 rate is lower than $S_{nozzle} \cdot V$, where S_{nozzle} is the nozzle section and V the nozzle velocity, or com-
204 pression forces in over-extrusion, where flow rate is higher than $S_{nozzle} \cdot V$, can be induced by the
205 material leaving the nozzle; see Figure 3. These effects could lead to surface cracking, layer
206 bending, or even coiling. These effects should be studied in some depth in order to allow for
207 full control of layer geometry during printing.

208 Some extrusion-based, selective material deposition methods [19] purposefully establish a di-
209 rect interaction between the nozzle and the extruded material: The nozzle presses the exiting
210 material, forcing the layer to have a thickness equal to the gap between the nozzle and the layer

211 below; it is assumed that the layer below is stiff enough to sustain this additional pressure with-
 212 out considerable deformation. This strategy has the advantage of eliminating uncertainty in the
 213 final height of the printed structures since the final printed structure height will only depend on
 214 the nozzle toolpath. While improving the overall geometry control of the printing process, the
 215 direct interaction of the nozzle and the deposited material induces additional pressure on the
 216 underlying layers at the level of the nozzle and can contribute to cracking or even to collapse
 217 of the printed element. Note that to date the tensile behavior of fresh, cement-based materials
 218 has not been extensively studied. One can mention the tensile strength measurement carried out
 219 by Mettler *et al.* [18] and Lo Monte *et al.* [20]. Nonetheless, a consistent explanation of the
 220 cracks' formation and the definition of mix-design solutions to prevent it, e.g. by adding fiber
 221 or polymer, are still open questions requiring extensive research.

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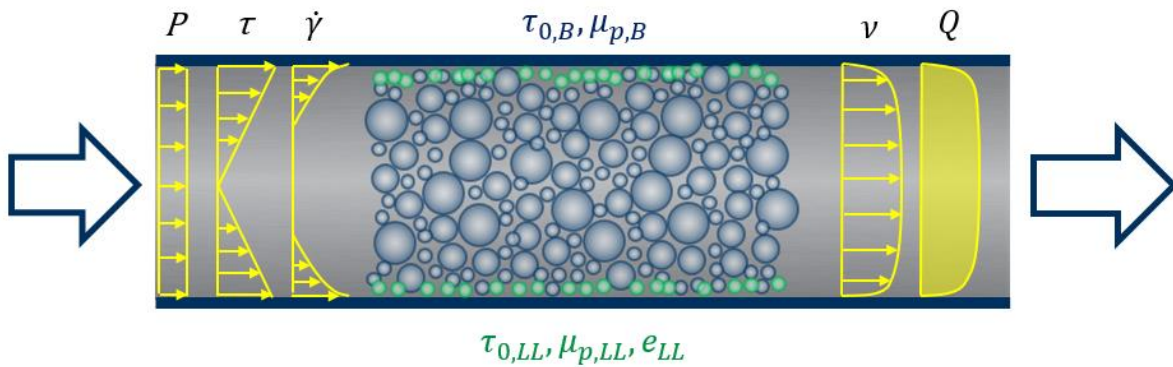


223 Figure 3. Challenges in extruding concrete or mortar: a) extrusion rate is too low compared to
 224 the nozzle velocity leading to discontinuous layers [21], b) extrusion rate is too fast compared
 225 to the nozzle velocity leading to “buckled” layers [21], c) “tearing” of extruded layers due to
 226 the overly high yield stress of the printed material; adapted from [22].

227 4. Pumping and extrusion

228 4.1 Pumping

229 Pumping is often used in transporting mixed concrete to the printhead or nozzle. It is performed
 230 by pushing material through a hose with the help of positive-displacement piston pumps or
 231 progressive cavity pumps, the latter is so far the most widely implemented means in extrusion-
 232 based additive fabrication. While the pumping process as such is similar to ram extrusion, there
 233 are some distinctions worth addressing: it usually covers considerably longer distances and re-
 234 sults in higher material discharge rates. Furthermore, ram extrusion usually implies compaction
 235 and shaping of the material by the nozzle narrowing towards its opening. Concrete pumpability
 236 as characterized by pressure losses depends primarily on the plastic viscosity. However, the
 237 influence of yield stress increases with increasing τ_0/μ [23]. Calculations using the Bucking-
 238 ham-Reiner equation for laminar flow of a Bingham fluid through a pipe overestimate the
 239 pumping pressure for a given flow rate by up to 5 times [13, 24, 25]. This is a consequence of
 240 the formation of a lubricating layer (LL) at the inner walls of the pipeline due to shear-induced
 241 particle migration (SIPM). The yield stress and plastic viscosity of the LL are generally about
 242 one fifth and one fifteenth of the corresponding values of the bulk material, respectively [24].
 243 The shear stress, shear rate, and velocity profiles of concrete during pumping are presented in
 244 Figure 4.



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Figure 4. Pressure P , shear stress τ , shear rate $\dot{\gamma}$, velocity v and flowrate Q profiles of a concrete flow inside a pumping pipeline. The yield stress and plastic viscosity of concrete are $\tau_{0,B}$ and $\mu_{p,B}$, whereas the yield stress, plastic viscosity and thickness of the lubricating layer are $\tau_{0,LL}$, $\mu_{p,LL}$ and e_{LL} , adapted from [26].

Shear rate and shear stress are zero at the central axis of the pipe and are maximal at the pipe wall. If the shear stresses at a given radial position are higher than the concrete bulk yield stress, then shearing of the bulk takes place. As a consequence of and in dependence on the properties of concrete and pumping conditions, concrete can flow in the slip flow mode, i.e. very thin slip layer + unsheared plug, or in the slip-plus-shear flow mode, slip + partial shear of bulk + plug [13, 27]. Kaplan *et al.* [27] introduced analytical models for predicting concrete pumping pressure in the cases of slip and slip-plus-shear flow modes using rheological parameters for bulk $\tau_{0,B}$, $\mu_{p,B}$ and for interface $\tau_{0,LL}$, $\mu_{p,LL}$ as well as geometrical data of the pipeline. In the case of infinite brick extrusion with stiff material, the dynamic yield stresses and plastic viscosities are a few hundred Pa, and few Pa·s, respectively.

Whereas in the case of free flow deposition, these values are few tens of Pa and few tens of Pa·s. Although the free flow material has lower yield stress, this does not necessarily mean lower pumping pressure losses in comparison to infinite brick extrusion, in which the high yield stress bulk does not deform during pumping, as proven in earlier research on pumping [23, 24]. Assuming a flow rate of 5 m³/h in a DN125 pipeline, the pressure loss per unit length can be in the range of 3.5 kPa/m for the infinite brick case and 5 kPa/m in the case of free flow deposition. This translates into pumping pressures of 70 kPa and 100 kPa for a DN125 pipeline of length 20 m, respectively. Similarly, higher pressure losses are expected in the case of free flow deposition printable concrete than infinite brick printable concrete when pumped in pipelines of smaller diameter e.g. DN50.

The optimal rheological requirements of the material with respect to pumping are in competition with those favorable in respect of deformation behavior after layer deposition. In general, printable concretes are supposed to have high static yield stress to overcome gravitational and other forces acting after material deposition. The influence of high structural buildup and thixotropy rates of 3D-printable concretes, especially those used in infinite brick extrusion on pumping behavior have yet to be investigated. De Schutter and Feys [13] emphasized that “short interruptions during pumping lead to major difficulties in resuming pumping operations due to the sometimes tremendous effect of internal structural buildup”. With large-scale pumping tests, they have observed that a 20-minute delay leads to an increase of pressure loss by 50% for a mixture with a structural buildup rate A_{thix} of 0.3 Pa/s. Despite increasing the pumping pressures to 35000 kPa, the pumping operations could not be resumed after the delay. Therefore, pumping very high-yield-stress concretes or, pumping chemically accelerated concrete may not be an optimal solution for large-scale, extrusion-based additive fabrication.

285 An alternative solution is to pump low viscous and low yield stress concretes over the long
286 distance and then activate them at the printhead, shortly before deposition. Such an approach
287 requires the use of inline, second stage mixing techniques to ensure precise dosage and uniform
288 homogenization of the admixture; see Section 5. Note that even in the case of free flow depo-
289 sition, it is necessary to ensure that the rheological properties are sufficiently high to prevent
290 filtration under high-pressure and segregation.

291 The methodology in characterizing the pumping-relevant rheological properties of concrete and
292 lubricating layer, and the prediction of pumping pressures is a considerable challenge. The in-
293 adequacy of empirical design charts, which consisted mostly of a slump or spread parameter
294 [13], has been overcome with recent, advanced test approaches and prediction models [28, 29,
295 30, 31, 32]. However, most of these approaches require the accurate quantification of the con-
296 crete's and the LL's rheological properties, which has proven challenging for high yield-stress,
297 printable concretes [33].

298 Numerical models can help in understanding the SIPM during pumping and characteristics of
299 LL, as well as predicting pumping pressures. However, such tools have so far been simplified
300 by the assumption of either a single-fluid homogeneous medium [23, 24, 25, 28, 34] or discrete
301 granular elements [35]. In the case of concretes with high granular fractions, the interactions
302 between aggregate and paste are crucial. Fluid-solid numerical models coupled with the mod-
303 elling of time- and shear-dependent variations of pumping characteristics are yet to be devel-
304 oped.

305 The pumping process can alter the rheology of concrete, for example, due to dispersion of ce-
306 ment particles at higher shear rates, activation of residual superplasticizer, or air-induced by
307 pistons. However, earlier observations of pumping-induced changes in concrete yield stress are
308 inconclusive [28, 36]. This can cause critical consequences for the continuity of the extrusion
309 process or the stability of the resulting extruded layers, respectively, as a consequence of the
310 increase or decrease in yield stress, thus necessitating further research to develop inline rheol-
311 ogy monitoring tools that can be embedded, for instance, in the printhead.

312 313 **4.2 Extrusion**

314 In the context of additive fabrication, the extrusion step can be considered as the action of con-
315 ferring the shape of its section to the deposited layer of cementitious material. This step is per-
316 formed by forcing the material through a section contraction. This complex flow system has
317 been studied extensively, providing insights into the rheology of visco-plastic fluids [36, 37,
318 38, 39, 40]. However, when dealing with cement-based materials, especially with a view to the
319 additional plastic materials used in the "infinite brick" extrusion approach [10], i.e. to material
320 with yield stresses higher than several kPa, extrusion-induced segregation has been reported in
321 the literature due to the multiphase nature of the cementitious materials under study [33, 41, 42,
322 43]. Such behavior can be described as competition between a) the cementitious materials' characteristic drainage time that is the result of the pressure drop at the extruder die and b) the extrusion time [33]. If the drainage time is shorter than the extrusion time, segregation is likely to occur, leading to the material's stiffening. For stiff cement-based mixtures which are close to the solid randomly packed fraction, drainage can also drastically change the material's rheological behavior from plastic to frictional [29, 33, 41, 43]. Such types of behavior are likely to induce surface defects or, more damaging, a stoppage of flow. Hence, securing a sufficient extrusion flow rate helps reduce the effect of drainage.

330 Another solution is to add external energy in such a way as to ease the flow, reducing wall
331 friction and locally liquefying the cementitious materials. Such energy-based approaches to
332 extrusion are a recent field of interest that has not been widely studied yet. Among these meth-
333 ods is local vibration of the cementitious material, which was studied to reduce the apparent

334 yield stress of the material and to allow for a reduction of the extrusion pressure due to contrac-
335 tion flow [30].

336 The effect of vibration on an extrudate surface produced with a screw extruder can be seen in
337 Figure 5, which indicates cracks which appear on the extrudate surface when no vibration is
338 applied. When adding adapted vibration, the extrudate surface becomes smooth, and for even
339 higher values of vibration energy, the extrudates are no longer able to withstand gravitational
340 force.

341 The strategy of using vibration not only for facilitating material transport in the large printhead
342 but also for extrusion-based material deposition was used by the HuShang Tengda company
343 [31]. Because of the vibration of the nozzle orifice, relatively stiff, “ordinary” concrete could
344 be deposited layer-wise and compacted in the process according to the personal observation of
345 this article’s first author. Note that for the purposeful use of such an approach a precise deter-
346 mination of the area of influence of vibration is required since the action of vibration on the
347 already deposited layer could compromise the structural buildup process and lead to collapse
348 of the printed layers. Some studies on the effects and influence zones of vibration have been
349 carried out, but they are limited to the axisymmetric case of a vibration needle immersed in
350 concrete [32, 44]. Further studies on the influence of vibration on cementitious materials’ ex-
351 trusion flow are therefore necessary, specifically in the context of additive fabrication.

352 For printable cement-based materials, which are usually highly thixotropic, understanding the
353 effect of vibration of extruded layers is a complex task, the zone of influence might be smaller
354 than is usually predicted [44]. Alternatively to vibration, it is possible to use the electric poten-
355 tial difference within the steel extruder to promote the formation of a lubricating layer at the
356 extruder’s inner walls [45, 46]. Local application of a magnetic field is also envisaged to change
357 the rheology of cementitious materials containing magneto-sensitive particles that can thus be
358 uniformly oriented [2, 47]. While such approaches seem promising, substantial additional ef-
359 forts are required to assess the effect of those external energy applications on the extrusion flow
360 of cementitious materials in the context of additive fabrication. In the first place, the determi-
361 nation of the precise region of influence of external solicitation is mandatory, since such influ-
362 ence could compromise the stability of the already printed layers.

363



364

a) No vibration

365

b) Frequency 7Hz
Amplitude 12 mm

366

c) Frequency 30 Hz
Amplitude 12 mm

367

368 Figure 5. Effect of external vibration on the surface of extruded mortar (same screw rotational
369 velocity of 5 rpm; mix design can be found in [30].

370

371 Furthermore, the design of the die is crucial because it governs the orientation and shape of the
372 layer of material being deposited [6, 48]. Likewise, the velocity profile is important to control
373 within the cross-section of the extruder in order to ensure proper placement of the layers and
374 good adherence between layers [6]. Additionally, the extrusion flowrate should be controlled
375 and estimated based on the rheological behavior of the cementitious materials, since the veloc-
376 ity of the material leaving the extruder has to be adjusted to the robotic arm to control the section

377 of the deposited layer; see also Figure 3. Finally, some shapes of the nozzle make it necessary
378 to rotate the nozzle when printing curves.

379 Moreover, in the case of accelerated mixes, mostly using chemical admixtures, it is important
380 to use die geometries which prevent the formation of a dead zone, since the material “captured”
381 there can harden during the process and eventually lead to process stoppage due to blockage.
382 Consequently, mastering the flow behavior within the extruder leads to better control of the
383 material’s residence time in the die and avoid the buildup of heterogeneities, thus facilitating
384 an adequate flow rate and trouble-free material deposition.

385 While the velocity profiles within an extruder have been studied for the simplest case of ax-
386 isymmetric ram extrusion [38, 36, 41, 49], no study has dealt with the determination of the
387 strain and stress fields within the extruder for complex die or nozzle geometries. Hence, there
388 is need to provide numerical tools which can predict accurately the flow of cementitious mate-
389 rials through nozzles of any given geometry.

390 A last open question deals with the study of the flow of cementitious materials in a hopper
391 screw geometry or in a progressive cavity pump extruder [21]. Predicting the cementitious flow
392 rate from its rheological and tribological behavior and from the screw’s rotational velocity is a
393 challenge. Adequate numerical and analytical tools appropriate to the rheological properties of
394 a given cement-based material are much required to ensure the continuous and controlled flow
395 rate of the materials during extrusion.

396

397 **5. Dispersion in the printhead**

398 Many printing techniques involve a so-called acceleration of the material during or after depo-
399 sition. Most of these techniques rely on the incorporation at the printhead level of either a chem-
400 ical accelerator able to modify the silicate/aluminate balance and accelerate one of these hydra-
401 tion reactions, or an organic flocculant able to bridge the finest particles in the system. They
402 could also rely on the mixing of the material with an alternative binder such as aluminate-based
403 substances [50]. The former strategy (dispersion) involves amounts of products below a very
404 few percent while the latter (mixing) often involves amounts above 10% by volume. All the
405 strategies above lead to a faster and enhanced phase-change of the printed material, allowing
406 for higher building rates and higher productivity. Moreover, these strategies allow for the mix-
407 ing, pumping, and feeding of the robot with an extremely fluid material, which then turns into
408 a pseudo-solid once the accelerator is added. However, these strategies require the dispersion
409 of an active agent in the printhead or the mixing of a slurry, both of which give rise to several
410 difficulties in terms of printhead design. The authors have focused here on dispersion.

411 From a process point of view, as soon as a dispersion technology is involved as a sub-process,
412 residence (or retention) time of the material in the dispersing zone becomes a key parameter
413 [51]. Considering that the nozzle cross-section is often close to the cross-section of the layer,
414 the average material velocity in the printhead is on the order of the nozzle velocity itself (typi-
415 cally from 1 to 10 cm/s [9]). Since most printheads’ overall lengths are in the order of several
416 tens of centimeters, the residence time should vary between 1 and 100 s.

417 The accelerators used in additive manufacturing are either inorganic compounds with sizes of
418 less than one nanometer or organic macromolecules of sizes on the order of 100 nanometers
419 [52]. Although the validity of the Stokes-Einstein equation (see Eq. 1) fades when the size of
420 the molecules gets closer to the size of the molecules of the solvent, it is used here to estimate
421 the typical diffusion length from the natural diffusion coefficient of these accelerators.

$$D = \frac{kT}{6\pi r\mu} \quad (1)$$

422 where k is the Boltzmann constant, T the temperature in Kelvin, r the size of the accelerator
 423 molecule and μ is the viscosity of the solvent. The typical diffusion length is of the order of
 424 \sqrt{Dt} , where t is the residence time. This leads to typical diffusion lengths between a few mi-
 425 crometers and a few hundred micrometers. Consequently, even for the smallest printheads for
 426 pastes, full dispersion of accelerators in the nozzle cross-section cannot rely on natural diffusion
 427 alone, requiring additional dispersion capacity. As the viscosity of cement-based materials is
 428 too high to allow for turbulent dispersion, it is in the field of convective mixing that solutions
 429 do exist [53, 54].

430 The idea behind convective mixing relies on the creation of a secondary flow in the nozzle so
 431 that the distribution of the accelerator molecules is allowed. By shearing the material and,
 432 hence, distributing the accelerator in sheared material layers, one can reach, after a sufficient
 433 residence time, the situation in which the typical distance between two sheared material layers
 434 is on the order of the typical diffusion length estimated above.

435 From a technological point of view with respect to the newest existing solutions, this translates
 436 into either so-called static mixers or screw-mixing devices. The former are immovable, as in-
 437 dicated by their description, and it is the overall flow of the material around the surface of the
 438 static mixer that disperses the accelerator or other admixture, such as pigments. The latter are
 439 simply additional mixing systems with their own controls. Both are inserted into the printhead.
 440 The dispersion intensity of the static mixer is proportional to the flow rate in the printhead,
 441 while at the same time it is an independently tunable parameter in the case of the screw mixer.

442 The above features have not been studied in detail for the specific conditions and requirements
 443 of extrusion-based printing with cement-based materials as yet. Printheads and their mixing
 444 devices are accordingly designed by trial and error. Numerical simulations [55, 56, 57] could
 445 in the future allow for a better understanding and progress in printhead design.

446 **6. Load-bearing and deformation behavior after deposition**

447 **6.1 Object failure during manufacturing**

448

449 **6.1.1 Failure modes**

450 Additive manufacturing methods of cementitious materials are, by definition, set apart from
 451 “conventional” formative methods by the absence of molds. As a result, objects may collapse
 452 during the manufacturing process. Considering that the material resistance is initially low rela-
 453 tive to its self-weight, the ‘buildability’ [58, 59] of a mortar is an important property in assessing
 454 its suitability for printing. The term, however, refers to a range of processes and properties that
 455 require elaboration in order that it becomes a meaningful concept.

456 At present, two mechanisms have been recognized as causes of collapse in extrusion-based,
 457 layered 3D concrete printing during manufacturing:

- 458 • material failure (Figure 6a), and
- 459 • loss of stability (Figure 6b).

460 Material failure occurs when the material strength is exceeded, resulting in yielding, flow, or
 461 fracture, whereas stability loss is defined by a loss of equilibrium of forces and moments, initi-
 462 ating uncontrolled deformations or displacements.

463



464
465
466
467
468
469

a) b)
Figure 6. Collapse in extrusion-based additive manufacturing: a) material failure, b) stability failure [60].

470 Either mechanism is triggered by a combination of support, load, and geometry conditions – of
471 which the former is generally constant, but the latter do change over time due to the gradual
472 buildup of the object during manufacturing. In the hitherto limited number of developed pre-
473 diction models for object failure during printing of cementitious materials, the centered self-
474 weight of the material is considered as the sole load condition exerted on the print object. This
475 allows the formulation of relatively simple relations between material strength and build height;
476 see Section 6.1.2, Eqs. 2 and 3. Several additional types of loading are conceivable, such as
477 kinematic pressure from the deposition of filament, vertically or horizontally, when layers are
478 squeezed together, non-vertical pressures from supporting filler materials as in Figure 7, eccen-
479 tric loading due to accidental misalignment of layers, or purposefully created cantilevering ge-
480 ometries.
481



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483
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487
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Figure 7. Structural failure of an object during 3D printing due to horizontal loading by a highly fluid infill material

In any specific case, ‘buildability’ – or the resistance to failure during printing – is not only dependent on the material’s characteristics but, also on the object design, e.g. size, geometry, and process parameters such as print speed. For example, at otherwise identical print settings,

489 a straight wall may fail sooner than a curved wall of the same path length due to the increased
490 stability provided by the curved wall's bend(s). In another case, a short wall could topple over
491 at fewer layers than a long wall, even though buckling is determined by the perpendicular sec-
492 tion, which is identical for both, because of the slower buildup and longer time for the material
493 to develop strength and rigidity [61].

494 To define the initiation of material failure, i.e., fracture, yielding, etc., many criteria have been
495 proposed [62], but it is not immediately clear which one(s) is (are) suitable for the mortars that
496 are used in extrusion-based concrete printing. This is primarily due to the transition that print-
497 able cementitious mortars undergo from a non-Newtonian fluid state to a solid state during the
498 manufacturing process. This process, which usually takes several hours but can be as short as
499 several minutes when fast-setting cements or accelerators are applied, includes reversible phys-
500 ical, e.g., flocculation-induced thixotropy, as well as irreversible chemical phenomena (hard-
501 ening due to cementing reactions). As a result, both strength and rigidity increase over time,
502 which in rheology is often labelled as the structuration rate of the yield stress, A_{thix} .

503 The selection of a suitable criterion to characterize material failure is further complicated by
504 the fact of printable mortars having considerably varying levels of initial yield stress and rigid-
505 ity, associated sheared or un-sheared flow regimes, and different structuration and hardening
506 rates as well while the level of material development in an individual print depends on the print
507 strategy and object design. Thus, the appropriate material failure criterion may depend on the
508 mortar, process and design characteristics in any specific case – further increasing the chal-
509 lenges in developing generic prediction models.

510 Considering the fluid-to-solid transition, the material could be assumed to behave as a highly
511 viscous fluid or as an initially very compliant visco-elastic or elasto-plastic solid. In the case of
512 the former, concepts from the field of fluid mechanics and rheology should be applied, whereas
513 solid mechanics should be applied in the case of the latter. The approach selected governs the
514 definition of material properties and the experimental methods required to determine them. Vice
515 versa, it should be recognized that not all experimental methods are suitable for all ages of print
516 mortar; for example, some mortars are too stiff to perform rotational rheometry without plug
517 forming, while others may be too fluid to perform unconfined compression tests. Indeed this
518 could serve as an indicator regarding the applicability of certain theoretical approaches.

519 The approach that is to be taken also determines the types of analysis that may be performed to
520 quantify 'buildability'. A solid approach governed by mechanics allows recognition of material
521 failure under anisotropic, multi-axial stress states as well as failure due to loss of stability. In
522 fluid mechanics, only isotropic stress states are considered, and stability analyses are highly
523 unusual. However, it is more appropriate for flow analyses.

524

525 **6.1.2 Approaches in literature**

526 In the literature, a range of different approaches has been suggested to quantify buildability,
527 with the lack of unanimity being caused by the sheer number of aspects discussed above. The
528 summary provided in Table 2 shows that the spectrum of models not only stems from the fun-
529 damental approach to the material state, but also from other associated issues related to exper-
530 imental methods, validation, time dependency, and so on. In the elaboration provided below,
531 some quantitative data as well are presented for comparative purposes. However, it should be
532 noted that these values have been obtained from significantly different experimental methods,
533 the validity of which may still be under debate.

Table 2. Approaches to the quantification of ‘buildability’ in literature.

Publication	Failure mode(s)	Theory	Material failure criterion	Experimental Procedure	Time effect	Geometrical Model	Validation
[63]	Material	Solid mechanics	Drucker-Prager	Unconfined uniaxial compression	Constant	2D in vertical section plan	Unconfined uniaxial compression
[15]	Material	Rheology	Yield stress of non-Newtonian fluid	Rotational rheometer	Linear & exponential	vertical stack, 1D	Unconfined uniaxial compression
[1]	Material	Rheology	Yield stress of non-Newtonian fluid	n/a*	Linear	vertical stack, 1D	n/a
[17]	Material, stability	Solid mechanics	Mohr-Coulomb	Shear box + unconfined uniaxial compression	Linear	vertical stack, 2D in vertical section plan	Cylindrical print trial
[64]	Material, stability	Solid mechanics	Mohr-Coulomb	Shear box + unconfined uniaxial compression**	Linear & exponential decaying	Linear wall structures, 2D and 3D	Wall print trials
[10]	Material, stability	Mixed	Yield stress of non-Newtonian fluid	Rotational rheometer*	linear & exponential	vertical stack, 1D	n/a
[65]	Material	Rheology	Bulk yield stress from Benbow-Bridgewater model	Ram-extruder	Constant	vertical stack, 1D	Cylindrical print trial
[66]	Material	Solid mechanics	Shear stress	n/a**	Linear	Vertical stack, 2D along print path length	n/a
[61]	Material, stability	Solid mechanics	Mohr-Coulomb	Tri-axial compression	Linear	vertical stack, 2D in vertical section plan	Wall print trials
[67]	Material	Rheology	Yield stress of non-Newtonian fluid	Rotational rheometer	Bi-linear	vertical stack, 1D	Cylindrical print trial
<p>* theoretical, based on volume fraction and an average interparticle force. ** analytical paper, material failure properties based on data from other study. *** analytical paper, no direct relation to experimental work.</p>							

537 Di Carlo *et al.* [63] investigated the compressive strength of a fresh print mortar using a uniaxial
 538 compression test and found compressive strengths of 5.52 kPa to 88.3 kPa and moduli of elas-
 539 ticity of 77.9 kPa to 1241 kPa, for mortars at ages of 11 to 288 minutes (strength 10.7 kPa after
 540 47 minutes). They recreated the experiments in a Finite Element Model (FEM) in which a
 541 Drucker-Prager failure criterion was applied.

542 Alternatively, Perrot *et al.* [15] presented a buildability approach based on the rheological yield
 543 stress τ_c of the print mortar as the material failure criterion, which was relatively easily obtained
 544 from rotational rheometry and validated against uniaxial compression tests on shallow samples.
 545 Eq. (2) presents the basis of their approach in slightly rewritten form, with α_{geom} a geometrical
 546 factor, $\tau_0(t)$ the time dependent yield stress of the mortar, ρ its density, g the gravitational
 547 acceleration, and $h(t)$ the time dependent object height:

$$\alpha_{geom}\tau_0(t) \geq \rho gh(t) \quad (2)$$

548 The yield stress of the mortar applied in that study developed linearly over time until reaching
 549 an approximate age of 40 minutes, from 4.13 to 6.29 kPa ($A_{thix} = 54$ Pa/min), but it developed
 550 exponentially when considering a time frame of 0 to 80 minutes. The validation experiments fit
 551 the predicted failure moments very well. Further experimental rheological data on printable
 552 mortars can be found in [68, 69, 70].

553 Wangler *et al.* [1] adopted a very similar approach, but applied a von Mises plasticity criterion,
 554 by introducing a factor $\sqrt{3}$ into the buildability equation. On the other hand, a geometry factor
 555 was not applied. Rewritten, this yields:

$$\tau_0(t) \geq \rho gh(t)/\sqrt{3} \quad (3)$$

556 The first method to take both material and stability failure into account was presented by Wolfs
 557 *et al.* in [17], and further developed in [61, 71]. This method is based on FEM expanded with
 558 time-dependent properties, and the adoption of the Mohr-Coulomb failure criterion, including
 559 the development of experimental procedures to determine the full failure envelope. The dual
 560 failure criterion of plastic yielding and elastic buckling requires the experimental determination
 561 of five time-dependent material properties: the apparent Young's modulus $E(t)$, Poisson's ratio
 562 $\nu(t)$, the cohesion $C(t)$, angle of internal friction $\phi(t)$, and dilatancy angle $\psi(t)$. For two com-
 563 mercially available mortars this yielded linearly developing compression strengths from 5 to 90
 564 minutes of 7.1 kPa to 18.9 kPa (rate: 139.6 Pa/min) and 2.8 kPa to 29.7 kPa (rate: 317.2 Pa/min),
 565 and moduli of elasticity of 80.2 kPa to 186 kPa (rate: 1244 kPa/min) and 35.9 kPa to 325 kPa
 566 (rate: 3400 Pa/min). Suiker [64] introduced a parametric mechanistic model which focuses on
 567 the competition between elastic buckling and plastic collapse, and can be used to predict struc-
 568 tural failure of straight, free-standing walls during 3D printing.

569 A mixed methodology combining the rheology material failure criterion with the stability con-
 570 siderations from solid mechanics was suggested by Roussel [10]. Besides (3), this yields (4),
 571 again, slightly rewritten, representing the material and stability failure criteria, respectively.
 572 Through the relation $E = 2G(1 + \nu)$ between Young's modulus E , shear modulus G and Pois-
 573 son's ratio ν on the one hand, and the relation $\tau_c = G\gamma_c$ between yield stress, shear modulus
 574 and critical shear strain γ_c on the other, the transition height can be determined according to
 575 (5), at which one failure criterion becomes dominant over the other.

$$E_c(t) \geq 3\rho gh(t)^3/2\delta^2 \quad (4)$$

$$h_t = 2\delta \sqrt{\frac{1+\nu}{3\sqrt{3}\gamma_c}} \quad (5)$$

576 Continuing from the material assumptions by Perrot *et al.* [15], Wangler *et al.* [1], and Roussel
577 [10], Kruger *et al.* [67] suggested, rather than a linear structuration rate, a bilinear rate, which
578 yielded a shear strength development of 1.15 kPa to 2.73 kPa at a re-flocculation rate of $R_{\text{thix}} =$
579 413 Pa/min over the first 230 s, and further up to 6.37 kPa at 60 minutes at $A_{\text{thix}} = 64.8$ Pa/min.

580 A rheological approach with a Benbow-Bridgwater model rather than a Bingham model was
581 adopted by Chaves Figueiredo *et al.*, [65]. This yields the material parameters bulk yield stress
582 and shear yield stress, of which the former seems to be primarily responsible for the aspect of
583 buildability.

584 Finally, an analytical model was proposed by Jeong *et al.* [66], who took the age of the material
585 along the print path length into account. The shear strength (solid mechanics) was adopted as
586 failure criterion without extensive elaboration.

587 **6.1.3 Research needs**

588 Many aspects surrounding in-print failures are still unclear. Globally three topics seem to be
589 particularly in need of research.

591 First, little is known about the actual loads acting on in-print objects. Although self-weight, the
592 most important load, can be established relatively easily, the effects of kinematic pressures,
593 support materials, and eccentricities are not quantitatively known.

594
595 For material failures, the appropriate failure criterion or criteria need(s) to be established. A
596 fundamental underlying question is in which cases and whether at all it is relevant to allow for
597 anisotropic stress states for such a criterion, which requires considerably more complicated
598 analyses and experimental procedures. The accuracy of buildability predictions based on dif-
599 ferent methods and associated material property experiments should be rigorously tested
600 through extensive printing trials. Rather than focusing on a single case, such trials should en-
601 compass a range of different geometries, print speeds, print resolution, and object sizes as well
602 as their being performed on different printing facilities to eliminate equipment-dependent bias.
603 The development of a range of benchmark print geometries for comparative studies would be
604 beneficial.

605
606 Once the theoretically most satisfying approach has been found, the correlations between vari-
607 ous (experimental) methods may be studied to determine which approach is to be preferred
608 from a practical standpoint. In this campaign it should be recognized that the appropriateness
609 of a certain approach may depend on the properties of the individual print mortar or its rela-
610 tionship to the physical condition. For instance, it should be expected there are thresholds of
611 yield stress or viscosity above or below which solid or fluid mechanics are clearly more appli-
612 cable, but it is conceivable that transitional stages exist in which each approach may work sim-
613 ilarly well and the preferred approach may depend on other factors such as practical arguments.
614 Only a few multi-modal material characterizations have yet been performed [72, 73], and cor-
615 relation attempts to link different approaches are still minimal.

616
617 Finally, the transitory characteristics of print mortars should be more comprehensively studied.
618 Current buildability models only consider properties such as strength and modulus of elasticity
619 as a linear or non-linear function of time. It is, however, likely that such developments are a
620 function of multiple parameters, the most obvious of which is temperature. The temperature
621 dependency of buildability has tentatively been established [74], which indicates that time-de-
622 pendent development functions must be derived for a multitude of temperatures, or through a
623 combined parameter such as maturity, indeed a combination of time and temperature, as has
624 already been defined for concrete of ages beyond the setting time. Considering printing's not

625 always taking place under fully conditioned environmental conditions, this should be consid-
626 ered relevant. Possibly other parameters, such as the shearing history of the mortar may also
627 need to be included.

628

629 **6.2 Deformation behavior**

630

631 **6.2.1 Deformation mechanisms**

632 The actual deformations which occur in cementitious mortars during selective material deposi-
633 tion by extrusion *before* failure have hardly been studied. Two mechanisms should be distin-
634 guished. First, in some cases, depending on the solidification strategy of the process in question
635 (cf. Sections 3 and 4), the mortar flows briefly upon deposition before a rapid increase in
636 strength initiated by viscosity modifiers or accelerators stops the visco-plastic flow and the ma-
637 terial assumes a stable shape.

638 Other processes deposit mortars in an unsheared condition, directly in a shape-stable state,
639 which can last from minutes to hours. The visco-elastic deformations that may occur during
640 this period may influence both the geometrical conformity and the object stability during print-
641 ing.

642 To determine the elastic deformations, the apparent Young's modulus E may be determined
643 from unconfined compression experiments directly; cf. Section 6.1.2. Obtaining E through the
644 shear modulus G taken from rheometric tests does not seem feasible, as this requires the as-
645 sumption of a value for the Poisson's ratio ν .

646 Depending on which of the deformation mechanisms, flow or (visco-)elastic deformation, is
647 being studied, different modelling approaches can be adopted: Computational Fluid Dynamics
648 (CFD) for the former [75] or Finite Element Modelling (FEM) for the latter. A combination of
649 both may be required to capture the full deformation and displacement behavior. The Discrete
650 Element Method (DEM) might also be suggested, as it is suitable for granular materials and
651 theoretically capable of modelling both flow and deformations [76, 77].

652

653 **6.2.2 Research needs**

654 The accurate prediction of deformations and displacements is relevant to geometry-related fail-
655 ure behavior, i.e., primarily stability failure and, to a lesser extent, material failure, but to issues
656 of the geometric conformity of the finished product as well. Because little data on fresh mortar
657 deformation behavior is currently available, more research into the related material properties
658 is required. This includes the development of suitable experimental methods, for which some
659 initial suggestions have been made in [8]. Modelling methods need to be developed and vali-
660 dated, which also calls for the establishment of suitable means of measurement during or after
661 printing. The interaction between in-print failure and deformation behavior should be further
662 explored.

663

664 **7. Physical properties and their evolving over time**

665 **7.1 Viscosity**

666 Viscosity is the measure of the internal resistance of a fluid to its being deformed by shear
667 stresses. In this context the less viscous a fluid is, the easier it flows. Due to the flocculation
668 and stiffening of the cement-based materials, the apparent viscosity of the cement-based mate-
669 rial increases with time at rest. However, for yield stress this behavior is reversible in time since

670 the mixing power is sufficient to break down the links between particles. Here it is important
671 to note that the viscosity of the cement-based materials can evolve due to process-induced ma-
672 terial variation during pumping or extrusion. Such variation can be associated, for example,
673 with particle migration under shear flow.

674 **7.2 Yield stress**

675 Yield stress is the material property that denotes the transition between solid-like and fluid-like
676 behaviors. On the microscale, interparticle forces between the solids in a suspension result in a
677 yield stress that must be overcome to start flow. An applied stress lower than the yield stress
678 will result in deformation behavior like that of a solid. In other words, yield stress is the mini-
679 mum stress that makes the fluid flow as a viscous material. This minimum level of solicitation
680 depends on the stress history of the cementitious materials. Consequently, static and dynamic
681 yield stresses can be distinguished. Static yield stress is an increasingly important parameter
682 that depends on shearing and resting periods and is related to microstructural buildup and is
683 defined as the shear stress required to make the material flow, while dynamic yield stress can
684 be defined as the yield stress measured under the steady-state flow of material with unstructured
685 cement paste.

686 The major challenge associated with the 3D printing of cement-based materials is the determi-
687 nation of increase in static yield stress after a layer has been deposited. Since such measure-
688 ments are not trivial; see e.g. [78]. And there are numerous activities on developing adequate
689 test methods and protocols; see e.g. [79, 80].

690 **7.3 Thixotropy parameter / rate of structural buildup / structural breakdown**

691 Fresh cement-based materials undergo cement hydration, which impacts their physical and rhe-
692 ological behavior. It is this change in mechanical characteristics that allows the fresh material
693 to build up and be able to support the increasing load generated by the successive deposits of
694 the layers of the printed structure. Therefore, it is necessary to rely on the kinetics of the me-
695 chanical structural buildup of the material as they relate to the activity of cement in water [1,
696 15].

697 To understand why the cement material becomes rigid, it is necessary to take into account the
698 organization of the network of cement grains in suspension in the water. After intense mixing
699 culminating in de-structuring, the cement grains begin to flocculate under the influence of col-
700 loidal interactions. This flocculated lattice structure induces an increase in rigidity and strength,
701 over a time period of several tens of seconds [10, 81]. Then, over longer periods of time, the
702 rapid formation of C-S-H linkages in the contact zones between the grains goes on to form the
703 origin of the continued mechanical structural buildup of the cement material left at rest [81].
704 This phenomenon of the nucleation of the cement grains occurs during the period known as the
705 dormant period, before the cement setting. These nucleations are a chemically irreversible phe-
706 nomenon. However, if the power of the pumping and/or mixing system suffices, these links
707 may be broken. It is important to note that in the 3D printing of cement-based materials, accel-
708 erators are often added to the cementitious materials leading to the early formation of other
709 hydrates such as ettringite, monosulfate, and aluminate hydrates, which are responsible for the
710 early increase in static yield stress.

711 Therefore, the parameters of mechanical behavior are subject to the structural buildup kinetics
712 over time when the cement materials are left to set after being deposited in layers. To illustrate
713 and describe this phenomenon, the changes in the stiffness and the yield stress values have often
714 been reported by several researchers [81, 82, 83].

715 The increase in the progression of yield stress over time is often considered to be linear and, as
716 such, allows a structural buildup rate to be defined as A_{thix} in $\text{Pa}\cdot\text{s}^{-1}$ [81, 82, 83]. In this case,
717 the equation is written in this way:

$$\tau_0(t) = \tau_{0,t=0} + A_{thix}t \quad (6)$$

718 with $\tau_{0,t=0}$, the shear yield stress of the material in a de-structured state and t being the duration
719 of the resting period of the material.

720 The linear modelling of shear yield stress is generally valid during the first hour of the resting
721 period of the cement-based material [84]. Beyond that the change accelerates and the kinetics
722 of the structural buildup become exponential [85, 86]. This change in rates can be explained by
723 the beginning of the setting, the increase of the solid volume fraction with the possible inter-
724 penetration of the C-S-H crystals created. Perrot *et al.* [85] proposed a law for the exponential
725 progression of shear yield stress, tending towards the linear model over short time periods. This
726 model is given by (7):

$$\tau_0(t) = A_{thix}t_c(e^{t/t_c} - 1) + \tau_{0,t=0} \quad (7)$$

727 with t_c as a characteristic time over which the behavior can be considered linear. The model of
728 (7) can be used to describe the progression of the shear threshold over longer periods. Other
729 more sophisticated models have recently been reported in the literature [17, 18, 86]. Some au-
730 thors have even shown that a Von Mises-type plasticity criterion may ultimately not be well-
731 suited to account for the breakage of the material. Effectively, at a certain stage in the devel-
732 opment of the setting, the behavior of the material displays a pressure-sensitive, granular type
733 of behavior, probably related to the interconnection of the hydrates, then becoming sensitive to
734 pressure. The mechanical behavior thus represents a progressive dissymmetry, with a resistance
735 that is always higher in compression than in tension. Hence, the transition to hardened concrete
736 behavior can be seen to start at this time.

737 Mantellato *et al.* [87] recently reported that the increase in the yield stress associated with an
738 irreversible flow loss can be associated with an increase in the solid particles' specific surface
739 induced by chemical cement hydration. At the same time the critical deformation decreases
740 slightly with the hardening of the material [10, 17, 18].

741 Additionally, this increase in stiffness over time is reflected in an increase in the elastic modu-
742 lus. This increase in stiffness and strength allows the material deposited to withstand the in-
743 creased loads associated with the printing of the structure. In this way it is possible to calculate
744 and predict the optimal manufacturing speeds to guarantee the stability of the structure printed
745 and to ensure the compensation for the elastic deformation.

746

747 **7.4 Measurement procedure of rheological and early age properties**

748 To assess the stability of the in-print cementitious materials, it is necessary to follow the evo-
749 lution of the materials' early-age properties after being deposited. Then effort must be made to
750 monitor inline the evolution of stiffness and strength of the cementitious material.

751 If oscillatory rheometry and ultrasound test measurements are reliable methods to estimate the
752 evolution of a material's elastic modulus over time, they require expensive and sensitive de-
753 vices that are not easy to implement in a printing environment [50].

754 To relieve this situation, instantaneous or continuous penetration tests or gravity-induced flow
755 tests are being developed in order to monitor, using simple tools, the evolution of the materials'
756 properties over time. For example, the penetration test (cone plunger) was used by filling the
757 cone mold with mortar in two layers, then releasing the cone plunger, thus allowing the plunger

758 to penetrate into mortar under its own weight [88]. Good relationships ($R^2 = 0.86 - 0.89$) be-
759 tween the penetration and slump flow were established. It was reported that the increase in the
760 penetration values followed the same trend with the increase in the slump flow of mortar.

761 **8. Summary and conclusions**

762 3DCP offers great potential to facilitate development towards a sustainable Construction Indus-
763 try 4.0, tackling existing problems such as low productivity and shortage of skilled labor in the
764 process. Among 3DCP approaches, those based on material extrusion seem to be particularly
765 promising at this stage with respect to both overall technological readiness level and economic
766 viability. This article focuses on this specific 3DCP approach. The explanations and statements
767 are the result of collective research performed by the authors in the framework of the RILEM
768 Technical Committee 276 “Digital fabrication with cement-based materials”.

769 Three main categories of extrusion-based 3DCP are identified as: i) extrusion of stiff material,
770 similar to conventional extrusion, ii) extrusion of flowable material with or without addition of
771 admixture(s) in the printhead, and iii) extrusion of material using additional energy input, e.g.
772 vibration to facilitate delivery and deposition of stiff mixtures. To each category and each pro-
773 duction step, i.e., transportation of build material, printhead process, deformation of build ma-
774 terial during deposition, and behavior of build material after deposition, relevant processes and
775 corresponding physical fundamental are presented. In particular, gravitational flow, pressure-
776 induced flow during pumping and extrusion, and dispersion of admixture in the printhead are
777 considered. Also, attention is paid to the load-bearing and deformation behavior of the mate-
778 rial/printed object after material deposition. Two major failure modes are defined for object
779 failure during manufacturing, both crucial to buildability: i) material failure, where the material
780 strength is exceeded, and ii) stability failure due loss of the equilibrium of forces and moments.
781 Various models to estimate buildability are presented and critically discussed. Deformation
782 mechanisms are explained as well. Finally, key physical properties of cement-based materials
783 in the fresh state – viscosity, yield stress, and thixotropy – as well as their evolution in time are
784 identified, followed by brief remarks on measurement procedures of rheological and early-age
785 properties.

786
787 In summarizing the knowledge as presented, it can be concluded that underlying physics are
788 well understood for most processes of large-scale additive manufacturing by material extrusion.
789 This understanding can and should be utilized for the purposeful design of 3DCP systems rather
790 than trying to use a trial-and-error approach in shaping the 3DCP process. Purposeful, system-
791 atic approaches based on the associated physics should facilitate material development and me-
792 chanical engineering design as well as optimize process regimes and process control. For some
793 processes analytical, scientifically based formulas already offer reasonable predictions with re-
794 spect to material flow in the case of relatively simple geometries. Nevertheless, further research
795 is needed in order to enable the development of reliable tools for quantitative process analysis
796 and for predictions based on the underlying physics. The major challenges in analyzing 3DCP
797 systems arise out of the complexity of flow regimes and patterns in various production steps as
798 well as the time- and shear-history-dependent behavior of cement-based materials, which are
799 inherently complex multiscale, multiphase, densely packed suspensions. Much effort needs to
800 be invested in studying and describing specific flow behavior and developing adequate testing
801 technics to quantify key material parameters. Numerical simulation can contribute greatly to
802 analyze flow processes under consideration of complex geometric boundaries; this has the po-
803 tential to be developed into a powerful design tool for shaping the 3DCP processes. The deri-
804 vation of model parameters is the main issue here, requiring appropriate experimentation. This
805 deliberation holds true as well for the estimation of buildability. Material behavior, geometry

806 of the printed element, particularities of printing process, and other aspects need to be consid-
807 ered collectively. While analytical formulas may deliver reasonable predictions for relatively
808 simple cases, numerical simulation constitutes a promising approach for analysis of more com-
809 plex cases.

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813

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826 **References**

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