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Wen, C. F., Zhou, M., Liu, P. F., Liu, Y., Wu, X., Mao, F., Dai, S., Xu, B., Wang, X. L., Jiang, Z., Hu, P., Yang, S., Wang, H. F., & Yang, H. (2022). Highly ethylene-selective electrocatalytic CO₂ reduction enabled by isolated Cu-S motifs in metal-organic framework-based precatalysts. *Angewandte Chemie International Edition*, 61(2), Article e202111700. <https://doi.org/10.1002/anie.202111700>

Published in:
Angewandte Chemie International Edition

Document Version:
Peer reviewed version

Queen's University Belfast - Research Portal:
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Accepted Article

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This manuscript has been accepted after peer review and appears as an Accepted Article online prior to editing, proofing, and formal publication of the final Version of Record (VoR). This work is currently citable by using the Digital Object Identifier (DOI) given below. The VoR will be published online in Early View as soon as possible and may be different to this Accepted Article as a result of editing. Readers should obtain the VoR from the journal website shown below when it is published to ensure accuracy of information. The authors are responsible for the content of this Accepted Article.

To be cited as: *Angew. Chem. Int. Ed.* 10.1002/anie.202111700

Link to VoR: <https://doi.org/10.1002/anie.202111700>

Highly ethylene-selective electrocatalytic CO₂ reduction enabled by isolated Cu-S motifs in metal-organic framework-based precatalysts

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Abstract: Copper-based material is an efficient electrocatalyst for CO₂ conversion to C₂₊ products, most of which will reconstruct to *in-situ* regenerate active species. It is a challenge to precisely design precatalyst to obtain active sites for CO₂ reduction reaction (CO₂RR). Herein, we develop a local sulfur doping strategy over the Cu-based metal-organic frameworks precatalyst that disperses stable Cu-S motif in the framework of HKUST-1 (S-HKUST-1). The precatalyst exhibits a high ethylene selectivity in H-type cell with maximum faradaic efficiency (FE) of 60.0%, and delivers a current density of 400 mA cm⁻² with ethylene FE to 57.2% in a flow cell. Operando X-ray absorption results demonstrate that Cu^{δ+} species stabilized by Cu-S motif exist in S-HKUST-1 during CO₂RR. Density functional theory calculations indicate the partially oxidized Cu^{δ+} at Cu/Cu_xS_y interface is favorable for *CO intermediate coupling due to the modest distance of coupling sites and optimized adsorption energy.

Introduction

The electrochemical reduction of carbon dioxide (CO₂) into value-added feedstocks is a desirable route to mitigate the depletion of fossil fuels and alleviate the greenhouse gas emissions^[1]. The high-value multi-carbon (C₂₊) hydrocarbons products *via* CO₂

reduction reaction (CO₂RR) are highly attractive due to their practical applications^[2]; however, their selectivity and activity are severely limited by multi-step hydrogenation steps and competitive reactions^[3]. Copper-based material is the most efficient electrocatalyst for selectively convert CO₂ to C₂₊ products^[4], and its performance can be further influenced *via* modulating oxidation states^[5], grain boundaries^[6] and unsaturated sites^[7]. It's worth noting that these structural features of copper-based material can be *in-situ* changed due to the precatalyst reconstruction under CO₂RR conditions. Taken copper oxides as a classic example, they usually act as CO₂RR precatalysts, which can be partially reduced to regenerate biphasic copper/copper oxide, coupled with abundant grain boundaries and unsaturated sites as active CO₂RR species, exhibiting significantly increased CO₂RR performance than the pure copper^[5, 8].

Besides the widely reported copper oxides-based precatalysts^[9], other copper-based compounds^[10] have also been chosen as efficient precatalysts to introduce stable oxidized Cu^{δ+} species for C₂₊ production, due to the more stable Cu(X) (X means the B, N or other heteroatoms) species than the Cu(O) matrix under CO₂RR condition^[10]. Unfortunately, these copper-based compounds could not elaborately regenerate abundant biphasic Cu^{δ+} species which features isolated Cu(X) species in the

reconstructed catalyst. It still remains as a big challenge to precisely control the local structure of precatalysts to regenerate stable and active sites under CO₂RR condition, especially for C₂₊ products. Unlike metal compounds, metal-organic frameworks-based (MOFs-based) material is an ideal precatalyst due to its controllable size/shape, high surface area, chemical tunability and the open metal sites^[11], which affords a platform for incorporating isolated active species at atomic level. Therefore, we anticipate isolating stable Cu(X) species into the Cu-based MOFs precatalyst might regenerate homogeneously dispersed active sites under CO₂RR condition.

Herein, we rationally constructed isolated Cu-S motif on HKUST-1 precatalyst by a local sulfur doping strategy (S-HKUST-1). Although Cu(S)-based materials can electrochemically reduce CO₂ to major products of formate^[12], some Cu(S) catalysts have recently been reported with enhanced CO₂ reduction selectivity to C₂₊ products^[7], because of the local modification of geometric and electronic structure of Cu sites, which would influence its activity and selectivity. In this work, the S-HKUST-1 can be *in-situ* reconstructed to give Cu(S) matrix with abundant and active biphasic copper/copper sulfide (Cu/Cu_xS_y) interfaces, which can obtain a high ethylene (C₂H₄) selectivity in the H-type cell with the maximum faradaic efficiency (FE) of 60.0%, and delivers a current density of 400 mA cm⁻² with FE_{C₂H₄} reaching up to 57.2% and FE_{C₂} (FE of C₂H₄, C₂H₅OH and CH₃COOH) of 88.4% in the flow cell configuration. The systematic characterizations demonstrate the stable existence of Cu-S motif before and after CO₂RR. Density functional theory (DFT) calculations further reveal that the stabilized Cu^{δ+} at the Cu/Cu_xS_y interfaces featuring the modest distance of coupling sites and the optimized kinetic barrier for CO intermediate dimerization. This work provides a new pathway to design efficient CO₂RR electrocatalyst *via* rationally constructing precatalysts at atomic scale.

Results and Discussion

Synthesis and local structural characterization of S-HKUST-1 precatalyst

The S-HKUST-1 precatalyst was prepared by a wet chemistry method, that is, a certain amount of as-prepared HKUST-1 was immersed in the ethanol solution of thioacetamide for 1 hour (h) at room temperature. It is mentionable that the thioacetamide slowly react with the contained water in the HKUST-1 to generate H₂S for construction of local Cu-S coordination^[13], to give S-HKUST-1 precatalysts (Figures 1a and S1, see the Methods in Supplementary Information for details). The Cu-S motif could be atomically dispersed in the precatalyst and no aggregated copper sulfide could be generated if the amount of thioacetamide was elaborately controlled.

Both S-HKUST-1 and HKUST-1 precatalysts display the octahedral morphology (Figure S2), which were confirmed by scanning electron microscopy (SEM) characterizations. As shown in Figure S3, the scanning transmission electron microscopy (STEM)-based energy-dispersive X-ray spectroscopy (EDS) reveals that sulfur is distributed evenly throughout the S-HKUST-1 particle. Furthermore, sulfur can be observed on ultrathin slices of S-HKUST-1, which directly shows that sulfur is distributed

within the whole S-HKUST-1 particle, instead of just on the surface of the catalyst (Figures S3b, S3d, S4, Tables S1 and S2). The X-ray diffraction (XRD) pattern of as-prepared HKUST-1 is similar to the simulated HKUST-1 reference, suggesting the successful synthesis of crystalline HKUST-1^[14]. In addition, the XRD patterns of S-HKUST-1 (with different S contents) match well with HKUST-1 diffraction pattern, and no other crystalline species appeared (Figures 1b and S5), displaying that the framework structure of HKUST-1 was still maintained and no aggregated copper sulfide could be generated. From single-crystal X-ray diffraction (SXRD) data, the crystal data and structure refinement of S-HKUST-1 and HKUST-1 show no obvious change, which is similar to HKUST-1 reference (Figure S6, Tables S3 and S4).^[15] As shown in Figure 1c, the Fourier transform infrared (FTIR) spectrum of S-HKUST-1 is nearly similar to that of HKUST-1, showing asymmetric vibration of O=C-O at 1645 cm⁻¹, 1450 cm⁻¹, symmetric vibration of O=C-O at 1374 cm⁻¹, vibration of C-O at 1114 cm⁻¹, and vibration of phenyl at 758 cm⁻¹ and 731 cm⁻¹, respectively^[16]. The Raman spectra also indicate that the framework structure and organic bonding of HKUST-1 are maintained after S incorporation (Figure S7). The amount of S element in S-HKUST-1 precatalyst was estimated based on the inductively coupled plasma optical emission spectrometry (ICP-OES) analysis and the elemental analysis (EA). The ICP-OES and EA data in the Tables S5 and S6 show that the weight fractions of S elements in S-HKUST-1 is 0.33 and 0.17 wt%, respectively.

The surface composition and elemental states of S-HKUST-1 and controlled sample were further characterized by X-ray photoelectron spectroscopy (XPS). Compared with that of HKUST-1, a broad S 2p peak is visible at 162.5 eV in S-HKUST-1, which is consistent with Cu-S bonding^[12a, 17] (Figure 1d). This result unambiguously confirms the successful incorporation of S element in HKUST-1. In the XPS spectra of Cu 2p_{3/2} region, both S-HKUST-1 and HKUST-1 precatalysts possess a peak centered at the binding energy of 934.7 eV, which is attributed to Cu²⁺ species, demonstrating that partial Cu centers of the S-HKUST-1 still maintain coordination with trimesic acid. The peak at 932.4 eV associated with Cu⁰/Cu⁺ species was observed for S-HKUST-1 and HKUST-1^[18]. The peak intensity for Cu⁰/Cu⁺ components significantly increases in S-HKUST-1 (Figure 1e), further suggesting that Cu²⁺ species were partially reduced after S incorporation to form Cu-S coordinated species. The Cu LMM auger spectra of HKUST-1 and S-HKUST-1 were measured for the further clarification of Cu valency (Figure S8). Compared with HKUST-1, S-HKUST-1 exhibits a broad peak at the binding energy of 570.0 and 567.9 eV, which indicates the presence of Cu⁺ and Cu⁰ ^[19]. Sulfur is lower electronegative than oxygen, so that the attract-electrons ability of S atoms is weak, which might lead to a relatively low valence of Cu. We speculate that the local Cu-O bond in HKUST-1 might be broken by the treatment of thioacetamide molecules for construction of Cu-S motif.

To confirm the transformation process from HKUST-1 to S-HKUST-1 proposed, the changes of oxidation states and local coordination environments of Cu atoms were investigated by X-ray absorption near-edge structure (XANES) and extended X-ray absorption fine structure (EXAFS) analyses, respectively. Figure S9 shows the high-quality *k*-space (*k*³-weighted) Cu K-edge EXAFS spectra of HKUST-1, S-HKUST-1 and controlled Cu foil,

CuO, Cu₂S and Cu₂O samples, which can support the reliability of Fourier transformation (FT) of the EXAFS spectra^[20]. The FT of EXAFS spectra for S-HKUST-1 and original HKUST-1 exhibit similar curves featuring a main peak at approximately 1.53 Å (without phase correction if not mentioned) arising from Cu-O bond (Figure 1f), suggesting that the main structure of HKUST-1 is maintained. When incorporating S elements into HKUST-1 to form S-HKUST-1, the Cu-O peak exhibits a remarkable intensity decrease, which is consistent with those reported in literatures^[21], indicating that the water molecule directly linked with the Cu atoms are partially lost. Furthermore, the main Cu-O peak for S-HKUST-1 shifts to longer lengths, suggesting the expansion of the Cu-O bond due to the S incorporation to construct local Cu-S motif. From the XANES spectra of HKUST-1 and S-HKUST-1 (Figure 1g), two peaks were observed, including a pre-edge peak at 8981 eV corresponding to 1s→3d transition and a white-line peak at 8997 eV^[21]. When S incorporated into the HKUST-1, the decreased intensity for the white line peak and the increased intensity for the 1s→3d peak indicate a change in the coordination of Cu centers, which is resulted from the removal of water

molecule^[22]. The obtained precatalysts with different S contents also exhibit the similar phenomena, indicating the increased intensity of Cu-O peak, the decreased intensity of white line peak and the increased intensity of the 1s→3d peak (Figure S10). The electron paramagnetic resonance (EPR) signal intensity of S-HKUST-1 also increases and the peak becomes sharper compared to the HKUST-1, indicating that the Cu microstructure has changed^[16] (Figure 1h). As shown in Figure S11, the BET surface area of S-HKUST-1 was determined to be 816.54 m² g⁻¹, which is lower than that of HKUST-1 (855.52 m² g⁻¹). This may be due to the fact that part of the thioacetamide molecules entered the cavity of MOF, forming Cu-S species and reducing its pore cage, which eventually leads to a slight decrease in the BET surface area^[13]. Above characterization results have clearly shown the existence of the isolated Cu-S motif in HKUST-1 precatalyst. We speculate that the S heteroatoms generated from the thioacetamide might locally replace the flanking water molecule (O_w) attached to the Cu atoms and break the Cu-O_w bonds without destroying the long-range order of HKUST-1^[22b].

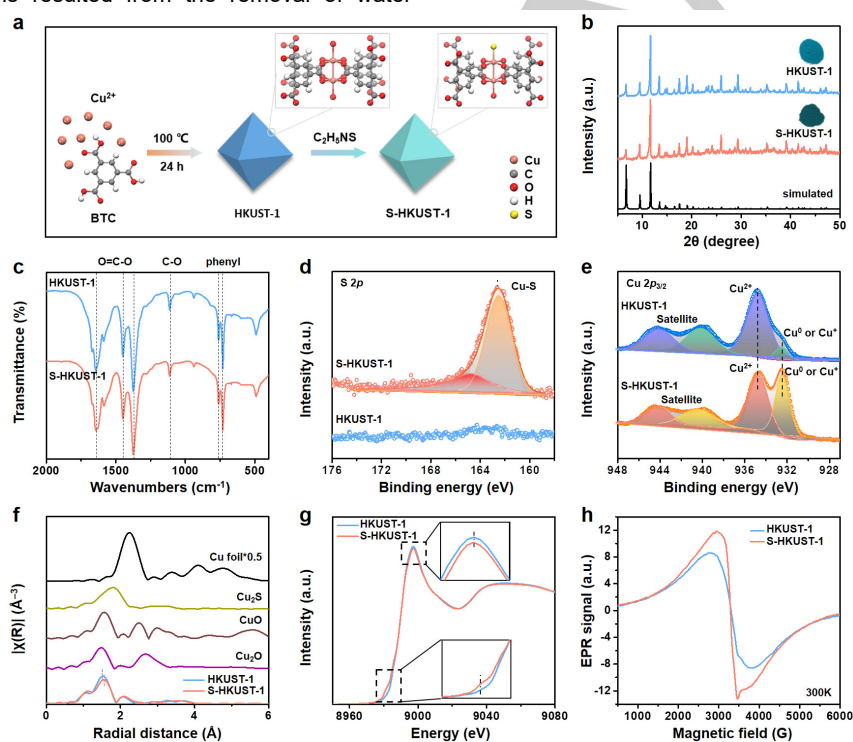


Figure 1. (a) Schematic illustration of the preparation of S-HKUST-1, which indicates the local H₂O molecule might be replaced by S heteroatoms. (b) XRD patterns of the prepared S-HKUST-1 and HKUST-1 precatalysts, which are well indexed to the simulated HKUST-1. The insets are digital photos of S-HKUST-1 (dark green) and HKUST-1 (blue). (c) FTIR spectra of S-HKUST-1 and HKUST-1 precatalysts, indicating negligible difference. The XRD and FTIR results prove that the long-range ordered structure has not been destroyed after S incorporation. (d, e) High-resolution XPS spectra of S-HKUST-1 and HKUST-1 precatalysts in (d) S_{2p} region, showing the characteristic Cu-S bonds in the S-HKUST-1 precatalyst and (e) Cu_{2p} region, showing the increased content of Cu⁺ or Cu⁰ species. (f) FT of the EXAFS spectra and (g) Cu K-edge XANES spectra of HKUST-1 and S-HKUST-1 precatalysts. The inset of (g) is the magnified image. The XAFS results in (f, g) prove the successful incorporation of local heteroatoms, which might be bonded to Cu atoms in MOFs. (h) EPR spectra of the samples measured at 300 K.

Electrocatalytic activity and selectivity evaluation of CO₂RR

The electrochemical CO₂RR performance was firstly evaluated using a gas-tight H-type cell, with S-HKUST-1 or HKUST-1 samples decorated glassy carbon electrode, Ag/AgCl and Pt mesh as working electrode, reference electrode and counter electrode, respectively (see more details in the Methods). We

firstly studied the effects of different incorporated S contents, and their corresponding electrocatalytic CO₂RR selectivity results (biased at -1.22 to -1.52 V vs. RHE, potentials were referenced to RHE if not mentioned, and all potentials were not *i*R-corrected) illustrate that as the incorporated S contents increase, the FE_{C₂H₄} gradually increases; however, when the S content further increases, the FE_{C₂H₄} is significantly reduced (Figure S12). In addition, the excess S incorporated sample of CuS obtained from

vulcanization shows a poor selectivity towards CO₂RR (Figures S12c and S12d), which proves the critical role of isolated Cu-S motif in the S-HKUST-1 precatalyst. The CO₂RR activities of the optimized S-HKUST-1 and pristine HKUST-1 were summarized in Figure 2. Linear sweep voltammetry (LSV) curves for S-HKUST-1 and HKUST-1 were conducted in a CO₂-saturated 0.1 M KHCO₃ aqueous solution (Figure 2a). The S-HKUST-1 shows higher reduction current densities relative to HKUST-1. The liquid products were collected and analyzed by using ¹H nuclear magnetic resonance (NMR) spectroscopy. As shown in Figure S13, no obvious detectable liquid products were produced, which might be attributed to ultralow content of liquid products in the H-type cell. Figure S14 exhibits the distribution of CO₂RR gas products at different applied potentials on S-HKUST-1. For S-HKUST-1, the FE_{C₂H₄} maintains values of above 50.0% in a range of potentials from -1.22 to -1.42 V, and are higher than that of HKUST-1 (Figure 2b and Table S7). A maximum FE_{C₂H₄} of 60.0±2.0% can be reached at -1.32 V, while this value of HKUST-1 is only 24.4±2.1% (Figure 2b). The electrochemical active surface area (ECSA) for S-HKUST-1 and HKUST-1 were measured (Figure S15). To compare the intrinsic activity for CO₂RR, the ECSA-normalized *j* of C₂H₄ was obtained (Figure 2c). It reveals that the higher activity for ethylene production on S-HKUST-1 than HKUST-1. The Nyquist plots in Figure S16 reveal that S-HKUST-1 is characterized with a lower interfacial charge-transfer resistance (*R_{ct}*) than HKUST-1, evidencing a faster electron transfer from the electrodes to CO₂ during CO₂RR process^[10b, 23], which may also contribute to the higher catalytic *j* of S-HKUST-1 than the HKUST-1 control (Figures 2a and 2c).

The reaction rate of CO₂RR might be limited by mass transport due to the low solubility of CO₂ in the H-type cell^[24]. Thus, we further evaluated the CO₂RR performance of S-HKUST-1 in a flow cell system with 1.0 M KOH as electrolyte (Figure 2d, digital photos were shown in Figure S17). H₂, C₁ and C₂ products were formed on bare copper and HKUST-1 controls, and the FEs of H₂ and C₁ products decreased and C₂ products increased over S-HKUST-1 (Figure 2e and Tables S8-S10). The C₂ products include ethylene, ethanol, acetate, and among them, ethylene is the major product (Figure S18). For comparison, the FE_{C₂H₄} on bare Cu and HKUST-1 electrodes are generally below 40% over the current densities from 100 to 500 mA cm⁻² (Figure S19 and Tables S8-S10). The FE_{C₂H₄} on S-HKUST-1 is obviously higher than that of HKUST-1 across the *j* range from 100 to 500 mA cm⁻². At the *j* of 400 mA cm⁻², H₂ selectivity remains below 20% FE, whereas the FE towards C₂H₄ surpasses 57.2% with the C₂H₄ partial *j* of 228 mA cm⁻² (Figures 2e and S20). This is the highest selectivity of C₂H₄ and the highest C₂H₄ partial *j* among the Cu-based organometallic catalysts and MOFs; furthermore, the S-HKUST-1 also shows comparable C₂H₄ selectivity and activity among the reported Cu-based electrocatalysts (Table S11). When we used the S-HKUST-1 precatalyst, the Faradaic efficiency of overall C₂ product is 70.0~88.4% from 100 to 500 mA cm⁻², with peak selectivity of C₂ products towards FE of 88.4% at 400 mA cm⁻², which is higher than those of the bare Cu and HKUST-1 electrodes (Figure S21). Noteworthy, the S-HKUST-1 also demonstrates good electrocatalytic stability, retaining nearly unchanged FE of C₂H₄ (~50%) at an applied *j* of 150 mA cm⁻² in a flow cell reactor (Figure 2f). Taken together, we demonstrate that isolating Cu-S motif in the HKUST-1 precatalyst could subtly activate the C₂H₄ performance.

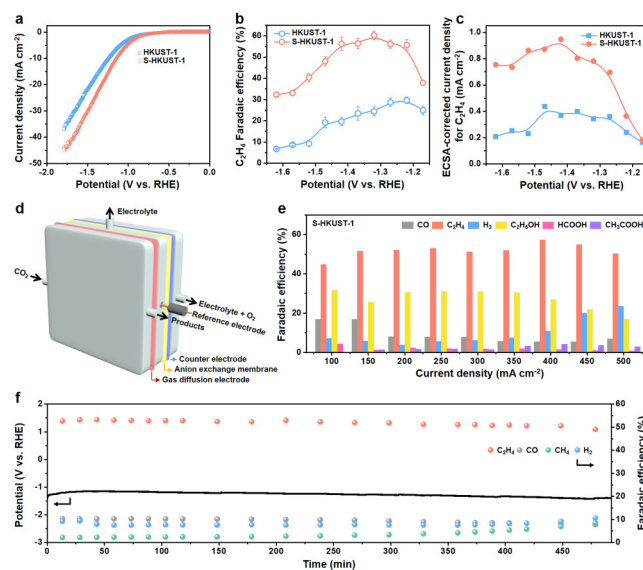


Figure 2. (a) LSV curves in the CO₂-saturated 0.1 M KHCO₃ aqueous solution at a scan rate of 10 mV s⁻¹ for S-HKUST-1 and HKUST-1. The S-HKUST-1 shows larger catalytic current density relative to HKUST-1. (b) FE of C₂H₄ and (c) ECSA-normalized *j* of C₂H₄ for S-HKUST-1 and HKUST-1 in the CO₂-saturated 0.1 M KHCO₃ at various applied potentials. (d) Schematic illustration of the flow cell reactor. (e) FEs of products with S-HKUST-1 precatalysts at various applied *j* ranging from 100 to 500 mA cm⁻² in liquid-electrolyte flow cell reactor. FE of CH₄ is below 1% at all applied *j* and was not plotted. (f) Long-time stability of S-HKUST-1 in 1.0 M KOH at an applied *j* of 150 mA cm⁻² in a flow cell reactor, showing stable activity and selectivity for C₂H₄ over 480 min.

Structural evolution of the precatalyst

We firstly conducted *ex-situ* characterizations to study the structural evolution before and after CO₂RR (after reacting in CO₂-saturated 0.1 M KHCO₃ at a potential of -1.32 V for 1 h) over the precatalysts. In order to avoid the oxidation of copper as much as possible, the entire cell was transferred to the nitrogen glove box immediately after CO₂RR (Figure S22). The XPS spectra of the S-HKUST-1 precatalyst given in Figure S23a show that the peak of Cu²⁺ species centered at 934.7 eV was disappeared, which indicates that the valence states of Cu changed from Cu²⁺ to Cu⁺/Cu⁰ after CO₂RR^[25]. Further surface state analysis was performed with a Cu LMM auger peak, which shows the Cu⁺ auger peak at 570.0 eV and Cu⁰ peaks at 567.9 eV (Figure S23b)^[19]. It is obvious that both Cu⁰ and Cu⁺ states are present on the surface of the catalyst after the CO₂RR. In the XRD pattern, the S-HKUST-1 precatalyst has lost the characteristic diffraction peaks of original MOFs, but exhibits typical diffraction peaks of Cu(111) and Cu(200) at 43.3° and 50.4° (PDF 04-0836), which is consistent with the XPS results (Figure S24). Concomitantly, the morphological changes are revealed by the TEM images of S-HKUST-1 and HKUST-1 precatalysts after CO₂RR, which indicates that both samples have changed from the initial micron sized octahedral particles to the final dendritic morphologies (Figures 3a, S25a and S26a), which is consistent with the SEM results (Figures S27). Moreover, the high-resolution transmission electron microscopy (HRTEM) images (Figures 3b, S25 and S26) display the lattice fringes with a spacing distance of 2.08 Å, which is corresponding to the Cu(111) crystal planes, also demonstrating that the Cu ion in precatalysts have been reduced

to metallic Cu during CO₂RR process. The HAADF-STEM and corresponding EDS elemental maps show that Cu and S elements distributed homogeneously throughout the whole catalyst structure (Figures 3c and S26d-S26e). We further replaced the Nafion solution with an anion solution (Sustainion® XA9 Ionomer solution), which doesn't contain the sulfur element. The EDS analysis revealed that sulfur also exist in the catalyst after CO₂RR, indicating that S element is derived from the S-HKUST-1 catalyst (Figure S28). The XPS result of S-HKUST-1 after CO₂RR also indicates the existence of S element, which is consistent with the above-mentioned characterizations (Figure S29 and S30). We speculate that the regenerative Cu(S) matrix during CO₂RR might give abundant Cu/Cu_xS_y interfaces on the atomic level local structure.

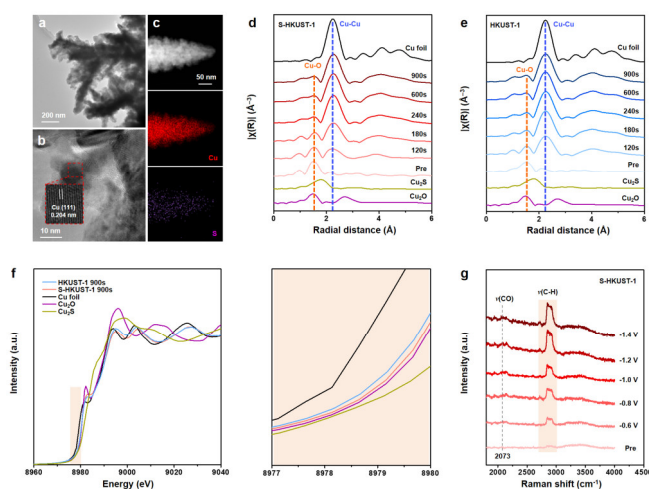


Figure 3. (a) TEM and (b) HRTEM images of S-HKUST-1 collected after CO₂RR electrocatalysis process. The inset of (b) is the enlarged HRTEM image showing the Cu(111) lattice fringe. (c) HAADF-STEM image for S-HKUST-1 after CO₂RR electrocatalysis and corresponding EDS elemental maps of Cu, S and mix. (d, e) Operando FT Cu K-edge EXAFS spectra with respect to time for (d) S-HKUST-1 and (e) HKUST-1 precatalysts under an applied potential of -1.30 V in CO₂-saturated 0.1 M KHCO₃. (f) Operando Cu K-edge XANES spectra of S-HKUST-1 and HKUST-1 after 15 minutes of reaction at -1.30 V vs. RHE in 0.1 M KHCO₃. (g) *In-situ* Raman spectra of S-HKUST-1 during CO₂RR at different potentials in CO₂-saturated 0.1 M KHCO₃.

In order to further study the local coordination environments and chemical states of the S-HKUST-1 and HKUST-1 under realistic CO₂RR conditions, operando X-ray absorption fine structure (XAFS) analysis was performed^[26]. In the operando EXAFS spectra for S-HKUST-1 and HKUST-1 precatalysts, the intensity of the Cu-O peak gradually decreases and the intensity of the Cu-Cu peak gradually increases (Figures 3d and 3e). This result confirms that the formed Cu⁰ species are regenerative from Cu²⁺ species in S-HKUST-1 and HKUST-1 precatalysts. In the operando XANES spectra in the Cu K-edge for S-HKUST-1 and HKUST-1 precatalysts at the working potential of -1.30 V (Figure S31), the curves of S-HKUST-1 and HKUST-1 gradually become similar to that of the Cu foil reference as the CO₂RR continues, suggesting that the pristine main components of Cu²⁺ in S-HKUST-1 and HKUST-1 are reduced to low-valent Cu during CO₂RR testing conditions, which is consistent with recently reported results for HKUST-1 precatalysts^[16, 25b, 27]. In particular, the comparison of Cu K-edge adsorption of S-HKUST-1 and

HKUST-1 catalysts with the reference samples suggests that the average oxidation states of Cu in S-HKUST-1 and HKUST-1 derived catalysts lie between 0 and +1 after 15 minutes of reaction at -1.30 V (Figure 3f). Furthermore, the absorption edge of S-HKUST-1 is on the higher energy side than that of HKUST-1, implying the higher Cu oxidation state^[28]. The result show that the regenerative Cu(S) matrix derived from predesigned S-HKUST-1 features more oxidized Cu^{δ+} species than the HKUST-1 control, which might give abundant active sites for C₂ products generation.

In-situ Raman measurement was carried out to further investigate the origins of the excellent CO₂RR performance of S-HKUST-1. A custom-built Raman setup was used to probe the surface of S-HKUST-1 and HKUST-1 during CO₂RR at different potentials in CO₂-saturated 0.1 M KHCO₃ (Figure S32). The vibration of C-H on S-HKUST-1 and HKUST-1 was observed in the area at 2700-3000 cm⁻¹ (Figures 3g and S33).^[29] The peaks are more obvious on S-HKUST-1 than that on HKUST-1, indicating that more reaction intermediates containing C-H bonds are produced on S-HKUST-1. Remarkably, S-HKUST-1 catalyst significantly improved the formation of key hydrocarbon intermediates, which might promote the subsequent C-C coupling step and thus the production of hydrocarbons.

Theoretical studies of biphasic Cu/Cu_xS_y interface

To elucidate the superior selectivity of Cu(S) matrix on catalyzing CO₂RR into C₂ products, we further performed the density functional theory (DFT) calculations to reveal the inherent mechanism (see the calculation details in the Methods). Essentially, it has been widely regarded that the bound *CO intermediate (* denotes an adsorption site) is the key species in the formation of both C₁ and C₂ products^[30], and the *CO hydrogenation (*CO + *H → *COH/*CHO + *) versus the dimerization process (2*CO → *OCCO + *) serving as the rate-limiting steps (RLS) largely limits the selectivity of C₂ formation^[31], which is confirmed by the thermodynamic energy profiles for these two processes computed over the Cu(111) surface (Figure 4a; see more details in Figure S34 and Table S12). Thus, we focused on these two elementary steps for comparison to understand the CO₂RR activity and selectivity on Cu(S) matrix. Regarding the construction of Cu(S) matrix, the possible structures are diverse including stoichiometric and non-stoichiometric sulfide phases. However, because of the experimental characterization on the valence states of surface and bulk Cu (around 0 ~ 1), the stoichiometric Cu(S) systems are ruled out (*i.e.*, CuS and Cu₂S). Estimated from the thermodynamic phase diagrams in Figure S35, the binary Cu/Cu₂S interface structure can be expected to dominantly exist, and thus we constructed the Cu/Cu₂S interface for modelling the realistic active site eventually (Figure S36, see more details in the Methods). Notably, the average charge of Cu atoms near the optimized Cu/Cu₂S interface is 0.14 |e| from the Bader charge analysis, indicating that these Cu sites are in the form of partially oxidized Cu^{δ+}, which is in line with the experimental observation.

The *CO dimerization step was calculated at the Cu/Cu₂S interface (Figure 4c). The effect of realistic solution environment is considered in the calculation, in which explicit water layers are introduced over the catalyst to simulate the liquid/solid interface by invoking the ab initio molecular dynamics (AIMD) simulations

(Figure S37). Figure 4d illustrates the located transition state, in which the $^{\circ}\text{OC}\dots\text{CO}^{\circ}$ intermediate is stabilized by the neighboring H_2O molecules via hydrogen bonds, and the corresponding barrier is calculated to be 0.67 eV. In comparison, this barrier is lower than that over $\text{Cu}(111)$ (0.91 eV, close to the calculation of Sargent et al.^[30a]), suggesting that the activity of oxidized $\text{Cu}^{\delta+}$ at the $\text{Cu}/\text{Cu}_2\text{S}$ interface is higher than pure metal $\text{Cu}(111)$ for C_2 production. Moreover, the bulk Cu_2S with high concentration of S atoms was examined and found less active for $^{\circ}\text{CO}$ dimerization because of the long distance between $\text{Cu}-\text{Cu}$ cations (3.927 Å), which prevented the $^{\circ}\text{CO}$ intermediates from bonding together, and this is also consistent with the experimental result. Therefore, we conclude that the $\text{Cu}/\text{Cu}_2\text{S}$ interface with isolated $\text{Cu}-\text{S}$ motif is active for catalyzing CO_2RR into C_2 product. The energy barrier for $^{\circ}\text{CO}$ hydrogenation to C_1 products at the $\text{Cu}/\text{Cu}_2\text{S}$ interface was also calculated to be 1.04 eV for generating $^{\circ}\text{CHO}$ intermediates (Figures 4c and 4d). Compared with $^{\circ}\text{OCCO}$ formation, this barrier is higher by 0.37 eV, demonstrating the more favorable C_2 production than C_1 at $\text{Cu}/\text{Cu}_2\text{S}$ interface.

To understand the intrinsic nature of $\text{Cu}(\text{S})$ matrix for good C_2 selectivity in depth, the geometric and electronic structures were further analyzed. As shown in Figure 4e, the distance between Cu^0 and $\text{Cu}^{\delta+}$ sites at $\text{Cu}/\text{Cu}_2\text{S}$ interface is modest, even a little smaller than that on pure $\text{Cu}(111)$ surface (2.458 Å vs. 2.579 Å), which conduces to the easy $^{\circ}\text{CO}$ dimerization geometrically. Moreover, as illustrated in Figure S38, $^{\circ}\text{CO}$ occupies the hollow-site and interacts with three Cu atoms (i.e., two Cu^0 and one $\text{Cu}^{\delta+}$) at the $\text{Cu}/\text{Cu}_2\text{S}$ interface, in which the interface $\text{Cu}^{\delta+}$ takes a Bader charge 0.14 |e|. Based on the crystal orbital Hamilton populations (-COHP) and integrated -COHP (-ICOHP) analysis (Figure 4b), the -ICOHP of $\text{Cu}^{\delta+}-\text{CO}$ bond at the $\text{Cu}/\text{Cu}_2\text{S}$ interface rises continuously and reaches a relatively steady value (~ 1.55 eV) at the Fermi level, which is smaller than that of Cu^0-CO bond (1.61 eV) on $\text{Cu}(111)$ surface. The smaller -ICOHP value demonstrates a weaker interaction between $^{\circ}\text{CO}$ species and $\text{Cu}^{\delta+}$ in comparison with the metallic Cu^0 site. Quantitatively, the adsorption energy at the $\text{Cu}/\text{Cu}_2\text{S}$ interface is -1.12 eV (-0.51 eV considering temperature effect at 298 K), which is weaker by 0.29 eV than that on $\text{Cu}(111)$ surface. In other word, the presence of partially oxidized $\text{Cu}^{\delta+}$ at the $\text{Cu}/\text{Cu}_2\text{S}$ interface weakens the $^{\circ}\text{CO}$ adsorption and thus facilitates the CO dimerization to C_2 products.

In addition, regarding previous studies reported S-modified copper catalyst for formate product, we also did some first-principles calculations to verify the peculiarity of our catalyst structure on reaction products. The CO_2 adsorption via the carbon atom instead of di-oxygen generally serves as a key precursor for $^{\circ}\text{COOH}$ formation^[32], and it tends to be further hydrogenated into $^{\circ}\text{CO}$ intermediate rather than the formate product. For our catalyst, it mainly comprises of a majority of metallic Cu^0 with abundant $\text{Cu}/\text{Cu}_2\text{S}$ interfaces, which might be different from the S-modified catalysts reported in previous studies^[33]. At the interface Cu region, the computational result shows that CO_2 tends to adsorb into $^{\circ}\text{CO}_2^{\circ}$ species via carbon atom rather than the di-oxygen atoms. Specifically, we calculated the progress of $^{\circ}\text{COOH}$ formation from $^{\circ}\text{CO}_2^{\circ}$ species, and the reaction barrier is only 0.10 eV with an enthalpy change of -0.66 eV (Figure S39). These $^{\circ}\text{COOH}$ intermediates can be further hydrogenated into $^{\circ}\text{CO}$, a widely accepted precursor prior to creating C_2 products^[34]. On the

contrary, S-modified copper will weaken the CO_2 adsorption and suppress the formation of $^{\circ}\text{CO}$ intermediate. Also, previous studies suggested that the sulfur doping on the copper surface could suppress the HER owing to the slower combined desorption rate of H_2 ^[33]; accordingly, the formate production competes with HER for surface accumulated H^+ , possibly via a Heyrovsky-like mechanism into HCOO^- product^[35].

In light of these analysis above, the superior selectivity of S-HKUST-1 precatalyst on catalyzing CO_2RR into C_2 product, compared with copper or copper sulfide, can be rationalized: benefited from the approximate $\text{Cu}^0-\text{Cu}^{\delta+}$ distance, $^{\circ}\text{CO}$ dimerization step can easily occur at the $\text{Cu}/\text{Cu}_2\text{S}$ interface, and excessively high S-content catalysts (i.e., Cu_2S) features with the long $\text{Cu}-\text{Cu}$ distance, geometrically hindering the $^{\circ}\text{CO}$ dimerization; the $\text{Cu}/\text{Cu}_2\text{S}$ interface presents a weakened binding energy to $^{\circ}\text{CO}$ species, which prompts the $^{\circ}\text{CO}$ dimerization than pure metallic Cu catalyst.

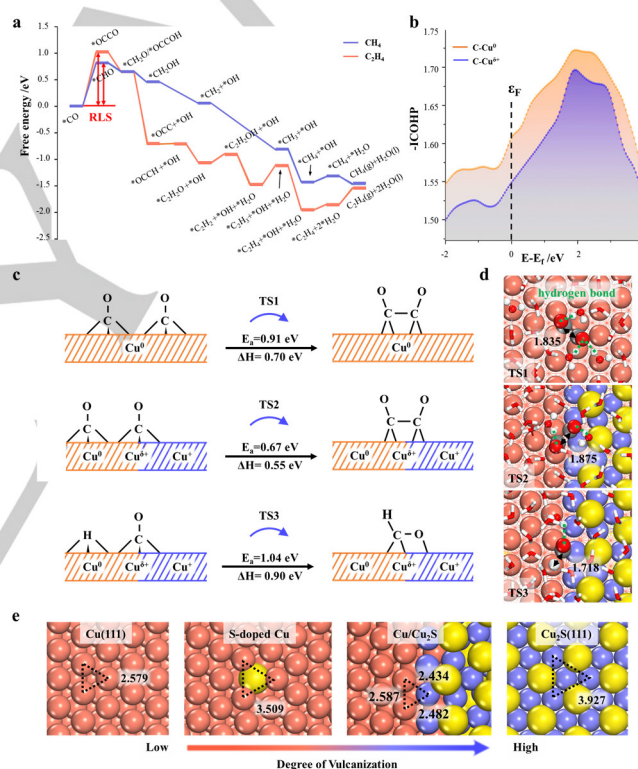


Figure 4. (a) Calculated free energy profiles for CO_2RR to CH_4 and C_2H_4 over pure $\text{Cu}(111)$ surface, indicating the initial $^{\circ}\text{CO}$ hydrogenation and dimerization steps basically determine the CH_4 and C_2H_4 products distribution. Detailed geometries and energetic information are listed in Figure S35 and Table S12. (b) Integrated crystal orbital Hamilton populations (-ICOHPs) curves for $\text{Cu}^{\delta+}-\text{CO}$ and Cu^0-CO bonds. (c) The reaction barriers together with enthalpies, and (d) corresponding transition states configuration for $^{\circ}\text{CO}$ dimerization and hydrogenation over $\text{Cu}(111)$ and $\text{Cu}/\text{Cu}_2\text{S}$ surfaces, respectively. Yellow, red, gray, white, orange and blue balls refer to S, O, C, H, Cu^0 and $\text{Cu}^{\delta+}$ atoms, respectively. (e) Surface configuration (top view) of Cu -based structures with different degrees of vulcanization. The distances between two neighboring Cu atoms on different surfaces are labeled, respectively (in Å).

Conclusion

In this work, we develop a rational predesign strategy to isolate stable Cu-S motif into the HKUST-1 precatalyst. The regenerative Cu(S) matrix exhibits a high C₂H₄ selectivity up to 57.2% at the operated a_j of 400 mA cm⁻². This is the best selectivity and activity value for C₂H₄ among the Cu-based metalorganic and MOFs for CO₂RR electrocatalysts. The operando XAFS analysis along with systematic characterizations demonstrate the stable Cu-S motif during CO₂RR, which can be stabilize more active Cu^{δ+} species. The theoretical results further illustrate the significant importance of the S stabilized Cu^{δ+} at the Cu/Cu_xS_y interface with optimized geometric and electronic structures for *CO dimerization. We believe this work opens up a new avenue to finely predesign precatalysts with stable and active species for C₂ products-selective electrocatalysts.

Acknowledgements

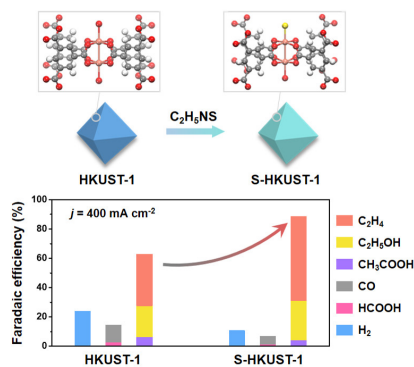
This work was financially supported by the International (Regional) Cooperation and Exchange Projects of the National Natural Science Foundation of China (51920105003), the National Natural Science Funds for Distinguished Young Scholars (51725201), the Innovation Program of Shanghai Municipal Education Commission (E00014), the National Natural Science Foundation of China (51902105, 21873028, 22072045), the Shanghai Engineering Research Center of Hierarchical Nanomaterials (18DZ2252400) and the Shanghai Sailing Program (19YF1411600). The authors acknowledge the support by Shanghai Rising-star and Shuguang Programs (20QA1402400, 17SG30). The authors acknowledge the Fundamental Research Funds for the Central Universities (JKD01211519, JKVJ1211022). Additional support was provided by the Feringa Nobel Prize Scientist Joint Research Center. The authors also thank the Frontiers Science Center for Materiobiology and Dynamic Chemistry. The authors also thank the crew of the BL14W1 beamline at the Shanghai Synchrotron Radiation Facility (SSRF) and the 1W1B beamline of Beijing Synchrotron Radiation Facility (BSRF) for their constructive assistance with the XAFS measurements and data analyses.

Keywords: CO₂ electrocatalysis; ethylene-selective; isolated Cu-S motif; operando XAFS; precatalyst

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The S-HKUST-1 precatalyst with a stable isolated Cu-S motif was prepared by local sulfur doping strategy, which can be *in-situ* reconstructed to obtain Cu(S) matrix with active biphasic copper/copper sulfide interfaces, delivering a current density of 400 mA cm^{-2} with Faradaic efficiency of ethylene to 57.2% in the flow cell.