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1 **Silicon Photoanodes for Solar-Driven Oxidation**
2 **of Brine: A Nanoscale, Photo-Active Analog of**
3 **the Dimensionally-Stable Anode**

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10

11 **Abstract**

12 We report the first results in which ALD-TiO₂ layers electronically couple silicon to an overlying
13 catalyst coating while inhibiting corrosion during brine splitting, a reaction that is capable of
14 generating not only a fuel (H₂) but also a disinfectant (Cl₂ or NaOCl). An n-type silicon photoanode,
15 and a p⁺-silicon anode are protected by a 1.7 nm coating of amorphous TiO₂ and 2 nm coating of Ir.
16 The p⁺-Si/TiO₂/Ir anode is able to effect the oxidation of chloride (0.5 M H₂SO₄, 3.5 M NaCl) in the
17 dark with a low overpotential compared to that for oxidation of water (0.5 M H₂SO₄). The 0.294 V
18 overpotential difference occurs despite the difference in standard redox potentials, $\Delta E = (E^\circ(\text{Cl}_2/\text{Cl}^-) -$
19 $E^\circ(\text{O}_2/\text{H}_2\text{O}))$, is 130 mV. Under 1 sun irradiation a photovoltage of *ca.* 0.566 V is achieved by
20 comparing the illuminated n-Si/TiO₂/Ir photoanode with the dark p⁺-silicon anode. Although the n-
21 Si/TiO₂/Ir photoanode is only one half of an eventual tandem cell needed for photosynthetic brine
22 splitting, its ABPE for chloride oxidation is 1.42%, *ca.* 28 times that for water oxidation. The
23 illuminated n-Si/TiO₂/Ir photoanode remained stable at 1 mA cm⁻² during a six-day
24 chronopotentiometry test.

25

26 **Key words:** silicon; anode; brine; chloride; oxidation; photoelectrochemistry

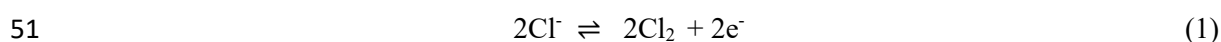
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28 I. Introduction

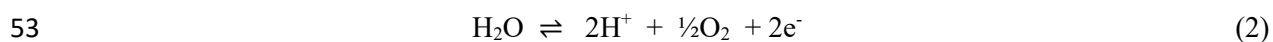
29 Conversion of solar to electrical energy using photovoltaic devices, such as the silicon solar cell is
30 well-established, but solar energy is diurnal and intermittent and so there is least of it when we most
31 need it, i.e. at night in winter. As a consequence, there is great interest in developing efficient,
32 inexpensive and stable solar energy conversion devices that generate readily utilizable chemical
33 fuels.^{1,2} A popular, current approach is to develop a solar-driven, water-splitting system, since it uses
34 the sun's energy to produce hydrogen, which can be stored, readily transported and used when
35 needed.² However, the overpotential for water oxidation, using the best platinum group metal
36 electrocatalysts¹, is still typically 300-400 mV at 10 mA cm⁻² and so represents a major barrier to the
37 creation of a long-lasting, high solar to chemical energy efficiency device.

38 Recently, Nocera et al. suggested that an inexpensive artificial leaf water splitting device will find
39 initial and widespread application in the developing world at less urbanized, off-the-grid, locations.^{3,4}
40 Advantages of such photoelectrochemical cells (PEC's) which combine light absorption and catalyzed
41 electrochemical reactions are 1) the potential simplicity of an integrated design, 2) their compact foot-
42 print and 3) their ability to accommodate less expensive catalysts operating at electrolysis current
43 densities better matched to solar photocurrent densities.

44 Compared to the photocleavage of water, the photo-electrolysis of brine has not been studied nearly as
45 widely, despite the fact that the overpotential for chlorine production is low.⁵ In addition, the
46 electrolysis of salt water: (i) generates H₂, alkali and Cl₂, (or, H₂ and NaOCl, if the Cl₂ and caustic are
47 allowed to combine), (ii) stores approximately the same amount of energy per electron as the water
48 splitting reaction, and (iii) generates a more valuable chemical oxidant feedstock, namely, Cl₂ or
49 NaOCl. However, it is not without its challenges, given the oxidation of chloride to chlorine in acid
50 solution



52 has a significantly higher standard redox potential ($E^\circ(\text{Cl}_2/\text{Cl}^-) = 1.36 \text{ V}$) than that of water



54 ($E^\circ(\text{O}_2/\text{H}_2\text{O}) = 1.23 \text{ V}$). Moreover, chlorine is a very aggressive oxidant, with a solubility in water
55 ($0.092 \text{ mol dm}^{-3} \text{ atm}^{-1}$) that is over 70 x's that of O_2 , thus increasing markedly the likelihood of its
56 reaction with cell components, including, most importantly, the semiconductor photoanode.

57 The industrial production of Cl_2/NaOCl from brine is worth currently \$46 Bn in the USA alone.⁶ Its
58 role in disinfection is important, since the need for potable (i.e. disinfected) water worldwide is great,
59 with a sixth of the world population (ca. 1.1 Bn people) having no access to improved water supplies
60 and with a much greater number consuming contaminated water every day. It is estimated that >
61 4000 children die of diarrhoea alone every day due to poor water quality.⁷ There is, therefore, a
62 compelling need for an inexpensive and compact solar-to-chemical energy conversion device capable
63 of producing not only a fuel (hydrogen), for heating and for electricity (via a fuel cell), when the sun
64 is not shining, but also a disinfectant – such as chlorine/hypochlorite – for use in generating potable
65 water and clean surfaces.

66 Herein we report a silicon photoanode which functions both as a nanoscale version of the
67 dimensionally-stable anode (DSA)⁸, the key technology in industrial chlor-alkali electrochemistry,
68 and as an efficient absorber of solar light. This photoanode could function as a key junction in a
69 multi-junction photoelectrochemical cell⁹ for unassisted brine splitting. Industrial DSA electrodes
70 have a high precious metal (Ir, Ru) loading and thick active layers¹⁰ mixed with TiO_2 and typically
71 coating a Ti substrate.^{11,12} However, such anodes typically operate at current densities that are 100 x's
72 greater than those (ca. 10 mA cm^{-2}) expected in any photosynthetic solar cell and In the reported
73 nanoscale DSA structure, atomic layer deposited TiO_2 is used to achieve stable brine splitting on
74 silicon, which is otherwise unstable under the conditions used in chloride oxidation. The TiO_2 thin
75 film electronically couples an overlying chloride oxidation catalyst film, in the form of an ultra-thin
76 iridium layer, to the silicon substrate, achieving a high yield for chlorine synthesis, photovoltages
77 exceeding 560 mV and greater than 6 days of continuous operation during chronoamperometry
78 testing.

79 **II. Experimental**

80 *Materials*

81 The silicon wafers used here were either: (i) heavily boron-doped p⁺-type Si (100) wafers ($\rho = 0.001$ –
82 $0.002 \text{ } \Omega \text{ cm}$, $500 \text{ } \mu\text{m}$ thick), which were used as conductive silicon substrates to study water and
83 chloride oxidation in the dark or (ii) lightly phosphorous-doped, n-type Si (100) wafers ($\rho = 0.1$ – 0.2
84 $\Omega \text{ cm}$, $500 \text{ } \mu\text{m}$ thick). The wafers were used as received, with a thin ($< 2 \text{ nm}$) SiO₂ layer as prepared
85 by the wafer supplier, Nova Electronic Materials. Onto both wafer types, a 1.7 nm amorphous TiO₂
86 layer was deposited via 30 cycles of atomic layer deposition, ALD, at 170°C with tetrakis-
87 dimethylamido titanium, TDMAT, as the titanium source and H₂O as the oxygen source. In all cases,
88 this was followed by a coating of a 2 nm Ir layer, deposited by electron beam evaporation. The
89 backside contacts for the n-Si and p⁺-Si substrates were e-beam evaporated Al and Pt, respectively.
90 These two electrodes are referred to throughout as n-Si/TiO₂/Ir and p⁺-Si/TiO₂/Ir electrodes. A
91 schematic of the n-Si/TiO₂/Ir photoanode used in this work for the photoelectrolysis of acidified brine
92 is illustrated in **Figure 1**.

93

94 *Methods*

95 All electrochemical and photoelectrochemical experiments were conducted using $0.5 \text{ mol dm}^{-3} \text{ H}_2\text{SO}_4$,
96 in the absence (for water oxidation) and presence (for chloride oxidation) of $3.5 \text{ mol dm}^{-3} \text{ NaCl}$. All
97 electrochemical work was performed using a modified version of the 'silo-like' electrochemical cell
98 used by the McIntyre and Chidsey group,¹³ which is illustrated in **Figure S1**. The cell components
99 and design are described in the supporting information. The gas inlet and outlet of the cell were used
100 in Cl₂ yield measurements to sweep out, using a continuous stream of Ar (flow rate: $100 \text{ cm}^3 \text{ min}^{-1}$),
101 the Cl₂ generated in the electrochemical cell into a 100 cm^3 KI trap the spectrophotometric analysis of
102 the latter allowed the total amount of Cl₂ to be assessed¹⁴. The bulk of the electrochemical
103 measurements were conducted using a Metrohm Autolab (PGSTAT128N) potentiostat. All linear
104 sweep voltammograms, LSVs, were recorded using a 1 mV s^{-1} sweep rate. Electrochemical
105 impedance spectroscopy was performed using a FRA32M module (Metrohm) in order to determine
106 the resistance of electrolytes used in this study and so compensate for the *i*R drop in the cell.

107 A sealed, quartz window recessed in the PTFE cap of the electrochemical cell allowed the flat, n-
108 Si/TiO₂/Ir anode, installed in the base of the cell, to be irradiated. Irradiations were conducted using
109 either: (i) a high power 455 nm LED (HBW = 16 nm; OSLO⁺ PowerStar Deep Blue 455 nm;
110 maximum wattage > 2.2 W) providing an irradiance of ca. 6.5 mW cm⁻² (at ca. 7 cm above the
111 electrode) for the chronopotentiometry experiments or, more usually, (ii) a 150 W solar simulator
112 (Sciencetech SS150W) fitted with a AM1.5 filter, simulating the irradiance of 1 sun (ca. 100 mW cm⁻²,
113 measured using a power meter, ThorLabs, PM100D). All UV/Vis spectra were recorded using an
114 Agilent Cary 6000i UV/Vis/NIR spectrophotometer. IPCE measurements were made using a standard
115 glass vial (A-001056, Biologic) with a PTFE cap holding all electrodes in place and a 1kW Xe-arc
116 lamp (OBB KiloArc), coupled to a 200 mm meter Czerny-Turner monochromator (OBB), and Gooch
117 & Housego (OL756) radiometer were used to provide the monochromatic light and associated
118 irradiance values, respectively. IPCE measurements were conducted on a n-Si/TiO₂/Ir photoanode
119 (area: 0.8 cm²) polarised at 1.8 v vs NHE in an electrolyte volume of 15mL (0.5M H₂SO₄ + 3.5 NaCl).

120 **III. Results and Discussion**

121 *Optical measurements and initial photoelectrochemical studies*

122 Silicon, a candidate material for the low-gap semiconductor of a tandem device, has a band gap of 1.1
123 eV and so absorbs strongly incident light of $\lambda \leq 1127$ nm, as indicated by its reflectance spectrum
124 which has been widely reported.¹⁵

125 In contrast to the absorption spectrum of Si, those of the protecting Ti oxide and the Ir film used in
126 this work have not been reported previously. Thus, the UV/Vis/NIR absorption spectrum of the thin
127 (1.7 nm) TiO₂ film, deposited using ALD on quartz, is illustrated in **Figure 2** and shows that it absorbs
128 no visible light, i.e. it is colourless, and that it absorbs very little UV radiation. Thus, its presence will
129 have very little effect on the amount of light absorbed by the underlying Si. In contrast, the
130 absorption spectrum of the 2 nm Ir film, also illustrated in **Figure 2**, reveals an average absorbance of
131 ca. 0.184 throughout the visible and NIR spectrum, which suggests that only ca. 65.5 % of the
132 incident visible light is transmitted by the Ir film to the underlying Si. Thus, although the deposition

133 of an Ir film is convenient and relatively easy to effect, the use of the Ir catalyst in the form of a
134 continuous film reduces the efficiency of the photoanode significantly by partly obscuring the surface
135 of the Si. This implies that the photocurrent may be improved markedly if a pattern of catalyst islands
136 is used instead of a film. This approach has been adopted in the work of Hu et al¹⁶ who used 100 nm
137 thick Ni-island electrocatalysts patterned in square arrays of 3 μm diameter circles on a 7- μm pitch, to
138 cover thick (44 nm), but 'electronically leaky', TiO_2 protective coatings on an n-Si photoanode film, so
139 that the islands only blocked 14.4% of the electrode surface.

140 In order to quickly compare and contrast the efficacies of the n-Si/ TiO_2 /Ir photoanode for water
141 oxidation (in 0.5 M H_2SO_4) to that of chloride (in 0.5 M H_2SO_4 plus 3.5 M NaCl), cyclic
142 voltammograms photocurrents were recorded under 1 sun irradiation and the results of this work are
143 illustrated in **Figure 4**.

144 Inspection of the results reveals that n-Si/ TiO_2 /Ir photoanode is able to effect the photo-oxidation of
145 chloride to chlorine in acid at lower potentials (typically, ca. 300 mV lower at 1 mA cm^{-2}) than that
146 for water oxidation, despite the fact that $\Delta E = (E^\circ(\text{Cl}_2/\text{Cl}^-) - E^\circ(\text{O}_2/\text{H}_2\text{O})) = 130 \text{ mV}$. This feature
147 results from the much higher overpotential for water oxidation, η_{O_2} , ($\eta_{\text{O}_2} \approx 0.428 \text{ V}$ on a p+-Si/ TiO_2 /Ir
148 anode, vide infra) compared to that for chloride oxidation ($\eta_{\text{Cl}_2} \approx \text{ca. } 0.134 \text{ V}$ on a p+-Si/ TiO_2 /Ir
149 anode, vide infra).

150 In contrast, in 1 M NaOH, the primary chloride oxidation reaction is:



152 where, $(E^\circ(\text{OCl}^-/\text{Cl}^-) = 0.81 \text{ V vs NHE})$, so that ΔE , now equal to: $(E^\circ(\text{OCl}^-/\text{Cl}^-) - E^\circ(\text{O}_2/\text{OH}^-))$, is much
153 larger (406 mV), than the 130 mV in acid, and this mostly off-sets the benefit gained by having η_{Cl_2}
154 $\ll \eta_{\text{O}_2}$. Thus, in 1 M NaOH the CV curves for water and chloride oxidation are no longer well
155 separated, but overlap, so that the photoanode generates a mixture of hypochlorite and O_2 . As a
156 consequence, in order to simplify this study to that of either water or chloride oxidation, but not a
157 mixture of both, the bulk of the work described here was carried out under acidic conditions (0.5 M
158 H_2SO_4) with or without 3 M NaCl. In acidified brine the overall photo-electrochemical reaction is the

159 oxidation of chloride to Cl_2 , i.e. reaction (1), at the n-Si/TiO₂/Ir photoanode, and the concomitant
160 reduction of H^+ to H_2 at the Pt counter electrode, as illustrated in **Figure 1**.

161 It can be shown¹³ that solar irradiation at 1 sun provides ca. 2.7×10^{17} photons cm^{-2} with energies \geq the
162 Si band gap, which in turn should produce a theoretical maximum photocurrent density of ca. 43 mA
163 cm^{-2} . From the results illustrated in **Figure 4**, the light-limited saturation currents for both water and
164 chloride oxidation, achieved at high bias potentials, were found to be ca. 31 mA cm^{-2} , i.e. ca. 72% of
165 the theoretical maximum. Given that ca. 35% of the incident light is obscured by the Ir film, the value
166 of 72% implies that, at high bias potentials, the n-Si/TiO₂/Ir photoanode, is ca. 100% efficient in
167 converting photons to current, i.e. the photogenerated holes are readily and efficiently conducted
168 through the TiO₂ overlayer to the Ir electrocatalyst. IPCE measurements were also made using the n-
169 Si/TiO₂/Ir photoanode biased at 1.8V vs NHE (Figure 4). The IPCE is high, across a wide range of
170 wavelengths, and consistent with the average value of 72% reported above, suggesting that, once
171 corrected for the absorbance due to the catalyst layers, this system, with its 1.8 V bias is ca. 100%
172 efficient in converting photons to current.

173 The broken vertical lines in **Figure 4** represent the thermodynamic redox potentials for water (black)
174 and chloride (red) oxidation in 0.5 M H_2SO_4 and in both photo-electrochemical reactions, reasonable
175 (up to 10 mA cm^{-2} and 1 mA cm^{-2} for Cl^- and H_2O oxidation, respectively) photocurrents are
176 generated, by the n-Si/TiO₂/Ir photoanode at 1 sun illumination, at potentials at and below these
177 thermodynamic limits, suggesting a reasonable degree of solar to chemical energy conversion,
178 although clearly much more so for chloride oxidation. However, as we shall see later, a useful
179 assessment of the applied bias photon efficiency, ABPE, requires the cell to be operated in 2-
180 electrode, rather than 3-electrode mode, and under such conditions the efficiency of cell is
181 significantly reduced.

182 The above results show that, upon band gap illumination of the n-Si, the photogenerated holes in the
183 Si valence band are able to move through the intermediate oxide layers, as illustrated in **Figure 1**, to
184 the Ir layer, where they are then able to oxidize surface adsorbed water, or chloride, from the
185 electrolyte solution. Further work on TiO₂-protected Si anodes,¹⁹⁻²² suggests that the unexpectedly

186 low barrier for hole conduction through the amorphous TiO₂ layer involves hole-tunnelling through an
187 ultrathin (< 2 nm) SiO₂ interface layer and then a polaronic or hopping conduction mechanism
188 through the TiO₂ protective layer to the Ir catalyst film. Importantly, in the absence of light, the
189 current is negligible, i.e. < 1 μA cm⁻², at the potentials used in **Figure 4**, due to the very low thermal
190 population of holes in the n-Si. Thus, in the dark the n-Si/TiO₂/Ir photoanode is not able to effect
191 either the water oxidation or chloride oxidation half-cell reactions, but is able to do so upon
192 illumination with ultra-band gap light, as illustrated by the photoelectrochemical data in **Figure 4**.

193 *LSV's: Tafel slopes, overpotentials and photovoltages*

194 Linear sweep voltammograms, LSVs, for water oxidation, i.e. reaction (2), were recorded for the n-
195 Si/TiO₂/Ir photoanode, under 1 sun illumination, and the p+-Si/TiO₂/Ir anode (in the dark) in 0.5 M
196 H₂SO₄ and the results are illustrated in **Figure 5(a)**. Similarly, LSVs for *chloride oxidation*, i.e.
197 reaction (1), were recorded for the n-Si/TiO₂/Ir photoanode – under 1 sun illumination, and the p+-
198 Si/TiO₂/Ir anode (in the dark) in a (0.5 M H₂SO₄ plus 3.5 M NaCl) electrolyte and the results are
199 illustrated in **Figure 5(b)**. The broken vertical lines highlight the formal redox potentials, E^o, of the
200 O₂/H₂O (black) and Cl⁻/Cl₂ (red) couples in 0.5 M H₂SO₄ and (0.5 M H₂SO₄ plus 3.5 M NaCl)
201 electrolytes, respectively, calculated, using the Nernst equation, to be: 1.212 and 1.328 V,
202 respectively.

203 The data associated with each of the LSV's illustrated in **Figure 5** provides a good fit to the Tafel
204 equation, i.e. ²³

$$205 \quad \eta = a + b \cdot \log(i) \quad (4)$$

206 where, η = overpotential (= applied bias, V_b, - E^o), *a* and *b* are constants (units: V) and *i* is the current
207 density (units: mA cm⁻²). It follows from eqn (4) that the value of the constant '*a*' is the overpotential
208 necessary to generate a current/photocurrent density of 1 mA cm⁻² and '*b*' is the Tafel slope, which can
209 provide an insight into the mechanism that underpins the associated electrochemical oxidation. A
210 summary of the *a* and *b* terms determined by this analysis for the two electrodes in the two different
211 electrolytic solutions, derived from Tafel plots of the data in **Figure 5**, is given in **Table 1**. The '*a*'

212 values in parenthesis for water oxidation are those reported by Chen et al.¹³ for similar electrodes,
213 although with a thicker Ir layer, i.e. 3 nm rather than the 2 nm used here, and in 1 M H₂SO₄.
214 Reassuringly both studies report a very similar photovoltage, i.e. 0.568 V compared with the value of
215 0.532 V reported by Chen et al.¹³. As noted by these workers, this photovoltage is similar to that of
216 the best Si photoelectrochemical solar cells.²⁴

217 **Table 1** shows the Tafel slope to be between 70-80mV/decade, with previous works reporting Tafel
218 slopes in the range of 34-54mV/decade^{5,25,26} for iridium oxide catalysts. Slopes around 40mV/decade
219 are commonly attributed to a Heyrovsky-type rate limiting step,²⁷ and are associated with
220 electrochemical desorption being slow.⁵ It is worth noting that these previous works study iridium
221 oxide materials, synthesized either by electrochemical cycling or thermal decomposition²⁵. This work
222 employs an iridium metal catalyst that may not fully oxidize during operation, which may result in
223 different Tafel slopes for those more reduced regions. This is in agreement with Tilak's work, which
224 subjected iridium anodes to cathodic current prior to testing and saw higher Tafel slopes (46-
225 54mV/decade) for a more reduced iridium catalyst.⁵ Despite these slightly sub-optimal Tafel slopes,
226 these ALD-TiO₂ protection layers have been shown to support a wide variety of catalyst materials for
227 the water oxidation reaction^{19,28}, implying the protection layer is robust enough to support further
228 optimization of the catalyst layer.

229

230 *Cl₂ yield and stability*

231 As note earlier, examination of either the CV data in **Figure 4**, or the LSV data in **Figure 5**, reveals a
232 significant separation (i.e. ca. 0.290 V from data in **Table 1**) between the two photocurrent LSV plots
233 for the n-Si/TiO₂/Ir, under 1 sun illumination, in the two different electrolytes, 0.5 M H₂SO₄ and (0.5
234 M H₂SO₄ + 3.5 M NaCl), respectively. A similar separation is found for the p+-Si/TiO₂/Ir anode (i.e.
235 0.294 V from data in **Table 1**). This feature suggests that when operated at a modest photocurrent, or
236 current, (i.e. 1-2 mA) the n-Si/TiO₂/Ir photoanode, or p+-Si/TiO₂/Ir anode, should, almost exclusively,
237 mediate the oxidation of chloride, reaction (1), rather than that of water, reaction (2), so that the yield

238 of chlorine should be very high. In order to test this prediction, the Si/TiO₂/Ir photoanode (under 1
239 sun irradiation, polarized at 0.72 V vs Ag/AgCl) and the p⁺-Si/TiO₂/Ir anode (in dark, polarized at
240 1.18 V vs Ag/AgCl) were operated in chronoamperometric mode for 1 hour, during which not only
241 was the current monitored but also any chlorine generated in the electrochemical cell was swept, by a
242 continual stream of Ar, from the electrolyte to an aqueous trap solution (100 cm³), comprising a
243 mixture of KI (0.36 M), NaOH (0.025 M) and potassium hydrogen phthalate (0.049 M)²⁹. Previous
244 work has demonstrated that the latter solution acts as a very efficient trap for chlorine in a stream of
245 an inert gas, but not for oxygen. In the trap the Cl₂ reacts with the iodide in the trap solution to form
246 tri-iodide, the concentration of which can then be assessed spectrophotometrically, given the molar
247 absorptivity $\epsilon(\text{I}_3^-) = 26400 \text{ L cm}^{-1} \text{ mol}^{-1}$ at 353 nm.¹⁴

248 **Figure 6** illustrates the chronoamperograms recorded for the Si/TiO₂/Ir photoanode and the p⁺-
249 Si/TiO₂/Ir anode in an electrolyte of (0.5 M H₂SO₄ + 3.5 M NaCl). The initial drift downwards in
250 current does not appear to be due to the loss of electrocatalyst as the same feature was observed in a
251 repeat experiment using the same, used, electrodes and was also observed when an Ir rod was used as
252 the anode. Instead, it appears to be due to gas bubble formation on the surface of the electrodes.
253 From the ratio of the total amount of triiodide trapped (3.13×10^{-6} moles for the n-Si/TiO₂/Ir
254 photoanode) to the total amount of charge passed ($Q = 0.71 \text{ C}$ for the n-Si/TiO₂/Ir photoanode)
255 Faradaic efficiencies for Cl₂ production were determined to be 85% and 92% for the nSi/TiO₂/Ir
256 photoanode and the p⁺-Si/TiO₂/Ir anode, respectively. Similar yields were determined using an Ir rod
257 as the anode and also a dimensionally stable anode, comprising a film of RuO₂/TiO₂ on a Ti foil³⁰,
258 and the slightly less than 100% Faradaic efficiencies for Cl₂ production for all these electrodes was
259 attributed to a small degree of reaction of the very aggressively oxidising chlorine with the
260 components of the system as it was swept from the electrochemical cell to the trap solution.

261 Chlorine is a very reactive oxidising agent, thus any protective coating has to be sufficiently robust
262 *chemically* to withstand the highly corrosive action of chlorine in a highly acidic environment.
263 Without this TiO₂ coating, even with an Ir catalyst layer, none of the p⁺-Si and n-Si anodes are stable
264 and able to effect the oxidation of either water or chloride. In order to probe the stability of the TiO₂

265 protective coating and the overall photostability of the photoanode, a chronopotentiogram was
266 recorded, with the current set at 1 mA cm^{-2} , for the Si/TiO₂/Ir photoanode (irradiated with a UV 455
267 nm LED; 6.5 mW cm^{-2}) over a period of ca 6 days and the results are illustrated in **Figure 7**.

268 The short disruption in the current –time plot at 3 d, see **Figure 7**, was due to Cl₂ bubble formation,
269 which – once cleared – allowed the photoelectrode to function as before. The %Cl₂ yield was
270 measured for the first and last hour of this run and found to be in both cases ca. 85% as in **Figure 6**.

271 The results illustrated in **Figure 7** indicate that the n-Si/TiO₂/Ir photoanode is stable over a 6-day
272 period when used to photo-electrochemically oxidized chloride to chlorine under acidic conditions.

273 Films as-deposited are conformal, metallic iridium films, and remain unbroken after stability testing.
274 X-ray Photoelectron Spectroscopy confirms the iridium oxidizes during stability measurements, but
275 retains both oxide and metallic components. (see **Figures S2-4**) Scanning Auger Electron
276 Microscopy confirms the iridium film remains intact, with only slight carbon contamination. (**Figure**
277 **S4**)

278 *Efficiency*

279 Finally, the n-Si/TiO₂/Ir photoanode was used in a 2-electrode format, using a wound Pt wire (surface
280 area = 2.5 cm^2) as the counter electrode and the photocurrents, i , measured under 1 sun illumination as
281 a function of applied bias, V_b (in volts), in the following electrolytes: 0.5 M H₂SO₄ (water oxidation)
282 and (0.5 M H₂SO₄ plus 3.5 M NaCl) (chloride oxidation). These data were then used to calculate the
283 applied bias photon-to-current efficiency (ABPE) for the photocleavage of water and the
284 photoelectrolysis of the acidified brine by the n-Si/TiO₂/Ir photoanode under 1 sun illumination, based
285 on the following expression:

$$286 \quad \text{ABPE} = i(\text{mA cm}^{-2}) \times (E_{\text{ref}} - V_b) / P_{\text{total}} (\text{mW cm}^{-2}) \quad (5)$$

287 where, $E_{\text{ref}} = 1.212 \text{ V}$ in 0.5 M H₂SO₄ (for water oxidation) and 1.328 V in (0.5 M H₂SO₄ plus 3.5 M
288 NaCl) (for chloride oxidation) and $P_{\text{total}} = \text{incident illumination power density} = 100 \text{ mW cm}^{-2}$; note:
289 this equation assumes 100% Faradaic efficiency in the production of O₂ or Cl₂.

290 The plots of ABPE vs V_b are illustrated in **Figure 8** and reveal maximum values for water splitting
291 and acidified brine photoelectrolysis of 0.05 % and 1.42 %, respectively. The 28 times increase in
292 efficiency for the photo-oxidation of the brine, compared to that of water in acid solution is primarily
293 due to the fact that the overpotential for the former reaction is much lower than that for the latter, see
294 **Table 1**, so that a greater fraction of the photovoltage, ca. 560 mV is utilized for fuel (H_2) and
295 disinfectant (Cl_2) production. The ABPE value of 1.42 % compares very well with that of 0.6%
296 reported by Kim et al.³¹ for their record-breaking single crystal, worm-like hematite photoanodes for
297 water oxidation, although, in the latter case, the value of 0.6% is flattered by the use of a 3-electrode
298 cell, which excludes any loss in efficiency due to the counter electrode reaction, as is found for many
299 reported ABPE and Solar to Hydrogen Efficiency (STH) values.³² As noted earlier, ultimately the
300 photoanode will be paired with an appropriate photocathode in a tandem cell, or be made part of a
301 photovoltaic triple junction, which will reduce significantly the value for V_b and so improve markedly
302 the value of ABPE. However, as it stands this is the first use of a low bandgap semiconductor
303 photoanode to photoelectrolyze (acidified) brine so as to generate a fuel (H_2) and a disinfectant (Cl_2)
304 with an efficiency superior to that of leading reported photoanodes used to photodissociate water.³¹

305 **IV. Conclusion**

306 An n-type silicon photoanode, utilizing a 2 nm thin film iridium catalyst protected by a 1.7 nm
307 thickness coating of atomic layer deposited amorphous TiO_2 , is able to effect the photo-oxidation of
308 water to O_2 and of chloride to Cl_2 , with an incident photon-to-current efficiency that increases with
309 increasing bias voltage, eventually reaching ca. 100% efficiency if light loss due to Ir film absorption
310 is taken into consideration. In 0.5 M H_2SO_4 acid, the photo-oxidation of water occurs at potentials
311 that are ca. 290 mV greater than that for chloride oxidation (0.5 M H_2SO_4 plus 3.5 M NaCl), which
312 allows the n-Si/ TiO_2 /Ir photoanode (1 sun illumination) and dark p⁺-Si/ TiO_2 /Ir anode to generate Cl_2
313 with a high (>85 %) yield. Applied bias photon-to-current efficiencies of 0.05% and 1.42 % were
314 determined for the n-Si/ TiO_2 /Ir photoanode for water and chloride oxidation respectively. The latter
315 value is particularly encouraging, especially given the potential for improvement when the n-
316 Si/ TiO_2 /Ir photoanode is coupled to an appropriate photocathode in a tandem cell, or used in a triple

317 junction photovoltaic. This initial study illustrates the effectiveness of layered ALD-TiO₂ and Ir thin
318 film coatings in simultaneously protecting silicon photoanodes from corrosion and achieving low
319 overpotentials and large photovoltages to generate both a fuel but a disinfectant and a water
320 disinfectant in a photoelectrochemical device.
321

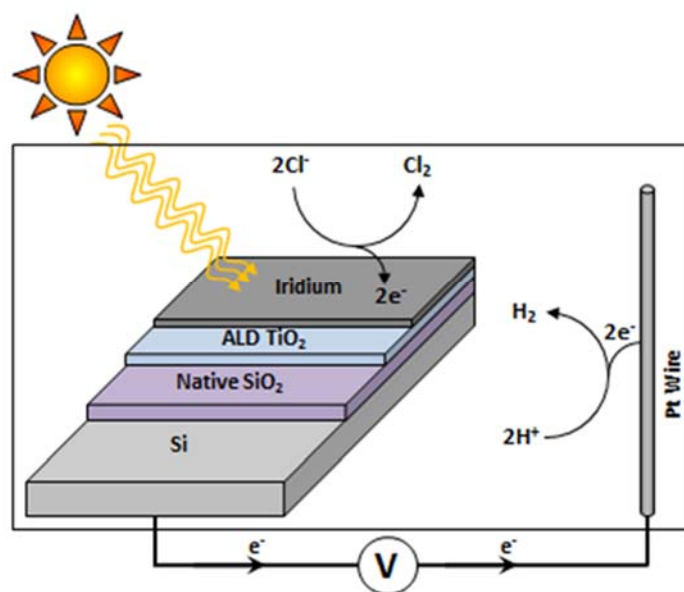
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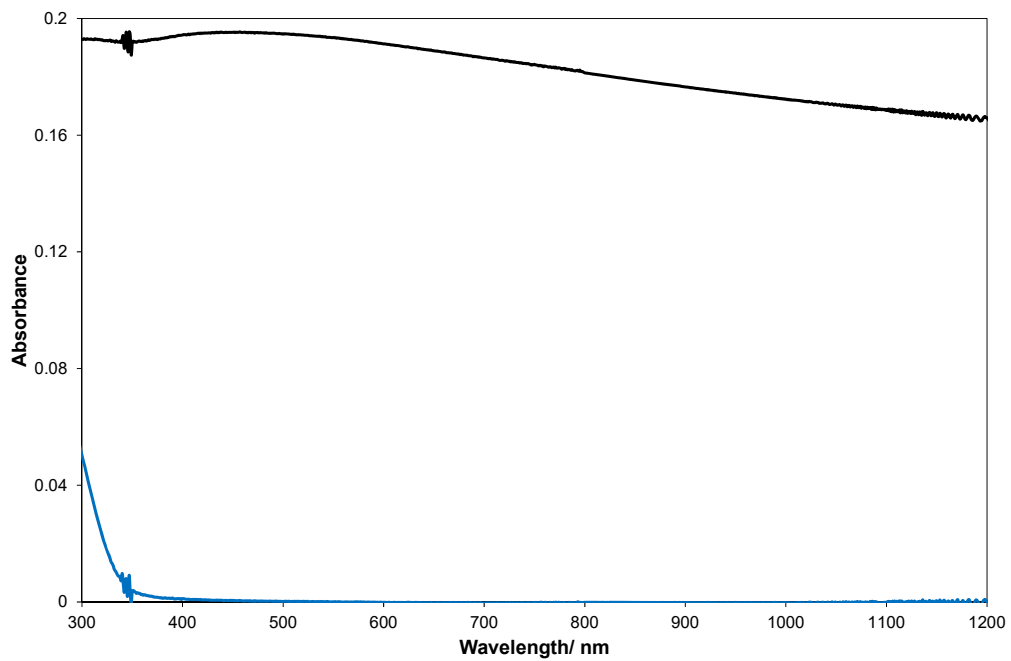
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401

402 **Figure 1:** Schematic illustration of the n-Si/TiO₂/Ir photoanode used to photo-oxidize chloride (3.5 M
 403 NaCl) to chlorine in acid (0.5 M H₂SO₄)

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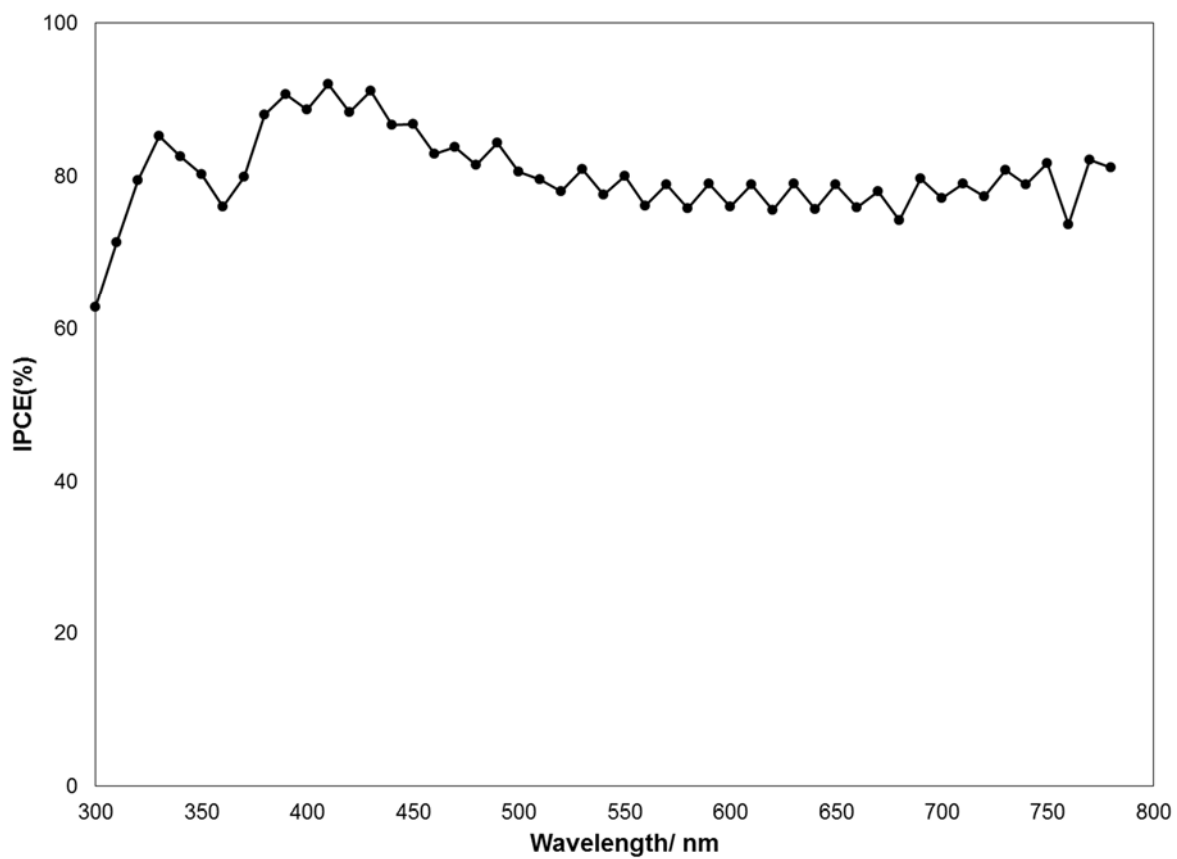


405

406 **Figure 2:** UV/Vis/NIR spectra of the following films on quartz: (i) 2 nm of evaporated Ir (black line)

407 and (ii) 1.7 nm of ALD-deposited amorphous TiO₂ (blue line).

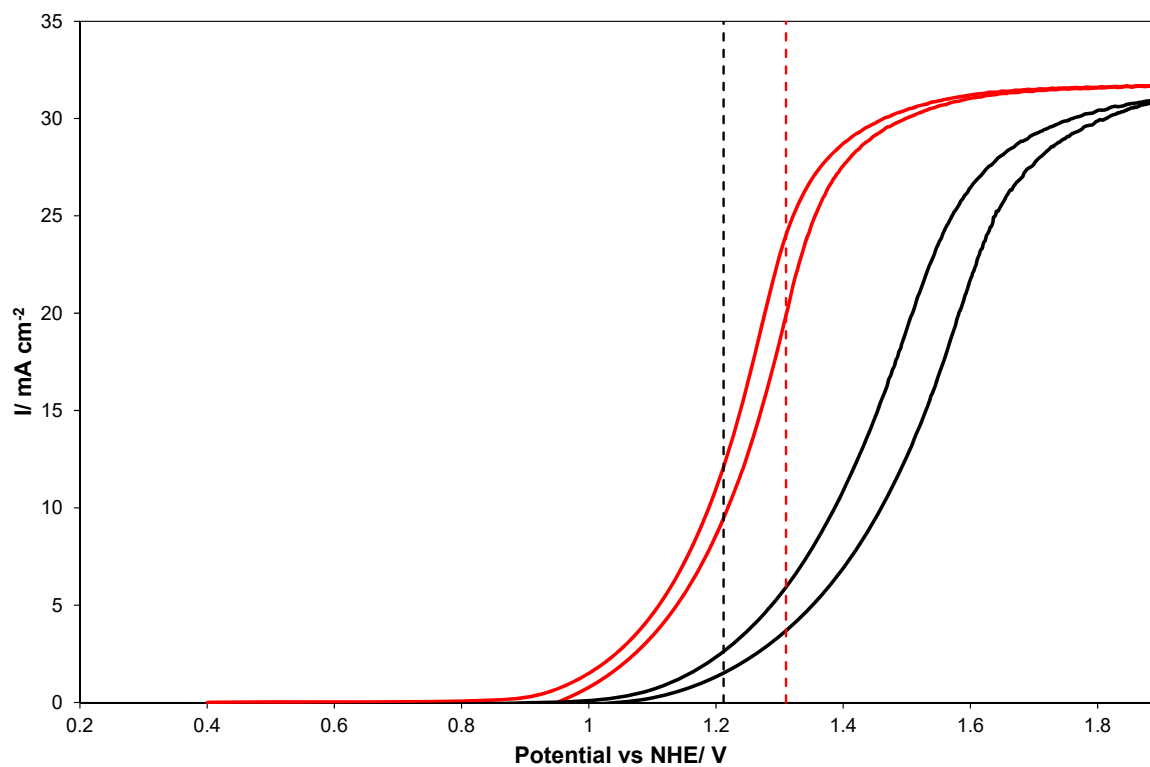
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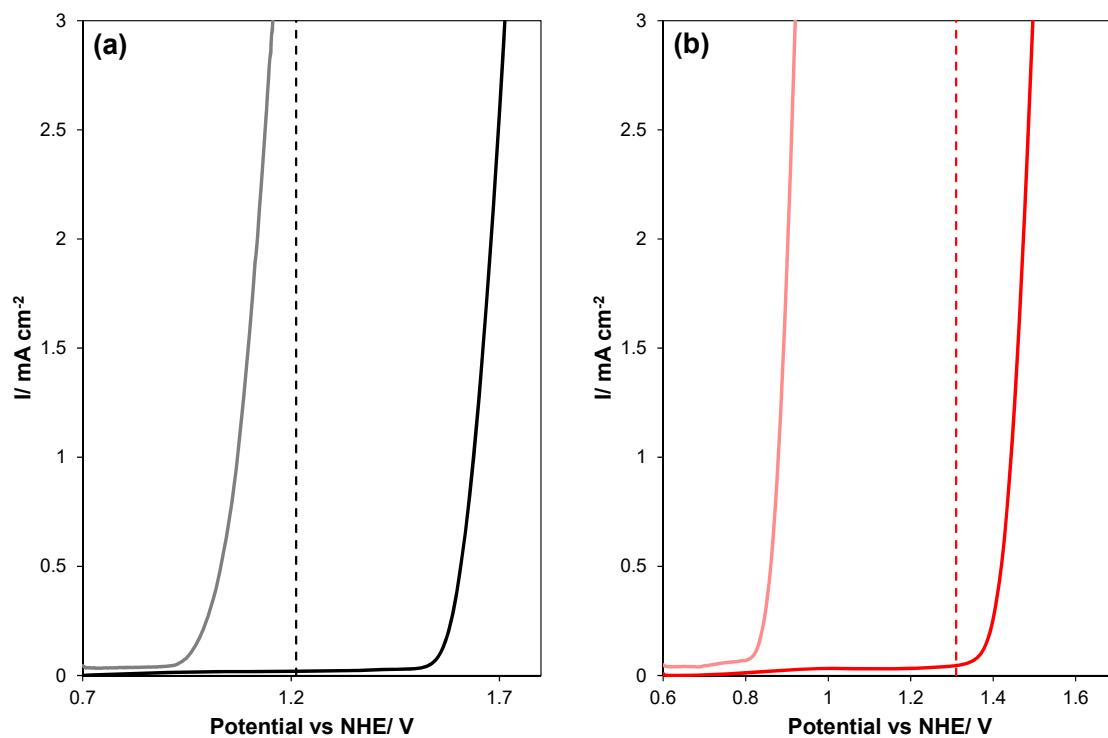
410 **Figure 3:** IPCE spectrum of the n-Si / 2 nm TiO₂ / 2nm Ir anode at 1.8V vs NHE and a 0.8cm² active
411 area. Electrolyte was 0.5 M H₂SO₄ + 3.5 M NaCl.

412



413

414 **Figure 4:** CVs for the n-Si/TiO₂/Ir photoanode under 1 sun irradiation, sweep rate: 100 mV s⁻¹
 415 recorded in 0.5 M H₂SO₄ (black line) or 0.5 M H₂SO₄ plus 3.5 M NaCl (red line) under 1 sun
 416 irradiation, i.e. 100 mW cm⁻². The broken vertical lines represent the thermodynamic potentials for
 417 reaction (1) (broken red line) and reaction (2) (broken black line).



418

419 **Figure 5:** LSV curves for (from left to right): a $\text{p}^+\text{-Si/TiO}_2/\text{Ir}$ anode (in the dark) and an $\text{n-Si/TiO}_2/\text{Ir}$,
 420 under 1 sun irradiation, in (a) $0.5 \text{ M H}_2\text{SO}_4$ (black lines) and (b) ($0.5 \text{ M H}_2\text{SO}_4 + 3.5 \text{ M NaCl}$), (red
 421 lines) respectively. The broken vertical lines in (a) and (b) represent the thermodynamic potentials for
 422 reaction (1) (red line) and reaction (2) (black line), respectively.

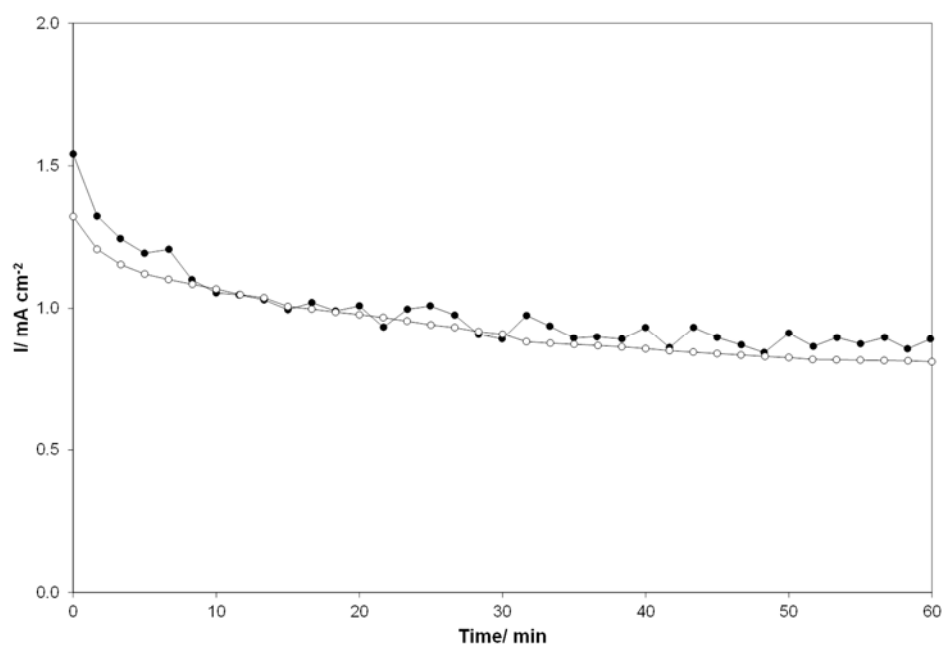
423 **Table 1:** Results of Tafel plot analysis of LSVs for the Si/TiO₂/Ir photoanode and p⁺-Si/TiO₂/Ir
 424 anode

Anode	a/V	b/V	$V_{\text{photo}}^{\dagger}/V$
Water oxidation (0.5 M H₂SO₄)			
p ⁺ -Si/TiO ₂ /Ir	0.428 (0.332) [‡]	0.122	0.568 (0.532)
n-Si/TiO ₂ /Ir*	-0.140 (-0.200)	0.144	
Chloride oxidation (0.5 M H₂SO₄ + 3.5 M NaCl)			
p ⁺ -Si/TiO ₂ /Ir	0.134	0.089	0.564
n-Si/TiO ₂ /Ir*	-0.430	0.069	

425 * 1 sun irradiation; 1100 mW cm⁻²

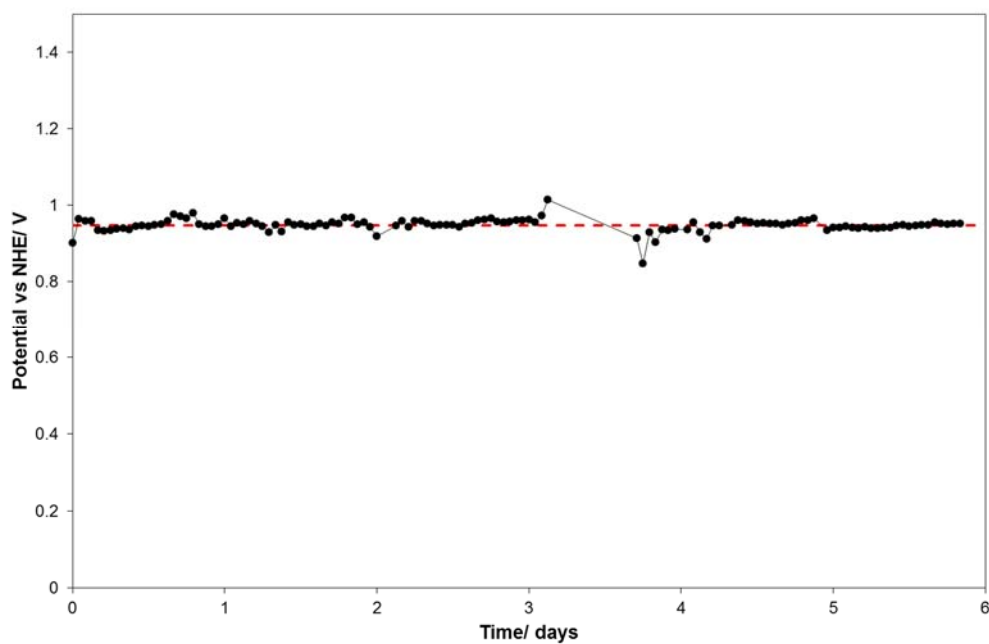
426 †: $V_{\text{photo}} = \text{photovoltage} = a(\text{n-Si/TiO}_2/\text{Ir}^*) - a(\text{p}^+\text{-Si/TiO}_2/\text{Ir})$

427 ‡: Values reported by Chen et al in 1 M H₂SO₄



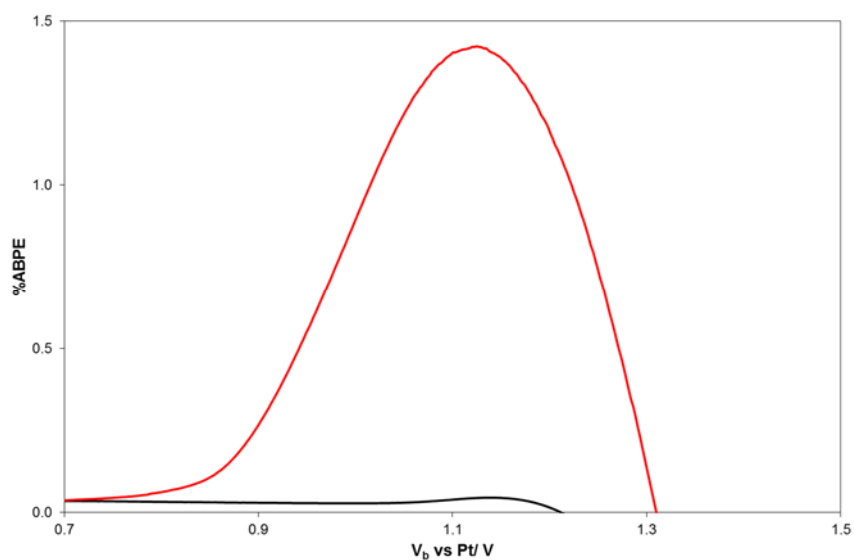
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429 **Figure 6:** Chronoamperograms recorded using the electrochemical cell illustrated in figure S1 and
 430 either: (i) a Si/TiO₂/Ir photoanode, ●, (under 1 sun irradiation, polarized at 0.72 V vs Ag/AgCl), or
 431 (ii) a p⁺-Si/TiO₂/Ir anode, ○ (in dark, polarized at 1.18 V vs Ag/AgCl) using a electrolyte of: (0.5 M
 432 H₂SO₄ + 3.5 M NaCl).



433

434 **Figure 7:** Six-day chronopotentiometry runs, at 1 mA cm^{-2} , recorded using the electrochemical cell
435 illustrated in **Figure S1** with an n-Si/TiO₂/Ir photoanode, ●, (under 455 nm LED irradiation, 6.5 mW
436 cm^{-2}), with an electrolyte of (0.5 M H₂SO₄ + 3.5 M NaCl) and the current set at 1 mA cm^{-2} . The
437 average applied potential was 0.95V (broken red line)



438

439 **Figure 8:** ABPE vs bias potential with respect to the Pt counter electrode, V_b plot for the n-Si/TiO₂/Ir
440 photoanode under 1 sun illumination in 0.5 M H₂SO₄ (for water oxidation – black line) and in 0.5 M
441 H₂SO₄ plus 3.5 M NaCl (for chloride oxidation – red line)